

1 A Thiadiazole-Based Covalent Organic Framework: A Metal-Free 2 Electrocatalyst toward Oxygen Evolution Reaction

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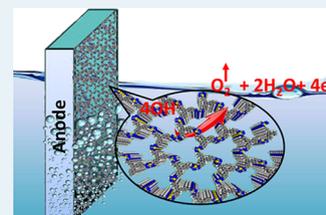
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Supporting Information

5 **ABSTRACT:** Covalent organic frameworks (COFs) have attracted surging interest lately due to
6 their wide potential in several frontline application areas like gas storage, sensing, photovoltaics,
7 fuel cells, active catalyst supports, and so on. However, only very few reports are available for the
8 metal-free electrocatalysis over COFs. Herein, we developed a new thiadiazole-based COF, C4-
9 SHz COF, through the reaction between 1,3,5-tris(4-formylphenyl)benzene and 2,5-dihydrazinyl-
10 1,3,4-thiadiazole that possesses a very high specific surface area of 1224 m² g⁻¹, unique molecular
11 architecture, high porosity, and abundant active sites. The as-synthesized C4-SHz COF displayed
12 superior electrocatalytic oxygen evolution reaction (OER) activity and excellent long-term
13 durability. The electrocatalytic performance of the C4-SHz COF achieved a current density of 10 mA/cm² at an overpotential of 320
14 mV. The higher activity of the C4-SHz COF could be attributed to the high Brunauer–Emmett–Teller surface area, porosity, and
15 network structure of the π -conjugated organic building blocks, which allowed fast charge and mass transport processes. This work
16 validates the promising potential of a metal-free COF electrocatalyst toward the OER and its capability to replace carbon-based
17 electrocatalysts.

18 **KEYWORDS:** covalent organic frameworks (COFs), thiadiazole moiety, metal-free electrocatalyst, oxygen evolution reaction (OER),
19 water splitting



20 ■ INTRODUCTION

21 Rapid decay of fossil fuels combined with environmental crises
22 associated with carbon emissions significantly affected the
23 economy and ecology of the whole world. Thus, the demand
24 for renewable energy is surging over the years, and electro-
25 chemical water splitting can be considered as a green and
26 promising technology¹ to overcome this problem. The
27 electrochemical water splitting plays a crucial role in advanced
28 energy technologies, such as solar fuel production, super-
29 capacitor, metal–air batteries, etc.^{2–5} Still, water oxidation
30 (oxygen evolution reaction, OER) requires a higher energy
31 input due to the involvement of the multi-electron transfer
32 reaction pathway and its low efficiency. Hence, the
33 commercialization of water splitting technology requires stable,
34 highly active, and low-cost water oxidation electrocatalysts.
35 Currently, precious transition metal oxide-based electro-
36 catalysts such as RuO₂ and IrO₂ are highly active toward the
37 OER from alkaline water. However, high cost, scarcity, and low
38 stability hamper the utilization of clean and sustainable energy
39 technologies^{6,7} in this context. Thus, for the past couple of
40 years, extensive efforts have been devoted for the development
41 of earth-abundant, low-cost, and efficient transition metal-
42 based electrocatalysts such as oxides, chalcogenides, phos-
43 phides, nitride, metal-free catalysts, etc., for the OER.^{8–14}
44 However, most of the transition metal-based catalysts still
45 suffer from lower catalytic activity, inferior conductivity, and
46 poor operational durability.

Simultaneously, over the past few years, intensive research
47 has been carried out for developing efficient metal-free
48 catalysts.^{2,3} Metal-free electrocatalysts have several unique
49 advantages, such as their environment-friendly nature, earth-
50 abundant, cost-effective, and resistance to a wide pH range.
51 For electrochemical OER applications, several metal-free
52 catalysts have been developed in recent times.^{15–17} Among
53 different metal-free porous nanomaterials, covalent organic
54 frameworks (COFs)^{18,19} are the emerging class of porous
55 organic polymers with precisely controllable structural motifs
56 linked through covalent bonds. COFs possess extraordinary
57 properties like large surface area, high crystallinity, tunable
58 pore size, and unique molecular architecture.^{20–24} Due to these
59 unique properties, COFs have been used in a wide range of
60 applications.^{25–27} High surface area, tunable structures with
61 appropriate building blocks, and porous nature of the COFs
62 made them ideal for electrocatalysis.²⁸ Accordingly, substantial
63 progress has been made for the design and synthesis of COFs
64 from low-cost monomer precursors. Till now, COFs have been
65 explored as a support material of graphene, conductive carbon, 66

