

1 **Biogas upgrading with novel cellulose nano-crystals and polyvinyl**  
2 **amine nanocomposite membranes**  
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## 18 **Abstract**

19 A novel crystalline nano cellulose (CNC) and polyvinyl amine (PVAm) based nanocomposite  
20 membranes were synthesized and evaluated for biogas upgrading. Different concentrations of  
21 CNC was incorporated in 3wt % PVAm solution on commercial polysulfone (PSf) sheet using dip  
22 coating method. The effect of feed pressure (5, 10 and 15 bar) was investigated for the CO<sub>2</sub>/CH<sub>4</sub>  
23 separation. The incorporation of CNC increased the crystallinity of membranes. The thickness of  
24 selective layer enhanced to 2.16 μm from 1.5 μm with increasing concentration of CNC. However,  
25 degree of swelling reduced from 75.88% to 68.93 with CNC concentration at 1.5 wt.%. The best  
26 results were shown by PVAm membrane with 1 wt. % CNC concentration i.e. CO<sub>2</sub> permeance of  
27 0.0216 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr and selectivity (CO<sub>2</sub>/CH<sub>4</sub>) of 41. The permeance decreased  
28 approximately 1.8 folds for PVAm/1CNC membrane with the increase in pressure from 5 to 15  
29 bar. However, selectivity dropped from 41 to 39 for formulated membranes.

30

31 **Keywords:** Polyvinyl amine; Cellulose nano-crystals; Biogas upgrading; Nanocomposite  
32 membranes; Fixed site carrier membranes.

## 33 **1 Introduction**

34 Reduction in Greenhouse gas (GHG) emission is driving the global concerns due to its profound  
35 impacts on our climate. Carbon dioxide (CO<sub>2</sub>) a primary greenhouse gas, is mainly produced by  
36 burning of fuels and is estimated that the amount of CO<sub>2</sub> will increased to 37 Gt by 2035 [1].  
37 Currently, 80% of world's energy demands are fulfilled by fossil fuels. If we follow the same  
38 trend, fossil fuels reservoirs will be consumed in approximately the next 100 years [2].

39 Furthermore, in order to limit global warming, it is required that energy use would have to be  
40 totally decarbonizes and renewable must provide 65% of global energy demands by 2050 [3].  
41 Therefore, it is necessary to find the sustainable and renewable energy resources with low carbon  
42 emission. Currently, 18.6% of total world's energy demands are being fulfilled by renewable  
43 energy sources. However, bioenergy accounts for approximately 14% [4]. Hence, bioenergy would  
44 be expected to the most potential and sustainable source of renewable energy for future global  
45 primary energy mix in 2050 [5].

46  
47 Biogas is a form of bioenergy and product of Anaerobic Digestion (decomposition in an oxygen  
48 deficient environment) of organic waste. It is mainly comprising of CH<sub>4</sub> (50-70) % and CO<sub>2</sub> (30-  
49 50) %. The relative concentration of these two gases largely depends on nature of raw material and  
50 pH of bioreactor [6]. Biogas is being used for heating, production of steam and generation of  
51 electricity. However, after improving energy contents it can also be used as fuel for vehicles and  
52 grid stations. Presences of CO<sub>2</sub>, mainly reduces the calorific value of biogas and limit its  
53 utilization. Upgraded biogas is also called as bio methane (>95 % CH<sub>4</sub> contents) and can meet the  
54 technical requirements to replace the natural gas. Furthermore, the bio methane fuel has potential  
55 to reduce the non-methane volatile organic compounds emission by 50% and NO<sub>x</sub> emissions by  
56 25%. Furthermore, a significant reduction in particulate emission [7].

57  
58 Biogas is upgraded by various techniques such as; water or amine scrubbing, pressure swing  
59 adsorption, membrane technology, and absorption [2]. The membrane separation is a proven green  
60 technology with cost effective CO<sub>2</sub> capture solution, and reduced footprints. Membrane  
61 technology has also been proven beneficial for low gas volumes and high CO<sub>2</sub> contents [8].

62 Therefore, membrane technology is highly recommended for the biogas upgrading by CO<sub>2</sub>  
63 removal [9]. Various strategies have been employed in past to manufacture polymeric membrane  
64 with high efficiency, cost effectiveness and ease of fabrication. However, due to transport  
65 mechanism mainly based molecular sieving and kinetic diameter, the inherent trade-off between  
66 selectivity and permeability of polymeric membranes is a challenge. Therefore, to overcome this  
67 limitation, Facilitated Transport Membranes (FTM) were first introduced as Supported Liquid  
68 Membranes (SLM). The moveable carriers react with dissolved CO<sub>2</sub>. This complex is then  
69 transported across the membrane by solution diffusion mechanism. However, leakage of carrier in  
70 permeate and loss of solution by evaporation reduced the membrane performance with time. To  
71 overcome this problem, a new class of membranes has been evolved known as Fixed Site Carrier  
72 (FSC) membranes [10, 11].

73  
74 In FSC, the carrier is covalently bounded to the main polymer matrix. However, it reduced the free  
75 mobility of carrier but enhanced the overall stability and performance of membranes. Recently,  
76 research has been focused to make membrane material more hydrophilic to take the advantage of  
77 liquid membranes in highly swollen conditions [12-15]. The FTM that works under highly swollen  
78 conditions facilitate the CO<sub>2</sub> transport as bicarbonate ion (HCO<sub>3</sub><sup>-</sup>) through the membrane [16].  
79 Utilization of such membranes have been reported in the literature and is revealed from the results  
80 that the degree of swelling is directly related to membrane performance [15, 17, 18]. Furthermore,  
81 number of nano filler and carrier molecules has been incorporated to enhanced swelling as well as  
82 affinity of composite membranes for CO<sub>2</sub> [13, 19].

83

84 Among different polymers used for acid gas separation, the polyvinyl amine is the most promising  
85 one. Due to the presence of abundant amine group and high degree of hydrophilicity, it gives high  
86 permeability and selectivity for CO<sub>2</sub>. Furthermore, PVAm is easily soluble in the water at room  
87 temperature [10, 20]. PVAm has been extensively investigated for CO<sub>2</sub> separation applications  
88 alone or with different combinations of fillers in mixed matrix membranes [10, 12, 21, 22]. The  
89 structure of PVAm consists of amine group in its chain (-NH<sub>2</sub>) which has a natural affinity for  
90 CO<sub>2</sub>. It acts as fixed site carriers and facilitate transport of CO<sub>2</sub> across the membrane. Recently,  
91 Zhao et al. has used PVAm in mixed matrix composite membranes with PANI/PS and results  
92 showed that the presence of PANI nanoparticles in PVAm matrix enhanced separation  
93 performance of composite membranes [23]. Further, Ming Wang and Zhi Wang et al. incorporated  
94 inorganic fillers such as MWCNT, SiO<sub>2</sub> and ZSM-5 and study their interfacial properties. The  
95 study suggested that addition of nanofillers to PVAm matrix is an effective way to improve  
96 interfacial properties. However, better results could be obtained if inorganic filler and polymer has  
97 same functional groups [22]. But, stability issues of PVAm particularly at high pressure can be  
98 overcome by using high molecular weight PVAm or by introducing finely dispersed second phase  
99 in polymer matrix with high mechanical strength.

