

1 Molecular dynamics simulations of structural transformation of
2 perfluorooctane sulfonate (PFOS) at water/rutile interfaces

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10 **ABSTRACT:** Concentration and salinity conditions are the dominant environmental factors
11 affecting the behavior of perfluorinated compounds (PFCs) on the surfaces of a variety of solid
12 matrices (suspended particles, sediments, and natural minerals). However, the mechanism has
13 not yet been examined at molecular scales. Here, the structural transformation of perfluorooctane
14 sulfonate (PFOS) at water/rutile interfaces induced by changes of the concentration level of
15 PFOS and salt condition was investigated using molecular dynamics (MD) simulations. At low
16 and intermediate concentrations all PFOS molecules directly interacted with the rutile (110)
17 surface mainly by the sulfonate headgroups through electrostatic attraction, yielding a typical
18 monolayer structure. As the concentration of PFOS increased, the molecules aggregated in a
19 complex multi-layered structure, where an irregular assembling configuration was adsorbed on
20 the monolayer structure by the van der Waals interactions between the perfluoroalkyl chains.
21 When adding CaCl_2 to the system, the multi-layered structure changed to a monolayer again,
22 indicating that the addition of CaCl_2 enhanced the critical concentration value to yield PFOS
23 multilayer assemblies. The divalent Ca^{2+} substituted for monovalent K^+ as the bridging
24 counterion in PFOS adsorption. MD simulation may trigger wide applications in study of
25 perfluorinated compounds (PFCs) from atomic/molecular scale.

26

27 **Keywords:** perfluorinated compounds; interfacial behavior; assembling structure; environmental
28 factors; concentration effect; salinity effect

29 **1. Introduction**

30 Perfluorinated compounds (PFCs) occur globally in water and soil environments (Lau et al.,
31 2007; Route et al., 2014). Perfluorooctane sulfonate (PFOS, $C_8F_{17}SO_3^-$) is the most commonly
32 measured PFCs, and has been included in the list of Stockholm Convention on Persistent
33 Organic Pollutants (POPs) in 2009 as a global contaminant (Wang et al., 2009). Different from
34 other POPs, PFOS has high water solubility. The high persistence and long-range transport of
35 PFOS in environments have prompted increasing concerns regarding its interfacial behavior that
36 is regarded as central to the environmental fate of PFOS and the treatment of high-concentration
37 wastewater (Pan and You, 2010; Xiao et al., 2011; Zhou et al., 2013).

38 So far, although PFOS has been investigated extensively, little is known about its interfacial
39 microstructure and the interaction that hold PFOS to the surface because it is inaccessible from
40 the existing experimental measurement (Du et al., 2014). However, the microscopic information
41 is essential to enhance our understanding of the environmental fate of PFOS and to develop high
42 performance adsorbents, because the change of assembling structure and interaction may
43 produce a significant impact on its reversibility, stability, and transport in the environment and the
44 efficiency of water treatment. Although the critical micelle concentration (CMC) of PFOS is
45 4573 mg/L, it is possible to form hemi-micelles on adsorbent surfaces when its concentration
46 reaches the value of 0.01–0.001 of the CMC (Johnson et al., 2007) (a concentration level of
47 PFOS in experiments and water treatment systems). Therefore, increase of PFOS concentrations
48 promotes the nucleation of hemi-micelles and micelles (Yu et al., 2009; Wang and Shih, 2011),
49 which may result in a substantial increase of PFOS at the interface, and a significant change in
50 the macroscopic sorption properties (e.g, the occurrence of multilayer adsorption (Zhao et al.,
51 2011)). Different types of hemi-micelles and multi-layered structures have been proposed based

52 on macroscopic experiments (Chen et al., 2011; Zhang et al., 2011b; Du et al., 2014). However,
53 these structures need to be identified at molecular scales.

54 PFOS is an anionic surfactant, and hence the ionic strength and cation type in solutions are
55 critical in the adsorption of PFOS on solid surfaces (Jeon et al., 2010; You et al., 2010; Wang
56 and Shih, 2011). Different types of cations may yield very different effects on both interaction
57 strength and structure. However, most sorption experiments provide only macroscopic
58 information (e.g., sorption capacity and sorption kinetics) about the interfacial behavior of PFOS,
59 but little insight into the interaction and microstructure. In fact, changes in the sorption isotherms
60 driven by concentration and salinity conditions depend on the interactions between hydrophobic
61 tails, repulsions between headgroups and interactions between surfactant molecules with the
62 solid surface (Yu et al., 2009; Wang and Shih, 2011). These mechanisms are required to be
63 verified by using atomic-molecular level technologies.