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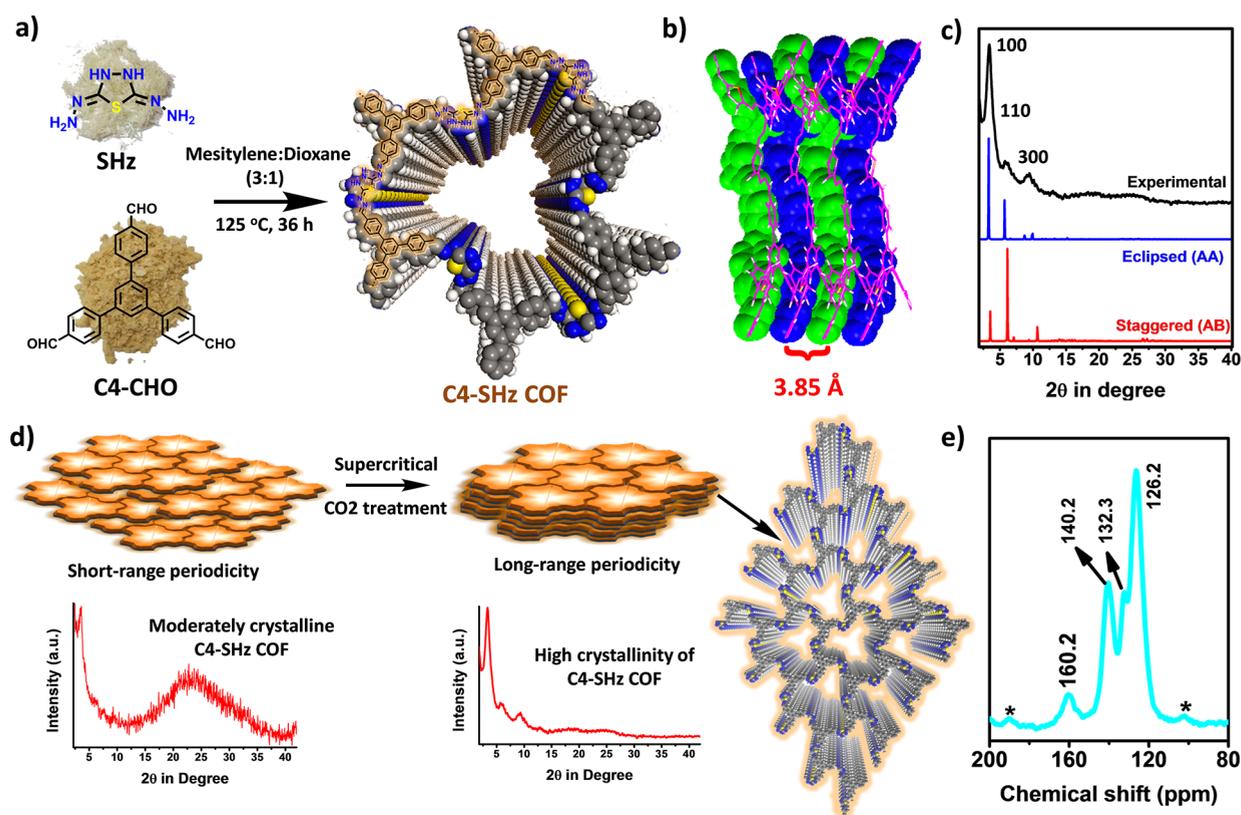


Figure 1. Schematic presentation of the COF (a) proposed structure and (b) packing diagram. (c) Comparison of experimental PXRD pattern of the crystalline C4-SHz COF with simulated PXRD patterns of the C4-SHz COF. (d) Graphical representation of switching shortly to long-range periodicity. (e) Solid-state ¹³C MAS NMR spectrum of the COF; the spinning sidebands are represented by asterisks.

67 carbon nanotubes (CNTs), etc., for electrocatalytic applica-
 68 tions. For example, Kamiya et al. have reported covalent
 69 triazine frameworks as support materials for Pt nanoparticle
 70 and studied its ORR activity.²⁹ Mullangi et al. used flexible
 71 COFs as support materials for electrocatalytic OER
 72 applications.²⁸ Similarly, Aiyappa et al. designed Co-COF
 73 and studied its OER activity,³⁰ whereas Fe and Co NPs
 74 supported over COFs have been employed in electrochemical
 75 oxygen reduction reaction (ORR).³¹ Although there are very
 76 few reports on the usage of COFs as photocathodes for light-
 77 induced hydrogen evolution reaction (HER) via water
 78 splitting,³² the use of a crystalline COF as a metal-free OER
 79 electrocatalyst has been rarely explored till date.³³

80 In this article, we have introduced an unprecedented imine-
 81 linked thiadiazole-based crystalline COF (C4-SHz COF) and
 82 explored its catalytic activity in the electrochemical water
 83 oxidation reaction. Through the general solvothermal protocol
 84 (Schiff base condensation using acetic acid),^{26,27,30} a nitrogen-
 85 rich thiadiazole moiety has been incorporated in the polymeric
 86 backbone of the C4-SHz COF, which exhibited superior
 87 activity toward OER. The relevant characterization data
 88 suggested that the C4-SHz COF possesses a high specific
 89 surface area and crystalline metal-free organic framework
 90 structure. The newly developed COF displayed an excellent
 91 electrocatalytic activity and durability under alkaline pH
 92 conditions, which is comparable to the other metal-free
 93 electrocatalysts reported so far.