100

101 The Crystalline Nano cellulose (CNC) has been used in this research due to its high affinity with  
102 water and reinforcing nature [13, 24-27]. Cellulose fibers have hierarchical microstructures and on  
103 acid hydrolysis give nanostructure of highly crystalline regions (CNC) and amorphous regions  
104 (CNF) [13, 28-30]. D- glucopyranose (C<sub>6</sub>H<sub>11</sub>O<sub>5</sub>) is a major component associated by β (1,4) links  
105 which is the repeating unit of cellulose [31]. The degree of polymerization of cellulose is difficult  
106 to determine but is reported to be near 10,000 if the molecular weight is around 3.2 x 10<sup>6</sup> g/mol.

107 CNCs have become the center of attention for researchers due to its unique properties that include  
108 outstanding mechanical attributes, reinforcing capabilities, low density, biodegradability and  
109 excellent surface area per unit mass[13, 18, 24, 28]. Moreover, cellulose is being abundantly used  
110 in bio and nanocomposites. Due to biodegradable nature, cellulose has replaced multiple synthetic  
111 fibers which also contribute in polluting environment. Cellulose is also being used as nonstructural  
112 biocomposite in doors, windows, ceiling tiles etc. [32]. Recently, CNCs have been reported as an  
113 additive with PVA and showed enhanced results of CO<sub>2</sub> separation up to 15 bars. Furthermore,  
114 NFC has also been reported for enhanced performance of composite membranes. CNCs disperse  
115 along the polymeric matrix and help in moisture uptake and promotes swelling. This moisture  
116 content helps to increase the rate of facilitated transport of CO<sub>2</sub> across the membrane [18].  
117 Furthermore, PVA has also been chemically cross-linked with CNCs resulting in excellent thermal  
118 stability and reinforcement capability. Cross-linked PVA/CNCs have also been utilized in  
119 biocompatible electronic skin sensor system [33, 34].

120

121 Thermodynamic properties of polymers play a vital role in the separation performance of  
122 membranes. Flory-Huggins theory describes the thermodynamics of polymer solutions and blends.  
123 It is a lattice model that explains the non-ideality of polymer mixtures. Comprehensive  
124 thermodynamic studies of polymeric blends have been carried out by Rana et al., for example,  
125 polyvinyl esters and polyacrylates, polystyrene-co-acrylonitrile and polyphenyl acrylate etc.  
126 Hydrogenated polymers were used as analogues of respective polymers and interaction energy  
127 densities were calculated [35-37]. However, in this work, polysulfone (PSf) and PVAm does not  
128 form blend and are chemically inert. Therefore, thermodynamics of these polymers have not been  
129 covered in this work.

130 This research work is carried out to improve the mechanical properties and water affinity of PVAm  
131 membranes to enhance CO<sub>2</sub> separation at moderately high pressures. CNC has been incorporated  
132 in PVAm matrix in order to get beneficial results. There has been no chemical crosslinking  
133 between PVAm and CNC. The membranes were investigated for the optimized concentration of  
134 CNC in 3% PVAm solution. Effect of addition of different concentrations of CNC on degree of  
135 swelling was investigated. The SEM analysis of membranes was conducted to find the effect of  
136 incorporation of CNC on morphology and thickness of selective layer. The effect on degree of  
137 crystallinity of PVAm/CNC nanocomposite membranes was investigated using XRD. The  
138 membrane rig used for CO<sub>2</sub> permeation testing was specially designed and has ability to work  
139 under humid conditions at moderately high pressure. Membranes were investigated under highly  
140 swollen conditions at 5, 10 and 15 bars. The results will be interpreted in terms of permeance of  
141 CO<sub>2</sub> and CH<sub>4</sub> and selectivity of CO<sub>2</sub>/CH<sub>4</sub>.

## 142 **2 Experimental**

### 143 **2.1 Materials**

144 Ultrafiltration flat sheet membrane of Polysulfone (PSf) (Molecular weight cut-off 50,000) of the  
145 commercial grade was purchased from Alfa Laval. CNCs were acquired from Cellulose Lab,  
146 Canada. The average width and length of CNC was 12nm and 170nm, respectively. Polyvinyl  
147 amine (MW 17,000-20,000) was purchased from Sigma Scientific. The solvent used for casting of  
148 membranes was deionized water.

149 **2.2 Preparation of composite membrane**

150 PVAm was added to deionized water and stirred for 3 hours to get 3 wt. % solution of polyvinyl  
151 amine. The mixture was left for rolling on mechanical roller overnight to obtain a clear solution.  
152 This procedure was carried out at room temperature. Afterwards, CNCs were added to the clear  
153 solution [38]. Different concentrations of CNCs with respect to weight percent of polymer i.e.  
154 0.5%, 1% and 1.5%, were added to get casting solutions as shown in Table 1. The solutions were  
155 mechanically stirred overnight and sonicated afterwards for 30 min. The air bubbles should be  
156 removed from resultant suspension to cast defect free membranes. Therefore, solution was left at  
157 room temperature for 2 hours. .Using a dip-coating technique, a selective and dense membrane  
158 was casted on PSf sheet as shown in Figure 1. The membranes were placed in fume hood overnight  
159 at room temperature for drying. [10].