64 Molecular dynamics (MD) simulation, which solves the movement of atoms and molecules,
65 provides detailed insights pertaining to structure, energetics, and dynamics of complex
66 multiphase systems. MD simulation has been successfully used to investigate the interfacial
67 behavior of alkyl surfactants, such as sodium dodecyl sulfate (SDS) and n-alkyl polyethylene
68 oxide (Srinivas et al., 2006; Tummala and Striolo, 2008; Dominguez, 2011; Lin et al., 2011;
69 Nunez-Rojas and Dominguez, 2011). Compared with alkyl surfactants, the perfluoroalkyl
70 surfactant may present different assembling features due to the super hydrophobicity and
71 counterion-bridging effects. To our knowledge, how the environmental factors such as
72 concentration and salinity affect the structure and behavior of perfluorinated compounds (PFCs)
73 at solid-water interfaces has not yet been investigated at molecular scales.

74 TiO₂ is one of the most common and technologically important metal oxides for modern
75 science and technology, and shows a relatively high affinity to PFOS (Yuan et al., 2001). Rutile
76 TiO₂ was chosen as the model oxide surface in this MD study. This study is expected to provide
77 an effective way to investigate the microstructures and interaction properties of PFOS at water-
78 metal oxide (e.g., Fe-, Al-, Mn-, and Ti-oxides) interfaces, and may trigger wide applications of
79 MD simulation in describing the structure and environmental behavior of PFCs as MD technique
80 is improving.

81 Here, the structural transformation and interaction properties of PFOS assembled at the
82 water-rutile interface as a function of concentration and salinity conditions were studied using
83 molecular dynamics (MD) simulation. The surface aggregation was characterized in terms of
84 structure, shape, and associated mode. Effects of salinity on the surface assembling properties of
85 PFOS were investigated by adding CaCl₂ to the simulated system. The conformational feature of
86 PFOS was described using the atomic density profile and angle distribution analysis. The role of
87 counterions (K⁺, Ca²⁺) in the formation of surface aggregates was characterized with the radial
88 distribution functions between atoms.

89 **2. Methods**

90 All simulations were carried out using the COMPASS (Condensed-phase Optimized
91 Molecular Potentials for Atomistic Simulation Studies) force field as implemented in the
92 Discover module in Materials Studio package (Accelrys Software Inc.). The functional form of
93 COMPASS force field is given as follows (Sun, 1998; Zhang et al., 2011a):

$$E_{pot} = \sum_b \left[K_2(b - b_0)^2 + K_3(b - b_0)^3 + K_4(b - b_0)^4 \right] \quad (a)$$

$$+ \sum_\theta \left[H_2(\theta - \theta_0)^2 + H_3(\theta - \theta_0)^3 + H_4(\theta - \theta_0)^4 \right] \quad (b)$$

$$+ \sum_\phi \left[V_1(1 - \cos(\phi - \phi_0^0)) + V_2(1 - \cos(2\phi - \phi_2^0)) + V_3(1 - \cos(3\phi - \phi_3^0)) \right] \quad (c)$$

$$+ \sum_\chi K_\chi(\chi - \chi_0)^2 \quad (d)$$

$$+ \sum_b \sum_{b'} F_{bb'}(b - b_0)(b' - b_0') \quad (e)$$

$$+ \sum_\theta \sum_{\theta'} F_{\theta\theta'}(\theta - \theta_0)(\theta' - \theta_0') \quad (f)$$

$$+ \sum_b \sum_\theta F_{b\theta}(b - b_0)(\theta - \theta_0) \quad (g)$$

$$+ \sum_b \sum_\phi (b - b_0)(V_1 \cos \phi + V_2 \cos 2\phi + V_3 \cos 3\phi) \quad (h)$$

$$+ \sum_b \sum_{b'} (b' - b_0')(V_1 \cos \phi + V_2 \cos 2\phi + V_3 \cos 3\phi) \quad (i)$$

$$+ \sum_\theta \sum_\phi (\theta - \theta_0)(V_1 \cos \phi + V_2 \cos 2\phi + V_3 \cos 3\phi) \quad (j)$$

$$+ \sum_\phi \sum_{\theta'} \sum_{\theta''} K_{\phi\theta\theta'} \cos \phi (\theta - \theta_0)(\theta' - \theta_0') \quad (k)$$

$$+ \sum_{ij} \frac{q_i q_j}{\epsilon_{ij}} \quad (l)$$

$$+ \sum_{ij} \left[\frac{A_{ij}}{r_{ij}^9} - \frac{B_{ij}}{r_{ij}^6} \right] \quad (m)$$