RESULTS AND DISCUSSION

94

The thiadiazole-based imine-linked COF (C4-SHz COF) was
 95 synthesized through the general Schiff base condensation
 96 polymerization between 1,3,5-tris(4-formylphenyl)benzene
 97 (C4-CHO) and 2,5-dihydrazinyl-1,3,4-thiadiazole (SHz)
 98 under the solvothermal condition for 36 h (Figure 1a and
 99 Figures S1–S3, Supporting Information). The resultant as-
 100 synthesized imine-linked framework has shown a low sign of
 101 crystallinity (Figure 1d), and this could often occur due to the
 102 self-assembly through weak π - π stacking interaction of
 103 individual COF layers. To get the well-defined molecular
 104 stacked framework, the as-synthesized C4-SHz COF material
 105 was activated using supercritical carbon dioxide treatment
 106 following the reported protocol by Medina et al.³⁴ The more
 107 prominent desired peaks in X-ray diffraction for the post-
 108 activated material is observed (Figure 1d), suggesting the well-
 109 defined structure of pores. Surprisingly, the measured S_{BET}
 110 values are also mirroring this trend by narrowing the pore size
 111 (Figure S4, Supporting Information).
 112

The crystalline structure of the C4-SHz COF is resolved
 113 through the experimental powder X-ray diffraction measure-
 114 ment in combination with theoretical simulations and Pawley
 115 refinement (Figure S5 and Tables S1 and S2, Supporting
 116 Information). As seen in Figure 1c, a distinct peak at $2\theta = 3.3^\circ$
 117 is assigned to the 100 crystal plane of the COF. Additionally,
 118 relatively weak peaks at $2\theta = 5.8$ and 9.9° correspond to the
 119 reflection from 110, and 300 planes are also observed. The
 120 weak broad peak centered at $2\theta = 20$ – 26° (002 plane) is
 121 possibly due to the interlayer π - π stacking of the COF, and
 122 distances between the individual 002 planes are ca. 3.85 Å
 123

124 (Figure 1b). A probable 2D model structure was constructed
 125 using Materials Studio 7 in the $P1$ space group. An asymmetric
 126 unit in the crystal structure and graphical representation of the
 127 2D layered COF is shown in Figure S6 (Supporting
 128 Information). The simulated PXRD pattern of the eclipsed
 129 AA stacking model is well fitted with the experimental
 130 diffraction data (Figure 1c). In contrast, the PXRD pattern
 131 obtained from the AB stacking model deviates from the
 132 experimentally observed data, especially in terms of peak
 133 intensity ratios. The perfect agreement in favor of the
 134 experimental X-ray pattern was further confirmed by Pawley
 135 refinement [$R_{\text{wp}} = 7.23\%$, $R_p = 5.32\%$] (Figure S5). The unit
 136 cell parameters were as follows: $a = 31.00 \text{ \AA}$, $b = 31.46 \text{ \AA}$, $c =$
 137 7.70 \AA ; $\alpha = 99.26^\circ$, $\beta = 87.67^\circ$, and $\gamma = 120.51^\circ$.

138 The specific surface area and porous nature of the C4-SHz
 139 COF were examined by conducting N_2 adsorption–desorption
 140 analysis at 77 K. As shown in Figure 2a, the C4-SHz COF

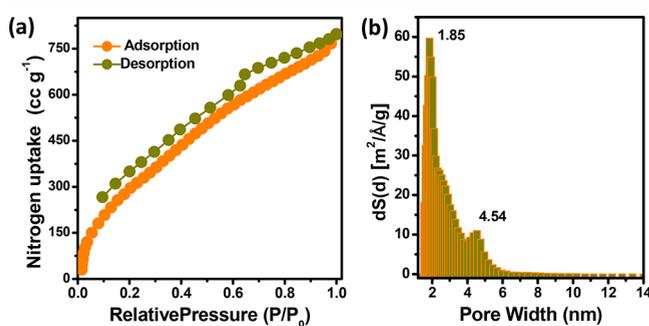


Figure 2. (a) N_2 adsorption–desorption analysis of the C4-SHz COF and (b) NLDFT pore size distribution plots.

141 exhibits a combination of type I and IV isotherms. A significant
 142 nitrogen uptake at low relative pressures ($P/P_0 < 0.03$)
 143 followed by a gradual increase in adsorption at a higher P/P_0
 144 value suggested the distinctive features for the coexistence of
 145 micropores and mesopores along with interparticle pores.^{35–38}
 146 The corresponding pore size distribution plot (Figure 2b), as
 147 determined from the adsorption branch of the isotherm based
 148 on nonlocal density functional theory (NLDFT), revealed the
 149 presence of micropores and mesopores in our C4-SHz COF
 150 material. The calculated BET (Brunauer–Emmett–Teller,
 151 S_{BET}) surface area from this isotherm was $1224 \text{ m}^2 \text{ g}^{-1}$, and
 152 the total pore volume was 1.12 cc g^{-1} . The simulated BET
 153 surface area of the AA stacked model of the C4-SHz COF has
 154 been computed using the DFTB calculations. The correspond-
 155 ing simulated specific surface area was found to be $2339 \text{ m}^2 \text{ g}^{-1}$
 156 (Figure S7, Supporting Information). The noticeable distinc-
 157 tion in BET surface area from the experimental data is
 158 related to the crystallinity of the COF material, and such
 159 deviations were also reported for other imine-based
 160 COFs.^{39–42}