160

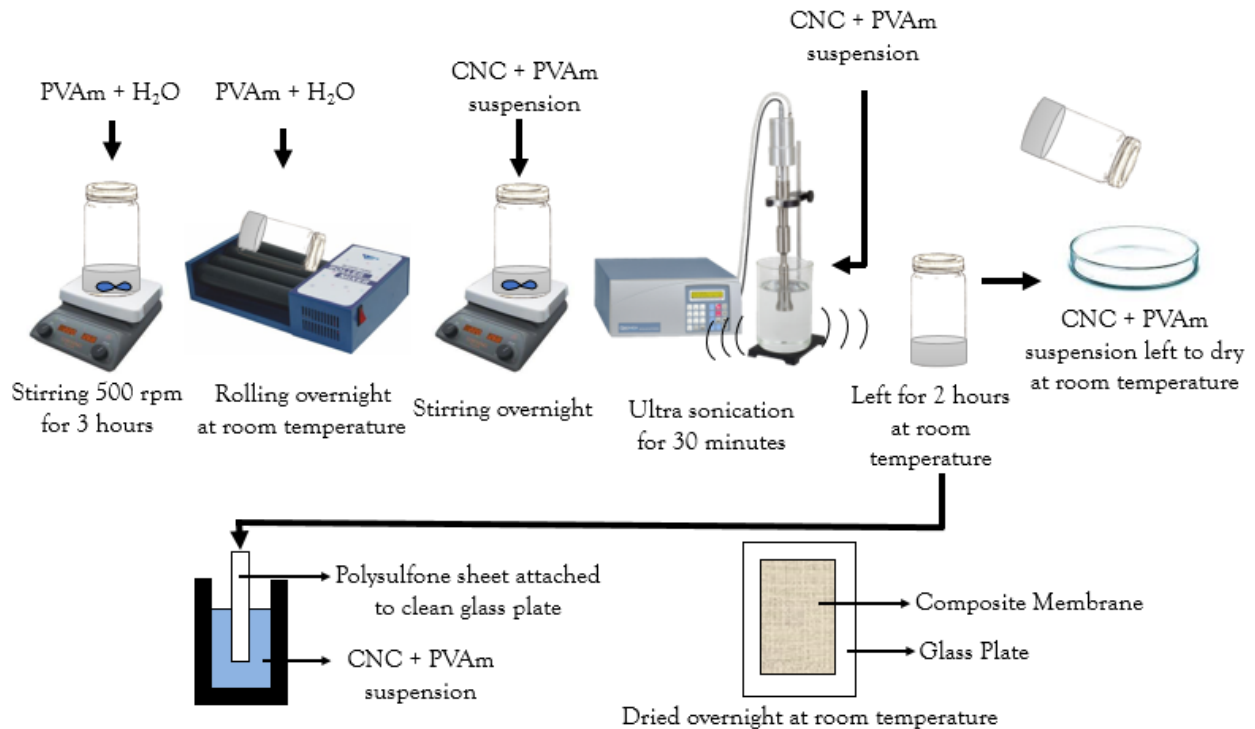
161 **Table.1:** Composition of casted membranes with their codes.

Membrane code	PVAm wt. %	Wt./Wt. CNC
Pure PVAm	3 g	-
PVAm/0.5CNC	3 g	0.5 %
PVAm/1CNC	3 g	1 %
PVAm/1.5CNC	3 g	1.5 %

162

163





164

165

**Figure1.** Cellulose nano-crystals and polyvinyl amine nanocomposite membranes.

166

### 2.3 Scanning electron microscopy

167

The morphology of nanocomposite membranes was investigated by using Scanning Electron

168

Microscopy (S-4700 Hitachi, Japan). The gold sputtering was carried out on membranes by ion

169

sputtering machine model JFC-1500 JEOL Limited. In order to find the thickness of selective layer

170

of membrane over polysulfone sheet, cross sectional view of membranes were also examined.

171

Liquid nitrogen was used to break membrane for the cross-sectional images.

172

### 2.4 X-ray diffraction

173

The crystallinity of nanocomposite membranes was determined by using STOE X-ray

174

Diffraction. The scan angle was set to 20-60° at a step size of 0.4° and a step time of 0.5

175

sec/step. The radiation energy used for x-ray diffraction was Cu K  $\alpha$ -1 frequency of 1.5406 Å. The

176 method mentioned in literature was used to determine the crystallinity index of casted membranes  
177 [39]. The area under the curve of the XRD spectrum between 20-60° was considered as total area.

## 178 **2.5 Percentage moisture uptake**

179 The degree of swelling of PVAm/CNC membranes was investigated at room temperature by  
180 subjecting them to a humid environment. The membranes were placed in a closed chamber with  
181 relative humidity (RH) of 0%. The degree of swelling of membranes was measured by subjecting  
182 them to environment with 87% relative humidity. The membranes were placed in such a way that  
183 there was no direct contact of membranes with the saturated salt solution. The degree of swelling  
184 was calculated after every 24 hours on basis of the increase in weight of membranes. The readings  
185 were taken for 10 days. Day 1 measurement was taken after 24 hours in which membranes were  
186 subjected from 0% RH to 87% RH. Equation (1) was used to calculate the degree of swelling [14].

$$187 \quad DoS (\%) = \frac{S-D}{D} \times 100 \quad (1)$$

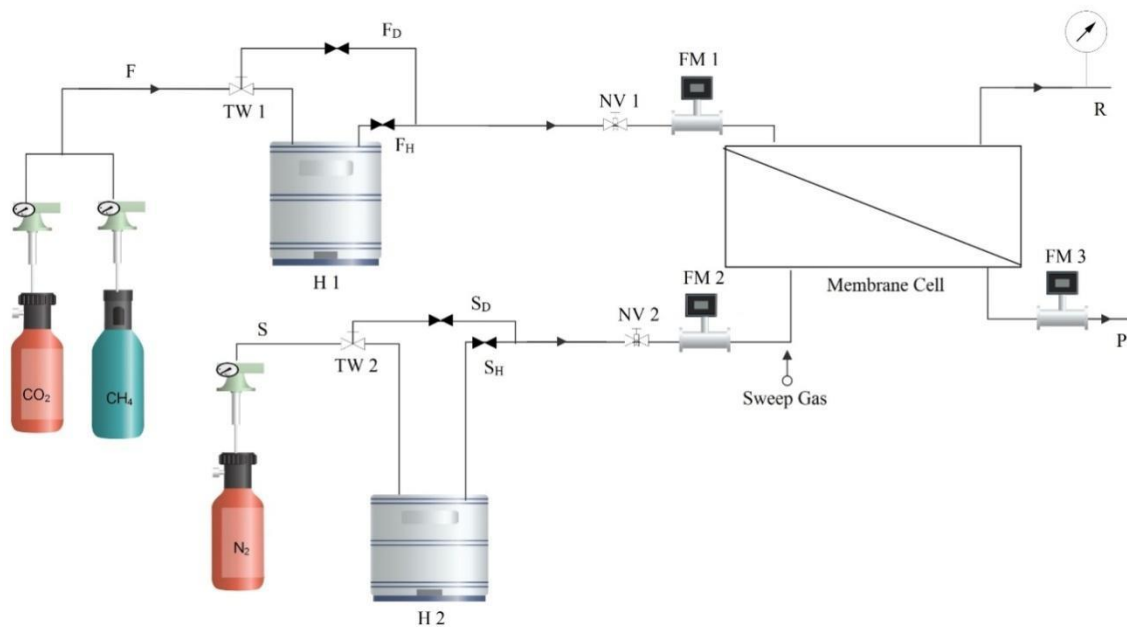
188 Where, D is the mass of dry membrane and S is the mass of swelled membranes.

## 189 **2.6 Permeation test**

190 Permeation testing was conducted to examine the performance of membrane for CO<sub>2</sub> capture. A  
191 membrane rig as shown in Figure 2 was used to conduct permeation testing. The rig has ability to  
192 test the membranes at moderately high pressures; up to 20 bars at both humid and dry conditions.  
193 Figure 2 shows the flow diagram of membrane rig.