94

95 The total potential energy (E_{pot}) contains the bond stretching (a), angle bending (b), torsion

96 (c), out of plane coordinate (d), cross-coupling (e) to (k), Coulombic interaction (l), and van der

97 Waals interaction (m) terms. K, H, F, and V are force-field parameters. The b, θ , ϕ , and χ

98 represent bond length, bending angle, torsion angle, and out-of-plane angle, respectively. The

99 parameters b_0 , θ_0 , ϕ_0 , and χ_0 are the ideal values at zero energy. COMPASS has proven to be

100 suitable for simulating titanium oxides (Kornherr et al., 2004; Kornherr et al., 2006; Zhang et al.,

101 2011a), surfactants (Ryjkina et al., 2002), and perfluorinated compounds (Prathab et al., 2006; Li

102 et al., 2013). The agreement between our calculated PFOS structure and the DFT-calculated

103 results (see Supplementary data) indicated that the COMPASS force field is reliable for the
104 description of PFOS. Partial atomic charges used in COMPASS are presented in Table 1.

105 The (110) crystal plane was used as the surface because it is the predominantly exposed
106 plane of natural rutile (60%) (Perron et al., 2007). The dimensions of the solid model were
107 $4.1 \times 3.9 \times 1.3 \text{ nm}^3$, which contained 766 TiO_2 molecules. The initial configuration of the
108 interfacial system was prepared from a monolayer of PFOS molecules (a sulfonate headgroup
109 attached to eight perfluorinated carbon atoms with a length of $\sim 1.0 \text{ nm}$) with the sulfonate
110 headgroups pointed to the surface ($\sim 8 \text{ \AA}$ above the substrate). The initial structure of PFOS is
111 shown in Table 2. K^+ ions were added close to the headgroups to neutralize the charge of anionic
112 PFOS (i.e., the potassium salt of PFOS). To reach a water density of $\sim 1.0 \text{ g/cc}$ within the
113 solution layer of $4 \times 4 \times 4 \text{ nm}^3$, 2000 water molecules were used to simulate the solvent
114 environment. The water was simulated using the simple point charge (SPC) model (Berendsen et
115 al., 1987), which worked reliably with the COMPASS force field. The atoms in rutile were
116 constrained to their equilibrium bulk positions, while the structures of PFOS and water
117 molecules were optimized during the simulations (Li and Choi, 2007; Prathab et al., 2007;
118 Nunez-Rojas and Dominguez, 2011). Periodic boundary conditions were imposed, and these
119 systems were separated from their periodic images by a vacuum gap of 6 nm to eliminate
120 spurious interactions between the periodic replicas in the Z direction (Dominguez, 2009; Monti
121 et al., 2012). The resulted final box was approximately $4 \times 4 \times 12 \text{ nm}^3$.

122 The simulation was performed under three different PFOS concentrations (25, 36, and 64
123 PFOS molecules) to investigate how the concentration levels affected the interfacial assembling
124 behavior of PFOS. In order to explore the salinity effects, a certain number of Ca^{2+} and Cl^- were
125 added to the systems to achieve a ratio of PFOS to CaCl_2 being 1:1. The systems were first

126 energetically minimized for 5000 steps using the smart minimizer method, which switches from
127 steepest-descent to conjugated gradient method as the energy derivatives decrease to accelerate
128 the computation. After the initialization, all the MD simulations were conducted in the canonical
129 ensemble (NVT) at a constant temperature of 298 K maintained using Andersen thermostat
130 (Andersen, 1980). The equations of motion were integrated with the velocity Verlet algorithm
131 with a time step of 1 fs (Verlet, 1967). The electrostatic interactions were treated using the Ewald
132 method (Plimpton, 1995), and the Van der Waals interactions were handled with atom-based
133 summation method using a cutoff distance of 9.5 Å with long-range corrections added. The
134 simulations were confirmed to have reached equilibrium within 15 ns by monitoring the structure
135 and shape of PFOS aggregates as a function of time. Therefore, all the systems were finally
136 equilibrated for 18 ns.