161 The structural growth and imine linkage formation of the
 162 COF are confirmed by FTIR and ^{13}C MAS NMR analyses. In
 163 the FTIR spectrum of the C4-SHz COF (Figure 3a), the
 164 absorption band at 1693 cm^{-1} corresponded to $\text{C}=\text{N}$
 165 absorption, which is also confirmed by the resonance signal
 166 at 160.2 ppm in ^{13}C NMR. The peak at 3416 cm^{-1} could be
 167 attributed to $\text{N}-\text{H}$ stretching vibration. The complete
 168 utilization of the aldehyde sources was confirmed by the
 169 missing peak at 1698 cm^{-1} in the IR spectrum and the absence
 170 of aldehydic resonance peak at 190 ppm in ^{13}C NMR. In the

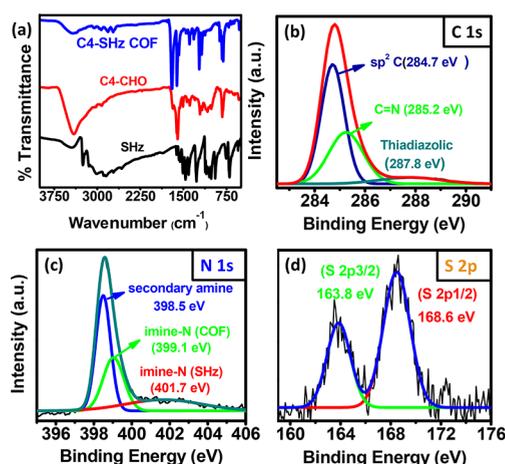


Figure 3. (a) IR spectrum of the C4-SHz COF along with the spectra of SHz and C4-CHO. (b) C 1s X-ray photoelectron spectroscopy (XPS). Deconvoluted high-resolution (c) N 1s XPS spectra and (d) S 2p XPS spectra of the thiadiazole-based COF.

spectrum in Figure 1e, the resonance signal at 160.2 ppm is
 171 ascribed to the carbon atom adjacent to the S atom of the SHz
 172 moiety. The peak at 140.2 ppm is indicative of substituted
 173 quaternary carbons of the polymeric network. The resonance
 174 peak at 132.3 ppm is attributed to the carbon atom of the
 175 aldehyde moiety that connected to the imine carbon of the
 176 organic framework. The additional peak at 126.2 ppm
 177 appeared due to the sp^2 -hybridized aromatic carbons.⁴¹

To gain further information about the surface chemistry and
 179 chemical composition, we have carried out the XPS analysis of
 180 the C4-SHz COF. As shown in Figure 3c, the N 1s spectrum
 181 can be deconvoluted into three component peaks ascribed to
 182 secondary amine ($-\text{NH}$) of the SHz moiety (398.5 eV),
 183 imine-N (399.1 eV) of the newly formed COF networks, and
 184 imine-N (401.7 eV) of the SHz moiety.⁴¹ The C nuclei with
 185 different environments were investigated by a deconvoluted
 186 high-resolution C 1s XPS scan (Figure 3b). The major
 187 component at 284.7 eV is attributed to the aromatic sp^2 C
 188 atom. Another component at 285.2 eV can be assigned to the
 189 N atom associated with imine linkage. The peak at a higher
 190 binding energy (287.8 eV) is probably due to thiadiazolic C of
 191 the COF. Strong characteristic S 2p doublet peaks (Figure 3d)
 192 were observed at 163.8 (S 2p3/2) and $168.6 \text{ eV (S 2p1/2)}$.⁴³

The thermal stability of the covalent framework was
 194 estimated by using thermogravimetric analysis (TGA) under
 195 the aerobic environment. As seen from Figure S8 (Supporting
 196 Information), an initial weight loss could be assigned to the
 197 trapped guest molecule in the porous framework. Then, with
 198 the continuous increase in temperature, the organic framework
 199 is stable up to $350 \text{ }^\circ\text{C}$. With a further increase in temperature,
 200 a continuous weight loss is associated with the burning of
 201 organics present in the material. The representative FESEM
 202 (Figure 4a) and HRTEM (Figure 4b) images of the as-
 203 synthesized thiadiazole COF have clearly shown regularly their
 204 spherical morphology with diameters of $0.5\text{--}0.8 \text{ }\mu\text{m}$. After the
 205 supercritical CO_2 activation, it was observed that the smooth
 206 spherical surface of the as-synthesized COF material trans-
 207 formed to the microflower-shaped structure (Figure 4c–f)
 208 with the size distribution of $1.2\text{--}1.8 \text{ }\mu\text{m}$. This is also reflected
 209 by the nitrogen sorption isotherm with enhancing the BET
 210 surface area (Figure S4).⁴¹