194

195



196

197 **Figure 2.** Flow diagram of gas permeation rig, F = Feed Line, S = Sweep Line, TW = Two Way  
 198 Valve, F<sub>D</sub> = Dry Feed, F<sub>H</sub> = Humidified Feed, S<sub>D</sub> = Dry Sweep, S<sub>H</sub> = Humidified Sweep, H =  
 199 Humidifier, NV = Needle Valve, FM = Flow Meter, P = Permeate, R = Retentate.

200

201 CO<sub>2</sub> or CH<sub>4</sub> are filled in cylinders which are allowed to flow in feed line, F, controlled by the  
 202 pressure regulators attached with cylinders. After leaving the cylinder, gas moves towards a two-  
 203 way valve which provides an option for the gas to be used in dry conditions, F<sub>D</sub>, or in humidified  
 204 conditions, F<sub>H</sub>. For dry gas, humidifier, H1, is bypassed. Needle valve NV1 is present, which  
 205 allows precise control of gas flow. Flow meter, FM1, indicates the flow rate of feed gas before  
 206 entering to membrane cell. The gas that pass through the membrane, exits the cell from bottom as  
 207 permeate, P. While that do not pass the membrane, exits from top as retentate, R. Flow rate of  
 208 permeate is measured manually by bubble flow meter indicated here as FM3.

209

210 N<sub>2</sub> is used as the sweep gas. The pressure is controlled and read using pressure regulator and gauge,  
211 respectively. Same phenomenon as above takes place in sweep line. Two-way valve, TW2, allows  
212 either sweep be humidified, S<sub>H</sub>, or dry, S<sub>D</sub>. Volume of humidifier is the same as for above. Needle  
213 valve, NV2, and flow meter, FM2, controls and measures the flow rate of sweep gas, respectively.  
214 However, sweep enters the membrane cell from bottom and exits the permeate line P. The pressure  
215 and flow of sweep gas are kept very low. Different membrane compositions were tested for their  
216 CO<sub>2</sub> and CH<sub>4</sub> permeability and selectivity. Sweep gas was humidified and then supplied to  
217 membranes so that maximum moisture uptake was achieved as indicated by the degree of swelling  
218 of membranes. The single gas testing for CO<sub>2</sub> and CH<sub>4</sub> were carried out by supplying the humid  
219 gases to the membranes. Effect of pressure was investigated by supplying feed gases at 5, 10 and  
220 15 bar. While, keeping flow rate and temperature constant. The permeability of the gases was  
221 measured manually by using a bubble flow meter.

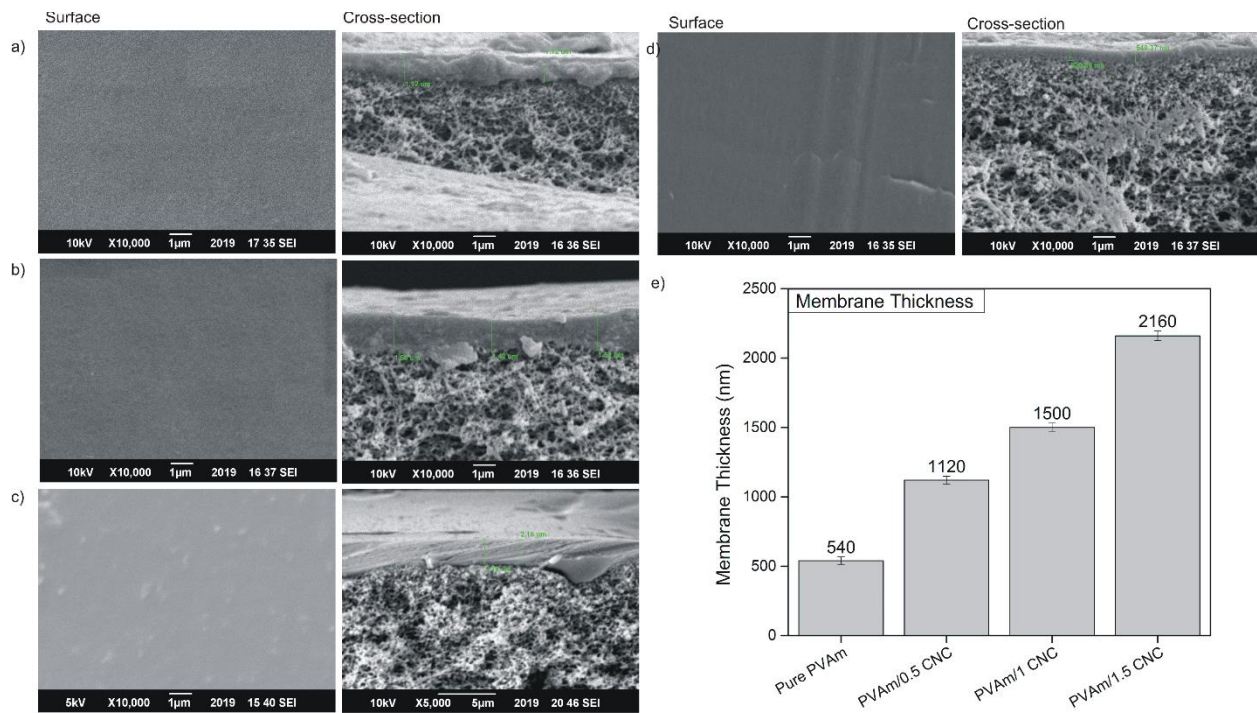
## 222 **3 Results and discussion**

### 223 **3.1 Characterization**

#### 224 **3.1.1 Morphology and thickness of composite membranes**

225 The scanning electron microscopy (SEM) analysis was performed to investigate the morphology  
226 and thickness of selective layer. SEM results revealed the smooth and defect free surface of  
227 PVAm/1CNC membrane as shown in Figure 3(a). Furthermore, no agglomeration of nano particles  
228 was observed on the membrane surface. This indicates the even dispersion of nano particles within  
229 the polymeric matrix. Moreover, no cracks are visible on the membrane surface. Hence, the surface  
230 morphology does not show any adverse effect with addition of CNC. The cross sectional view of  
231 PVAm/1CNC composite membrane are represented in Figure 3(b). Web like structure refers to the

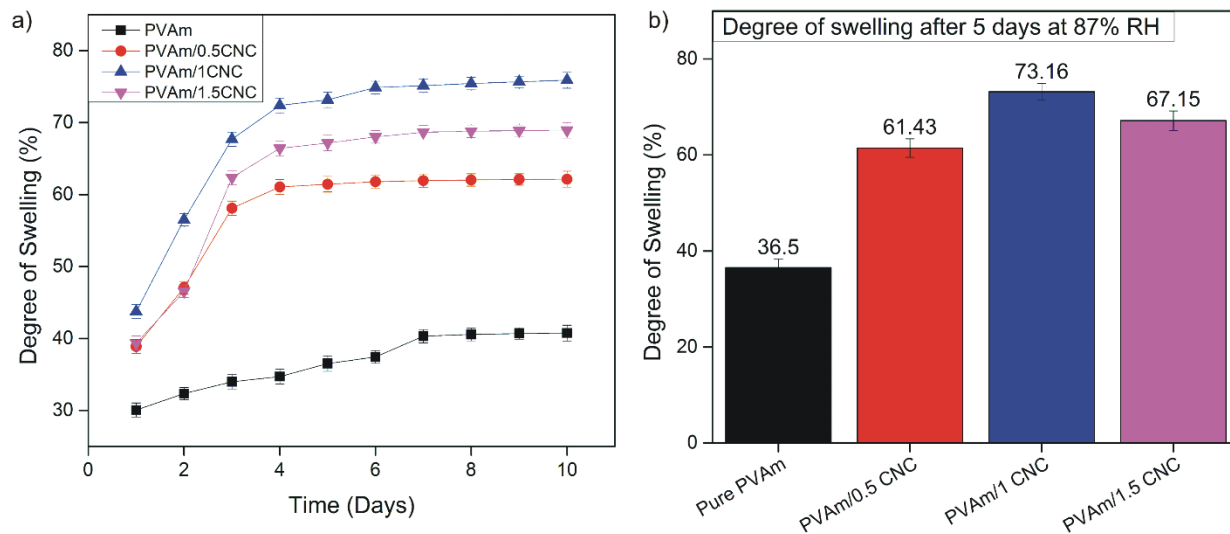
232 micro-porous PSf support with selective PVAm/CNC membrane on the top. Membrane thickness  
 233 was determined by taking measurements at different parts of the composite membrane. The  
 234 average thickness observed for PVAm/1CNC membrane was approximately  $1.5 \pm 0.12 \mu\text{m}$ .  
 235 Furthermore, the selective layer thickness enhanced with increase in the concentration of CNC in  
 236 the polymeric membrane as shown in figure 3(c). This can be ascribed to the increase in viscosity  
 237 of the casting solution with increasing the CNC concentration. The addition of 1.5wt % CNC  
 238 increases the average thickness of membrane three times of the thickness of pure PVAm  
 239 membrane. However, 1wt% CNC membrane shows optimized results with thickness of 1500 nm.  
 240 Hence, an increment in membrane thickness is observed for a successive increase in CNC  
 241 concentration [40].



242  
 243 **Figure 3.** SEM images of PVAm/CNC membranes with addition of 1 wt% CNC, (a) surface and  
 244 cross-section morphology of PVAm/0.5CNC, (b) surface and cross-section morphology of  
 245 PVAm/1CNC, (c) surface and cross-section morphology of PVAm/1.5CNC, (d) surface and cross-  
 246 section morphology of pure PVAm membrane, and (e) thickness of nanocomposite membranes.  
 247

### 248 **3.1.2 Swelling behavior**

249 Degree of swelling of composite membranes was investigated by subjecting them to a RH of 87%.  
250 Figure4 (a) shows the maximum moisture uptake over a period of 10 days. The maximum swelling  
251 was observed until day 5. However, the membrane with 1% CNC concentration absorbed the  
252 moisture until day 6. Furthermore, a gradual increase in the degree of swelling was observed for  
253 pure PVAm membrane between day 6 and 7. The highest degree of swelling was observed for  
254 membrane with 1% CNC concentration i.e.  $75.88 \pm 1.1\%$ . However, the membrane with 1.5% CNC  
255 contents showed a swelling percentage of  $68.93 \pm 0.9\%$ . The strong reinforcement capability of  
256 CNC molecules result in reduced degree of swelling when CNC concentration is increased. This  
257 effect resists the mechanical restraining and rejects the further uptake of water molecules resulting  
258 in a reduced degree of swelling of PVAm/CNC composite membranes [18]. Increasing CNC  
259 concentration imparts strong hydrogen bonding which affixes the polymeric chains thus reducing  
260 the moisture uptake capability of membranes [41]. Moreover, reduction in swelling degree can  
261 also be correlated with the rise in the crystallinity index of membranes with the increasing CNC  
262 concentrations as shown in figure 4(b).



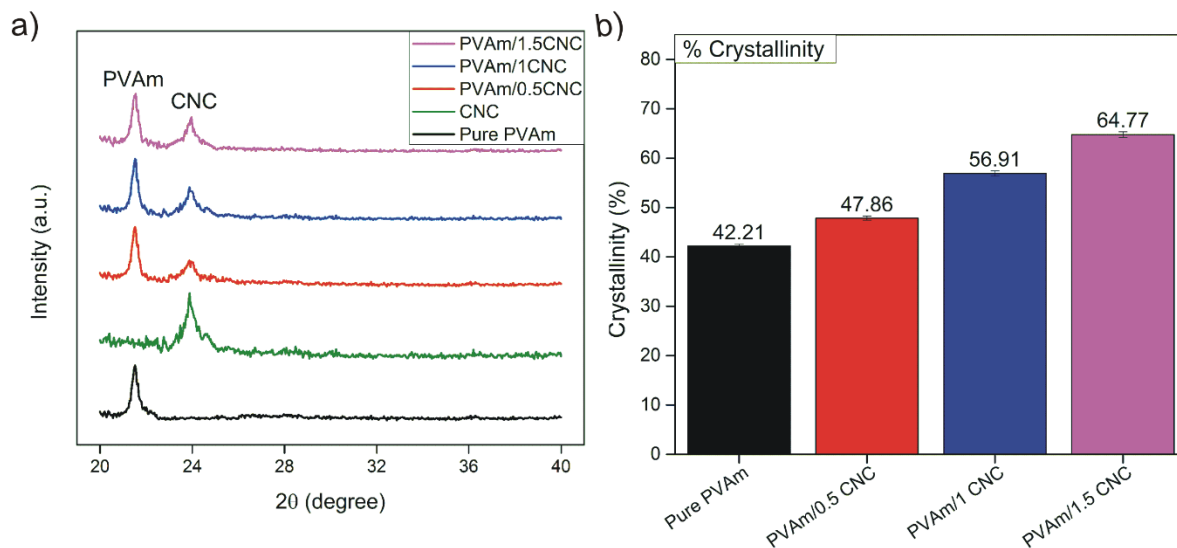
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264 **Figure 4.** Degree of swelling for ten days (a) degree of swelling of PVAm/CNC membranes  
 265 compared with pure PVAm membrane at 87% RH (b) maximum degree of swelling after 6 days  
 266 at 87% RH.