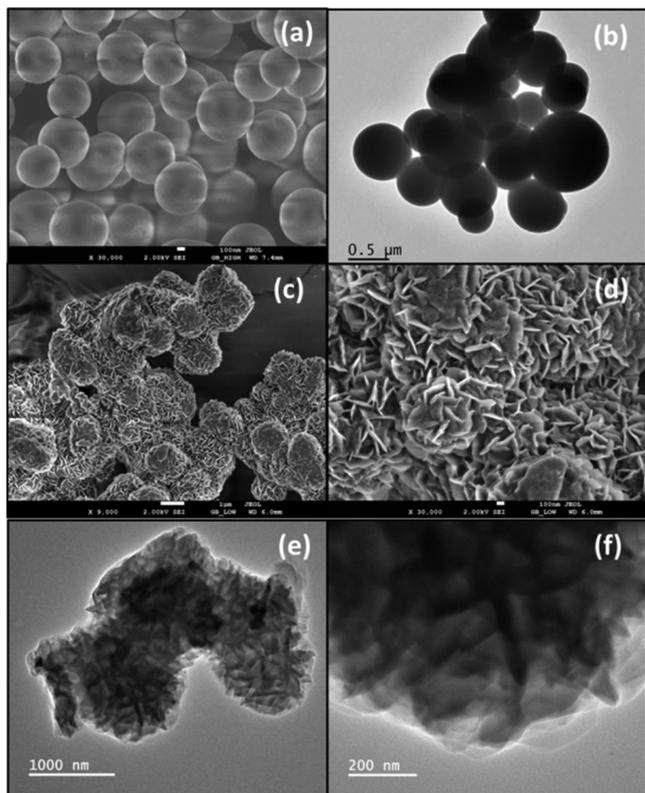


Figure 4. FESEM micrographs of the (a) as-synthesized and (c, d) crystalline C4-SHz COF. Representative TEM images of the (b) as-synthesized and (e, f) crystalline C4-SHz COF material.

To determine the electrocatalytic OER activity of the as-synthesized catalyst, linear sweep voltammetry (LSV) was recorded. The OER activity of the C4-SHz COF was compared with the state-of-the-art catalyst IrO₂/C and bare GCE under similar conditions. Before the LSV measurement,

the C4-SHz COF catalyst was electrochemically preconditioned to reach a stable state (Figure S9, Supporting Information). All the LSV plots are presented after *iR* compensation. The *iR* compensation process is explained in the Supporting Information (Figure S10). Figure 5a presents the *iR*-compensated LSV polarization plots. As expected, the bare glassy carbon electrode (GCE) exhibits a very low anodic current density with higher overpotential values, suggesting its negligible OER activity. The C4-SHz COF-modified GCE shows a sharp increase in the anodic current density, which indicates its intrinsic OER activity. For electrocatalytic activity, the overpotential (η) gives an idea about the extent of polarization upon passage of the faradic current and reflects the efficiency of the catalysts. The catalyst C4-SHz COF possesses the higher activity with a lower onset overpotential of 250 mV, which is higher than that of benchmark catalyst IrO₂/C. The current density of 10 mA/cm² is the critical value for driving solar fuel conversion and used as a benchmark for comparing electrocatalytic performance. The as-synthesized C4-SHz COF achieved a current density of 10 mA/cm² at a lower overpotential of 320 mV, which is lower than benchmark catalyst IrO₂/C. The overpotential of the C4-SHz COF is comparable with most of the reported metal-free OER electrocatalysis (Table S3, Supporting Information). This result suggests the promising electrocatalytic activity of the highly crystalline C4-SHz COF toward the OER. The OER activity is also compared with a moderately crystalline C4-SHz COF. The highly crystalline C4-SHz COF exhibits a lower overpotential with enhanced current density in comparison to moderately crystalline C4-SHz COF (Figure S11, Supporting Information). Further, the OER activity of the highly crystalline C4-SHz COF is checked in different conditions, i.e., acidic (0.5 M H₂SO₄) and neutral (1 M PBS) solutions. The C4-SHz COF exhibits the higher OER activity in 1 M KOH (Figure S12, Supporting Information).

The reaction dynamics and mechanism of the as-synthesized catalysts toward the OER were investigated by measuring the

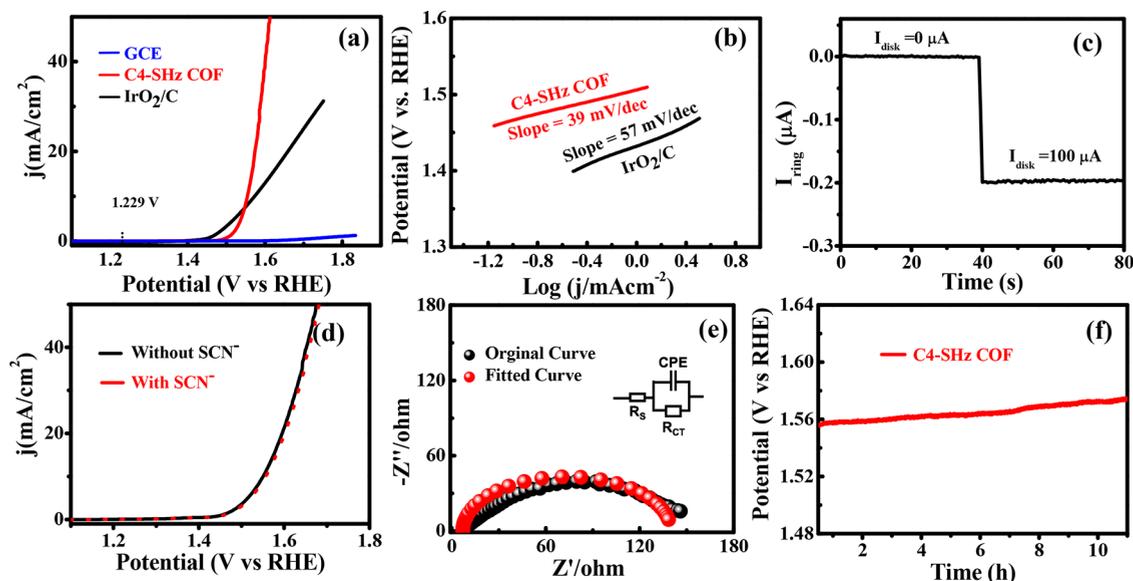


Figure 5. OER performance of the catalysts in 1 M KOH with a scan rate of 5 mV/s. (a) LSV polarization plots of the C4-SHz COF, IrO₂/C, and GCE. (b) Tafel plots of the C4-SHz COF and IrO₂/C. (c) Faradic efficiency calculation of the C4-SHz COF by using the RRDE technique. (d) LSV polarization plot of the C4-SHz COF before and after the addition of 10 mM KSCN. (e) EIS Nyquist plot for the C4-SHz COF. (f) Chronopotentiometry measurement shows the stability of C4-SHz COF.

254 Tafel slope from Tafel plots. The Tafel plots are obtained from
255 the LSV measurements and reflect the relationship between
256 the overpotential (η) and the log of current density. The Tafel
257 slope is derived from the linear part of the Tafel plot, and lower
258 the Tafel slope value indicates the higher electrocatalytic
259 activity. The C4-SHz COF exhibits the lower Tafel slope of 39
260 mV/dec, which is lower than that for IrO₂/C (57 mV/dec)
261 (Figure S**b**). The Tafel slope of the C4-SHz COF is lower than
262 that of other reported metal-free catalysts. The exchange
263 current density (j_0) of the C4-SHz COF is calculated from the
264 Tafel plot by the extrapolation method (Figure S**13**,
265 Supporting Information). The exchange current density (j_0)
266 of the C4-SHz COF is estimated to be 3.95×10^{-4} mA/cm².
267 The value of j_0 is also proportional to the electrochemically
268 active surface area. The small Tafel slope and large exchange
269 current density indicate that the C4-SHz COF can produce
270 superior metal-free OER performance. The faradic efficiency
271 (FE) of the catalysts is measured by the rotating ring-disk
272 electrode (RRDE) experiment.^{44,45} The RRDE experimental
273 details are discussed in the Supporting Information. The FE is
274 calculated from the ratio of the ring current to the disk current
275 (eq S**3**, Supporting Information). A constant current of 100 μ A
276 is applied to the disk electrode, and a 19.6 μ A current was
277 generated at the ring electrode (Figure S**c**), which corresponds
278 to the promising FE of the C4-SHz COF (98%). The
279 formation of the oxygen bubble during the OER was supported
280 by the polarization curve of the oxygen reduction reaction
281 (ORR) collected from the ring electrode at a constant disk
282 potential of 1.50 V (Figure S**14a**, Supporting Information).
283 The number of electron transfer during the OER was also
284 calculated from the RRDE measurement. During OER
285 polarization, a very negligible current density was observed
286 in the ring electrode that is assigned to the electro-oxidation of
287 generated H₂O₂ (Figure S**14b**). This observation validates the
288 four-electron pathway for the OER.

289 Generally, the precursor used for the synthesis of the COF
290 or the electrolyte used for testing the electrocatalytic activity
291 contains some metal catalysts (i.e., Fe³⁺), and it may contribute
292 toward the OER activity of the as-synthesized materials. To
293 rule out the contribution of metal ions (i.e., Fe³⁺), SCN⁻ tests
294 were carried out. The LSV data of the C4-SHz COF is
295 collected before and after the addition of the SCN⁻ ions, and it
296 suggests that SCN⁻ ions did not affect the OER activity, i.e.,
297 the OER activity is due to the metal-free C4-SHz COF catalyst
298 (Figure S**d**). This suggests that both the C4-SHz COF and
299 electrolyte (KOH) are free from metal impurities.¹⁴ The
300 enhanced catalytic activity of the C4-SHz COF was further
301 accessed by calculating active sites and the intrinsic catalytic
302 activity. Therefore, we explored the mass activity, specific
303 activity, electrochemically active surface area (ECSA), and
304 roughness factor (R_f) of C4-SHz COF catalysts. At $\eta = 320$
305 mV, the mass activity and specific activity for the C4-SHz COF
306 catalyst were found to be 286 A g⁻¹ and 0.011 mA/cm²,
307 respectively. To measure the electrochemically active surface
308 area (ECSA) and roughness factor, electrical double layer
309 capacitance (Cdl) of the C4-SHz COF was evaluated by
310 measuring the CV in the non-faradic region at scan rates of
311 10–100 mV/s in 1 M KOH at a potential ranging from 0 to
312 0.1 V (Ag/AgCl) and is shown in Figure S**15a**, Supporting
313 Information. The value of Cdl is estimated from the linear
314 slope and found to be 2.75 mF cm⁻² (Figure S**15b**, Supporting
315 Information). The ECSA and R_f values of the C4-SHz COF
316 were estimated to be 68.75 cm² and 968.30, respectively. Thus,