267

### 268 3.1.3 Crystallinity of composite membranes

269 X-ray diffractogram for pure PVAm and PVAm/CNC composite membranes is shown in figure  
 270 5(a). The crystallinity of all composite membranes was calculated as mentioned in section 2.4. At  
 271  $2\theta = 21.52^\circ$ , a sharp peak appeared in all diffractogram which indicated presence of PVAm. The  
 272 approximate crystallinity of the pure PVAm membrane was calculated around 42%. No peak shift  
 273 was observed when CNC was added to PVAm membrane. CNC showed its presence on  
 274 diffractogram in form of a sharp peak at  $2\theta = 23.88^\circ$  [38]. This showed that the addition of CNC  
 275 increased the crystallinity of composite membranes. Figure 5(b) showed the effect of CNC  
 276 concentration on crystallinity. It was observed that increasing the CNC concentration enhanced  
 277 the crystallinity of composite membranes. Due to highly crystalline nature of CNC, its peak was  
 278 visible even at low concentrations [42].



279

280 **Figure 5.** XRD analysis of composite membranes, a) pure PVAm and PVAm/CNC membranes,  
 281 b) Effect of CNC concentration on % age crystallinity.

282

283 PVAm/0.5CNC showed an increased crystallinity as compared to pure PVAm membrane.

284 Maximum crystallinity was attained with addition of 1.5 wt. % CNC in nanocomposite membranes

285 i.e. 64.77 %. Increasing crystallinity imparts rigidity in polymer and inhibits chain mobility.

## 286 3.2 Permeation results

### 287 3.2.1 CO<sub>2</sub> and CH<sub>4</sub> permeance and selectivity in effect with CNC concentration

288 Figure 6 shows the effect of increasing CNC concentration on membrane performance in terms of

289 permeance and selectivity for CO<sub>2</sub> and CH<sub>4</sub>. Addition of CNC had a positive effect on membrane

290 performance for CO<sub>2</sub> separation. It has been observed that permeance and selectivity of CO<sub>2</sub>

291 enhanced with increasing concentration of CNC in PVAm up to the addition of 1wt% of CNC.

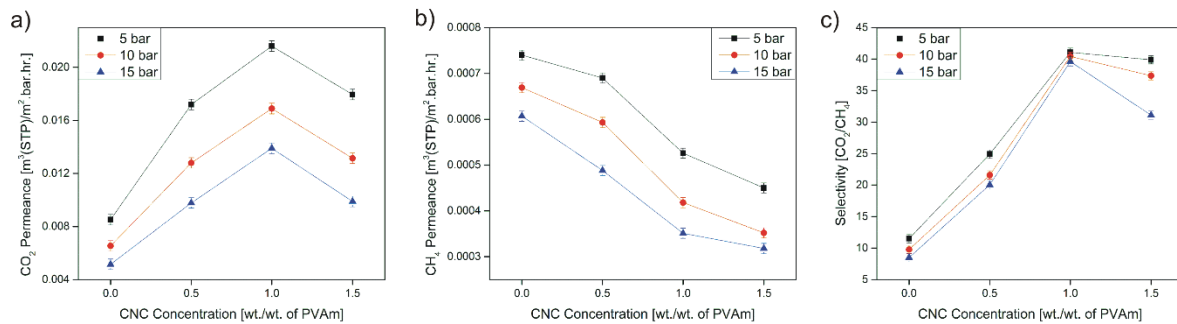
292 From the results as shown in Figure 6 a & c, it has been observed that the pure PVAm membrane

293 showed permeance of 0.00852 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr and selectivity of CO<sub>2</sub>/CH<sub>4</sub> around 12 at a

294 pressure of 5 bar. Furthermore, the addition of CNC shows enhanced CO<sub>2</sub> permeance and



295 selectivity for all compositions as compared to pure PVAm. The highest results for CO<sub>2</sub> permeance  
 296 and CO<sub>2</sub>/CH<sub>4</sub> selectivity were obtained by the addition of 1 wt % CNC i.e.0.0216  
 297 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr. and 41, respectively at 5 bar. However, the decreasing trend has been observed  
 298 for permeance of CH<sub>4</sub> with increasing CNC concentration. For PVAm/1CNC composite  
 299 membrane, CH<sub>4</sub> permeance was observed to be 0.000526 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr.at 5 bar.



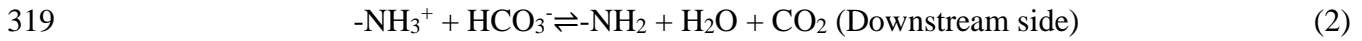
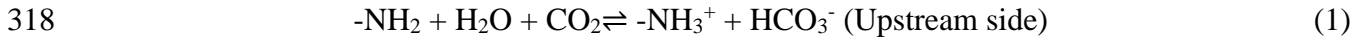
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301 **Figure 6.** Effect of increasing CNC concentration on permeation performance of PVAm/CNC  
 302 membranes (a) CO<sub>2</sub> permeance for pure PVAm and PVAm/CNC membranes (b) CH<sub>4</sub> permeance  
 303 for pure PVAm and PVAm/CNC membranes and (c) Selectivity of CO<sub>2</sub>/CH<sub>4</sub> for pure PVAm and  
 304 PVAm/CNC membranes.

305

306 In addition to membrane structure, the properties of gas pair to be separated is also very important  
 307 for the performance of membrane. In case of novel composite membranes casted in this work, the  
 308 separation layer includes polymer phase, dispersed nano filler phase and interaction of these two.  
 309 In PVAm/CNC composite membranes, the CH<sub>4</sub> is transported only by solution diffusion  
 310 mechanism. Whereas, in addition to solution diffusion mechanism CO<sub>2</sub> is dominantly transported  
 311 by facilitated transport mechanism. Addition of CNC in PVAm matrix enhanced the moisture  
 312 uptake capability of composite membrane. That results in the increased rate of facilitated transport  
 313 of CO<sub>2</sub>. CO<sub>2</sub> is a water soluble gas; therefore, the presence of high water contents increased CO<sub>2</sub>  
 314 transport in form of bicarbonate ions. Furthermore, the presence of amine groups on backbone of  
 315 polymeric chain act as fixed carries and selectively transport CO<sub>2</sub> through the membrane. The

316 chemical equation for the reactions occurring in membrane is defined by the following reactions  
317 [10]:



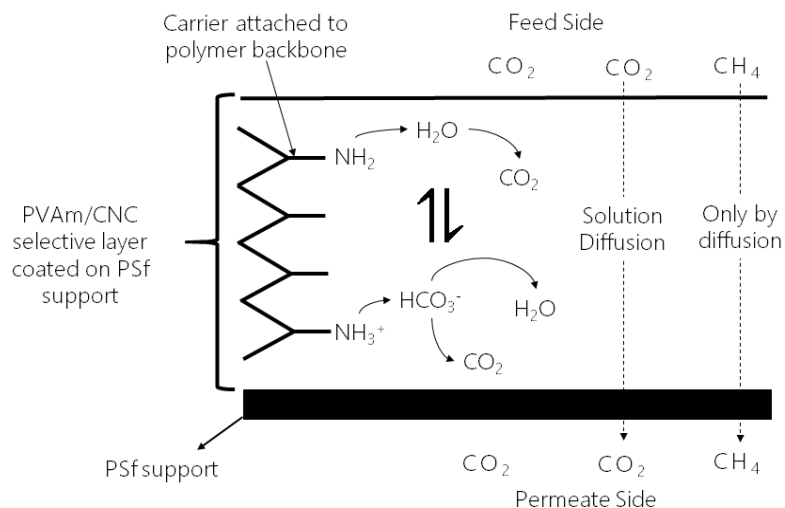
320

321 The addition of CNC in PVAm provides non selective and less resistant pathways for gas  
322 transportation. The moisture uptake induces swelling behavior which increases free volume in  
323 between the polymeric chains and also increases the chain flexibility. This increases the diffusive  
324 transport of gases across the membrane. As the kinetic diameter of CO<sub>2</sub> and CH<sub>4</sub> are 3.3 Å and 3.8  
325 Å, respectively. So, both gases can selectively pass through swelled areas [43]. Even though, the  
326 permeance of CO<sub>2</sub> is much higher as compared to CH<sub>4</sub>. This can be attributed to the high solubility  
327 of CO<sub>2</sub> in water.

328

329 Furthermore, it has been observed that the value of permeance for both CO<sub>2</sub> and CH<sub>4</sub> has reduced  
330 after addition of CNC above 1 wt%. The decline in membrane performance above 1wt% CNC can  
331 be attributed towards the reduced moisture content and increased crystallinity. Increased  
332 crystallinity not only induces rigidity in polymer matrix but also reduced moisture uptake ability  
333 of composite membranes as explained in section 3.1.3. Therefore, transportation of gas by solution  
334 diffusion mechanism has declined. Furthermore, as explained in section 3.3.2 that increasing CNC  
335 concentration in PVAm significantly enhanced the selective layer thickness beyond addition of  
336 1wt. % CNC. Hence, it has also contributed to reduce gas permeation through the membrane. The  
337 CH<sub>4</sub> permeates only by solution diffusion mechanism due to its non-reactive nature. Hence, its  
338 permeance showed sharper decline with increase in chain rigidity. However, the decrease in

339 moisture uptake ability dominantly affects the transport of CO<sub>2</sub> as it is a water soluble gas and  
 340 mainly transported through facilitated transport [21].  
 341  
 342 The transportation mechanism for both the gases is represented in Figure.7. It can be seen that CO<sub>2</sub>  
 343 is transported by i) reaction with amine carriers present on backbone of polymer matrix. However,  
 344 due to limited carrier concentration, there will be no further increase in rate of CO<sub>2</sub> transport once  
 345 a saturation state occurs. This is intrinsic property of PVAm membranes. Addition of CNC had no  
 346 effect on this property. ii) Transportation of CO<sub>2</sub> as bicarbonate ions aids by the presence of  
 347 moisture contents. Addition of CNC plays significant role in this transport as it enhanced the  
 348 moisture uptake ability of membrane, and iii) Transport of CO<sub>2</sub> by molecular diffusion. Whereas,  
 349 the only transport mechanism is molecular diffusion for CH<sub>4</sub>. Therefore, addition of 1wt% CNC  
 350 showed positive effects on CO<sub>2</sub> transport but does not enhance the membrane performance for  
 351 CH<sub>4</sub> transport.



352

353 **Figure 7.** Mechanism of gas transport through PVAm/CNC nanocomposite membranes.

354

### 355 **3.2.2 Effect of feed pressure on gas permeance of composite membranes**

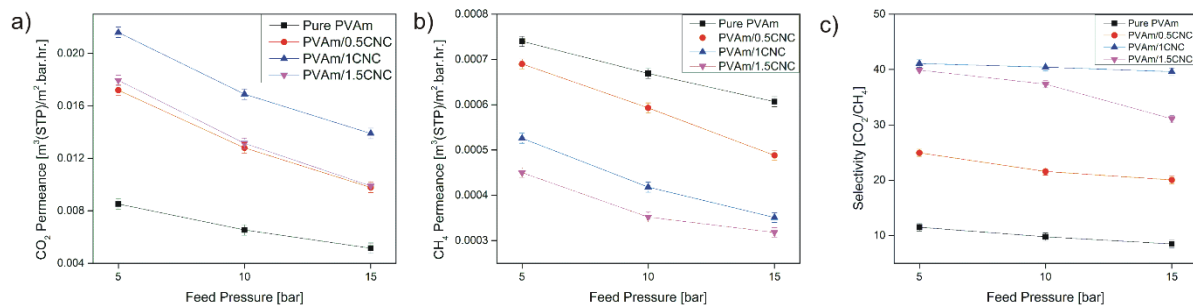
356 The pure PVAm and PVAm/CNC composite membranes were tested at a pressure of 5, 10 and 15  
357 bar for CO<sub>2</sub> and CH<sub>4</sub> permeance. It was observed that permeance of both gases and selectivity of  
358 CO<sub>2</sub>/CH<sub>4</sub> decreased as a result of increasing pressure for all the formulated membranes. The  
359 highest CO<sub>2</sub> permeance and CO<sub>2</sub>/CH<sub>4</sub>selectivity was achieved by PVAm/1CNC membrane at 5  
360 bar pressure i.e. 0.0216 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr. and 41, respectively as shown in Figure 8 (a & c).  
361 However, addition of 1wt. % CNC showed decline in permeance ofCH<sub>4</sub>using pure PVAm  
362 membrane at 5 bar. The values of CH<sub>4</sub> permeance for PVAm and PVAm/1CNC membranes are  
363 0.00074 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr.and 0.000526 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr., respectively. Furthermore, the drop  
364 in permeance for CO<sub>2</sub> is rapid in the beginning with increasing pressure. This behavior is attributed  
365 to the saturation state of all amine carriers present on polymer matrix. Once the carriers become  
366 saturated in pure PVAm membrane, CO<sub>2</sub> is transported only by solution diffusion mechanism.  
367 However, the dominating phenomenon is still facilitated transport in PVAm/CNC membranes with  
368 aid of moisture content present in membranes. Hence, the values of CO<sub>2</sub> permeance and selectivity  
369 of all CNC containing membranes are higher than pure PVAm membranes at all pressure.