the C4-SHz COF shows a high electrochemical catalytic 317
surface area that may be due to the exposed excess nitrogen 318
atom on the surface, which plays as the active sites for the OER 319
activity. The high surface area and the porous structure can 320
facilitate the diffusion of the electrolyte to access more number 321
of reactant species on the electrode surface that resulted in an 322
enhanced performance during the OER. The high mass and 323
specific activity of the C4-SHz COF may also be associated 324
with fast charge transfer during the OER process. The 325
electrochemical impedance spectroscopic (EIS) Nyquist plot 326
of the C4-SHz COF is also presented in Figure S**e**. The smaller 327
polarization resistance of the C4-SHz COF indicates the higher 328
charge transfer kinetics and faster electron transfer process that 329
supports the OER activity.⁴⁶ The durability is another key 330
factor to evaluate the catalytic activity in practical application. 331
The long-term durability of the C4-SHz COF has been studied 332
by chronopotentiometry measurements. The time-dependent 333
chronopotentiometry measurements demonstrate that the C4- 334
SHz COF is a stable material and can work efficiently for more 335
than 11 h at a current density of 10 mA/cm² (Figure S**f**). 336
Further, the stability of the C4-SHz COF at higher current 337
densities (20 and 50 mA/cm²) has been checked with time- 338
dependent chronopotentiometry measurement (Figure S**16**, 339
Supporting Information). The as-synthesized material shows a 340
quite stable response at higher current densities as well. The 341
above measurements demonstrate the enhanced efficiency and 342
robustness of the C4-SHz COF toward the OER and validate 343
its promising application in future energy devices. It is quite 344
essential to understand the structure, morphology, and any 345
alternation of the bonding connectivity of the C4-SHz COF 346
after the durability test. The post-OER analysis validates the 347
robust properties of the electrocatalyst for practical application. 348
Therefore, PXRD and FTIR analyses have been carried out 349
after the stability test. In the PXRD patterns, no change in the 350
crystal phase was observed (Figure S**17**, Supporting 351
Information). This result suggested the stability of the 352
crystalline COF structure in the electrocatalyst. FTIR spectra 353
confirm the retention of the bonding connectivity within the 354
organic framework (Figure S**18**, Supporting Information) after 355
the OER stability test, suggesting the C4-SHz COF as a robust 356
material. 357

The mechanism of the OER process and the efficiency of the 358
C4-SHz COF material toward the electrochemical OER are 359
rationalized through a density functional theory study based on 360
first-principle calculations using the VASP simulation package. 361
We have constructed a periodic monolayer model of C4-SHz o 362
(cell size: 31.09 \times 27.22 \AA^2) consisting of 123 atoms, as shown 363
in Figure 6a. The efficiency of the OER can be determined by 364
calculating the reaction free energies of the individual 365
elementary steps.⁴⁷ Therefore, the free-energy change for the 366
adsorption of the intermediates (i.e., OH*, O*, and OOH*) 367
on the C4-SHz surface has been calculated, and free-energy 368
profiles for OER pathways at $U = 0$ and 1.23 V are given in 369
Figure 6c. For OH⁻ adsorption on the C4-SHz surface, there 370
are seven possible sites, including the S atom (Figure 6b and 371
Figure S**19**, Supporting Information). However, adsorption of 372
OH⁻ is more favorable on the C2 site with the highest OH⁻ 373
adsorption energy of -1.61 eV (Figure 6b and Table S**4**). 374
However, there are very weak interactions between OH⁻ and 375
either the N3 or S site (Figure S**19e,f**). Therefore, the active 376
site for OH⁻ adsorption is predominantly the C2 site. From 377
the free-energy profile diagram (Figure 6c), it is evident that 378
OOH* adsorption with the highest (most positive) free-energy 379

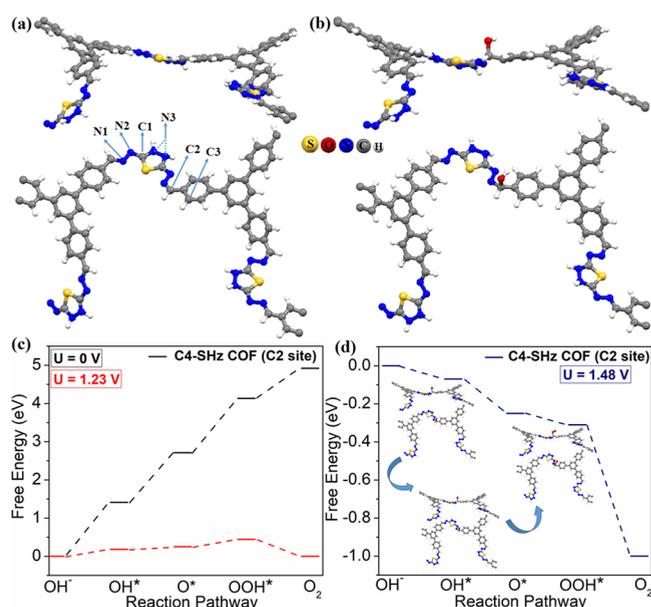


Figure 6. (a) Theoretically optimized structure of the C4-SHz monolayer (different C and N sites are marked) and (b) most stable structure of the OH* adsorbed C4-SHz surface (all other possible OH⁻ adsorption sites are shown in Figure S19, Supporting Information). (c) Free-energy profile for the OER pathway in alkaline medium on the C2 site at $U = 0$ and 1.23 V for the C4-SHz COF. (d) Free-energy profile at the experimentally determined onset potential, i.e., at an applied bias of $U = 1.48$ V (inset shows the structures of adsorption intermediates OH*, O*, and OOH* on the C2 site of the C4-SHz surface).

380 value is the rate-determining step during the OER process for
381 the C4-SHz COF.⁴⁷

382 The free-energy values of OOH* adsorption on the C4-SHz
383 surface are 4.13 and 0.44 eV at $U = 0$ and 1.23 V. This finding
384 is consistent with the earlier reported COF material as the
385 metal-free OER electrocatalyst.³³ To get further insight about
386 the excellent electrocatalytic activity of the C4-SHz COF, we
387 have performed density of states (DOS) analysis. Total and
388 projected density of states (TDOS and PDOS) analysis clearly
389 demonstrates that the VBM is mainly composed of the 2p
390 orbital of C atoms, whereas the 2p orbitals of heteroatoms, i.e.,
391 N and S atoms, contribute to the CBM (Figure S20,
392 Supporting Information). Among various C sites, as evident
393 from Figure S20b, the major contribution to the VBM comes
394 from the C2 sites. Therefore, upon adsorption of energy, holes
395 will be generated mainly on C atoms (especially on the C2
396 atom), and heteroatom sites will be electron-rich. For the OER
397 process in alkaline medium, OH⁻ is oxidized to O₂ after
398 absorbing holes from the active C2 sites, which is consistent
399 with previously reported COFs and heteroatom-doped
400 graphene systems.^{33,48,49} Besides, we have determined the
401 free-energy profile for the OER pathway at an applied bias of
402 1.48 V (Figure 6d), which is the experimentally determined
403 onset potential. Interestingly, the energy profile diagram
404 exhibits the downhill OER pathway at $U = 1.48$ V, validating
405 the experimental findings of the OER onset overpotential.
406 Moreover, the downhill OER pathway at $U = 1.48$ V also
407 indicates that the lower overpotential (merely 250 mV) arises
408 due to the C2 active site on the surface.³³

CONCLUSIONS

409

In summary, we have fabricated a novel thiadiazole-based
410 covalent organic framework, C4-SHz COF, through the Schiff
411 base condensation polymerization reaction between 1,3,5-
412 tris(4-formylphenyl)benzene and 2,5-dihydrazinyl-1,3,4-thia-
413 diazole under vacuum in a sealed tube. The resultant material
414 displayed a unique molecular architecture, high porosity, BET
415 surface area, and accessible active sites, which could enable this
416 COF as a superior self-supported metal-free OER electro-
417 catalyst. In the alkaline system, the material showed a low
418 onset potential of 270 mV, and the material achieved a current
419 density of 10 mA/cm² with a lower overpotential of 320 mV.
420 The accomplished OER activity is comparable to other best
421 reported metal-free catalysts with excellent durability. There-
422 fore, the catalytic performance exhibited by C4-SHz COF
423 suggests that it may replace metal-free carbon-based nanoma-
424 terials and could be a promising electrocatalyst for future
425 energy conversion from abundant water resources. 426

ASSOCIATED CONTENT

427

Supporting Information

428

The Supporting Information is available free of charge at
429 <https://pubs.acs.org/doi/10.1021/acscatal.9b05470>. 430

Materials, methods and characterization tools, exper-
431 imental details, Pawley refined powder X-ray profile with
432 other related data, simulated N₂ adsorption isotherm,
433 TGA profile, comparison table with other reported
434 catalysts, CV plot of materials before LSV measurement,
435 *i*R-corrected LSV curve, exchange current density plot,
436 RRDE measurement plot, ECSA plot, stability at
437 different density plot, PXRD and FTIR of used COF
438 materials, and computational OER mechanism related
439 data (PDF) 440

(PDF) 441

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476 approval to the final version of the manuscript.

477 Notes

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