370

371 Performance of water swollen membranes was high when the feed gas was fully humidified at low  
372 pressure. Increased feed pressure enhances the flow of gas. Further high pressure squeezes the  
373 membrane and increase the flow of moisture content through the membrane. Hence, moisture  
374 contents in the membrane get decreased even at high %RH [17]. Reduction of moisture contents,  
375 decrease the facilitated transport of CO<sub>2</sub> across the membrane [12]. Furthermore, at the high feed  
376 pressure plasticization occurs in polymeric membranes that reduces the chain mobility and impart

377 rigidity in membrane structure. This further reduced membrane performance at high pressures  
 378 [18]. In addition, the high feed pressures introduce compression on polymer matrix that reduces  
 379 the free voids available for solution diffusion of gases. Hence, results in the decline of permeance  
 380 of both CO<sub>2</sub> and CH<sub>4</sub>. Thus, the decrease in CO<sub>2</sub>/CH<sub>4</sub> selectivity was not very prominent except  
 381 for PVAm/1.5CNC membrane. The selectivity dropped from 40 to 31 when feed pressure was  
 382 increased from 5 to 15 bar as shown in Figure 8(c). PVAm/1CNC showed a very minute selectivity  
 383 drop i.e. 41 to 39, with increase in pressure from 5 to 15 bar.

384



385

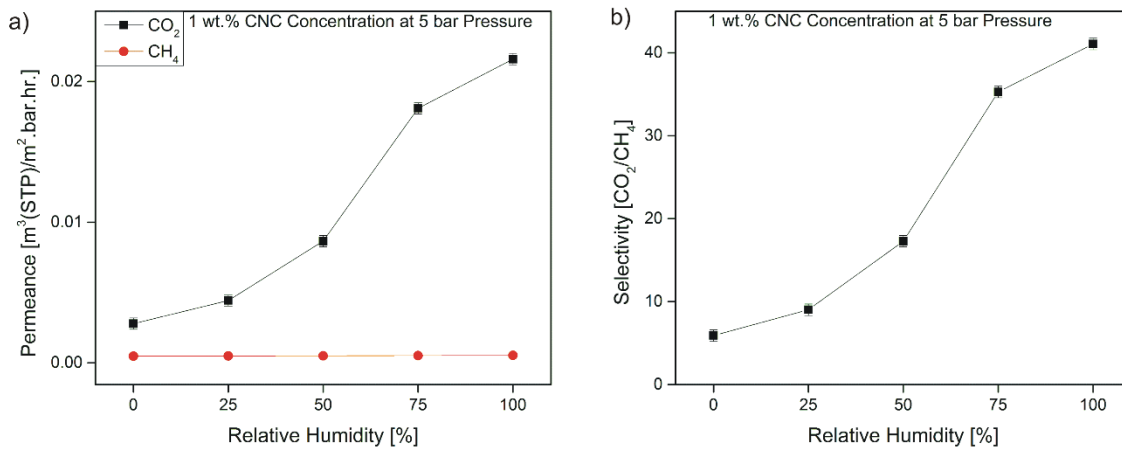
386 **Figure 8.** Effect of feed pressure on (a) CO<sub>2</sub> permeance (b) CH<sub>4</sub>permeance (c) Selectivity  
 387 CO<sub>2</sub>/CH<sub>4</sub>.

388

### 389 3.2.3 Effect of relative humidity on facilitated transport of CO<sub>2</sub>

390 PVAm/1CNC membranes were investigated for the effect of RH on transport of CO<sub>2</sub> and CH<sub>4</sub> at 5  
 391 bar pressure as shown in figure 9. The effect of %RH was investigated by subjecting the  
 392 membranes to different humidity levels of 0, 25, 50, 75 and 100% before testing. The different  
 393 humidity levels were achieved by data obtained from the degree of swelling graphs. Results  
 394 showed that increasing %RH improved the CO<sub>2</sub> permeance and selectivity. This is due to the fact  
 395 that higher moisture contents increases facilitated transport of CO<sub>2</sub> across the membrane [40].  
 396 Furthermore, an increase in %RH enhances the degree of swelling. Hence, open the amorphous

397 pores in polymer matrix providing non selective bypass for diffusion of CO<sub>2</sub> and CH<sub>4</sub> molecules.  
 398 However, CO<sub>2</sub> moves across the membrane by facilitated and diffusive transport but CH<sub>4</sub> only  
 399 passes through by diffusive transport. Hence, the increase in CH<sub>4</sub> permeance is very low with a  
 400 rise in %RH as compare to CO<sub>2</sub> permeance [44]. At 0% RH, CO<sub>2</sub> and CH<sub>4</sub> showed a permeance  
 401 of 0.00279 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr. and 0.000472 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr., respectively. Increasing the  
 402 %RH to 100%, CO<sub>2</sub> permeance enhanced approximately 10 folds. While, CH<sub>4</sub> permeance just  
 403 increased to 0.000526 m<sup>3</sup>(STP)/m<sup>2</sup>.bar.hr. The rise in the %RH from 0 to 100%, increased the  
 404 selectivity of membrane from 6 to 41, respectively. High selectivity rise can be attributed to the  
 405 increase in permeance of CO<sub>2</sub> with rising %RH.



406

407 **Figure 9.** Percentage RH effect on facilitated transport (a) CO<sub>2</sub> and CH<sub>4</sub> permeance (b) CO<sub>2</sub>/CH<sub>4</sub>  
 408 selectivity.

## 409 4 Conclusions

410 PVAm/CNC membranes were successfully fabricated and tested for the effect of CNC  
 411 concentration and feed pressure on separation performance of CO<sub>2</sub> and CH<sub>4</sub>. Addition of CNC has  
 412 improved the permeance and selectivity of PVAm membrane for CO<sub>2</sub> transport. However, addition  
 413 of CNC does not showed any significant change on CH<sub>4</sub> permeance. As compared to pure PVAm

414 membrane, enhanced separation performance was observed when CNC was added in membranes.  
415 Furthermore, addition of CNC enhanced the moisture uptake of formulated membranes. The  
416 moisture uptake was increased up to the addition of 1 wt.% CNC and started decreasing beyond  
417 this concentration. PVAm/1CNC showed a swelling degree of  $75.88 \pm 1.1\%$ . However, the  
418 thickness of selective layer and crystallinity of PVAm/CNC membranes showed increasing trend  
419 with increasing CNC concentration. In addition, it was found that by increasing feed pressure the  
420 permeance of both gases for the casted membranes declined. However, this decline was more  
421 prominent for CH<sub>4</sub> gas. Furthermore, it is highly recommended to work at maximum swollen  
422 conditions to get maximum benefit of facilitated transport of CO<sub>2</sub>. It was seen that PVAm/1CNC  
423 membrane showed the highest CO<sub>2</sub> permeance and selectivity of  $0.0216 \text{ m}^3(\text{STP})/\text{m}^2 \cdot \text{bar} \cdot \text{hr.}$  and  
424 41, respectively at a pressure of 5 bar. According to the results, 1 wt.% CNC was optimized  
425 concentration and at 5 bar pressure, membrane showed best results.

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