137 The selection of parameters and models was justified by performing test calculations (see
138 Supplementary data). Increasing the orbital cutoff from 9.5 to 12.0 Å had no obvious effect on
139 the interfacial structure and properties of PFOS. Similarly, no obvious change was found in the
140 equilibrium structure when we started the simulation with a different initial configuration (i.e.,
141 the PFOS molecules parallel to the surface), indicating that the MD results do not depend on the
142 initial conditions. These tests verified that the present computational settings and models were
143 reliable for describing the structure and properties of PFOS at water-rutile interface.

144 **3. Results and discussion**

145 *3.1. Concentration effects*

146 The equilibrium structures of PFOS assembled at the water-rutile interface under three
147 different (low, intermediate, and high) concentration conditions are present in Fig. 1. A structural
148 transformation driven by an increment of PFOS concentration was clearly identified. At the low

149 and intermediate concentrations, all PFOS molecules directly interacted with the rutile (110)
150 surface mainly by the sulfonate headgroups through electrostatic attraction, and well arrayed in a
151 monolayer (Fig. 1a-b). It was observed that the sulfonate headgroups close to the solid surfaces
152 were linked together by the K^+ ions, leaving the perfluoroalkyl tails away from the surface. As
153 the concentration increased, PFOS molecules arrayed in a different pattern, where a number of
154 PFOS molecules were adsorbed on the monolayer structure by the van der Waals interactions
155 between the perfluoroalkyl chains, forming a multilayer aggregate (Fig. 1c).

156 The PFOS-surface interaction was depicted from the density profiles of the O atoms of the
157 sulfonate headgroups in the Z-direction (i.e. normal to the solid surfaces, Fig. 2a). The
158 orientation of PFOS molecules at the water-rutile interface was determined by the angle (θ)
159 between the C1-C8 vector (C1 is the C atom attached to the sulfonate headgroup, and C8 is the C
160 atom at the end of the perfluoroalkyl chain) and the surface normal (Fig. 2b). The density
161 profiles showed that the sulfonate headgroups interacted with the rutile surfaces mainly at the
162 distance of 2.0–3.5 Å (see the main peaks in Fig. 2a). At low and intermediate concentrations,
163 more than 85% of PFOS molecules were located on the surface with the angle θ less than 45°
164 (Fig. 2b), indicating that most of the PFOS preferred to align perpendicular to the substrate. This
165 conformation is favorable for the electrostatic interaction between the sulfonate headgroups and
166 the solid surface. At high concentration the second adsorbed layer was observed, corresponded to
167 the peaks of the sulfonate headgroups at 17.5-25.0 Å in Fig. 2a. In the second layer, the PFOS
168 molecules were inclined to the surface with an angle range from 0° to 90° (Fig. 2b) and hence
169 formed an irregular assembling configuration (Fig. 1c).

170 *3.2. Salinity effects*

171 When Ca^{2+} ions were added to the system, PFOS displayed a different assembling behavior
172 in terms of the morphology and layer formation (Fig. 3 and Fig. 4a). The angles θ became
173 smaller as the PFOS concentration increased (Fig. 4b), indicating that the PFOS monolayer
174 underwent an ordering transformation. When Ca^{2+} was present, PFOS molecules were inclined to
175 the surface with an angle range from 0° to 80° and formed a relatively irregular monolayer
176 structure at the low concentration, whereas arrayed in a well-ordered monolayer pattern with the
177 perfluoroalkyl chains nearly perpendicular to the substrate (all the angles θ less than 30°) under
178 the high concentration condition. At high PFOS concentration, the multi-layered structure (Fig.
179 1c) changed to a monolayer again when adding CaCl_2 (Fig. 3c), indicating that the addition of
180 CaCl_2 enhanced the critical concentration value for the occurrence of PFOS multilayer
181 adsorption.

182 3.3. Counterion-bridging effects

183 To identify the role of the counterions (K^+ and Ca^{2+}) in the interfacial processes of PFOS,
184 the K-O(3) and Ca-O(3) [O(3), the O atoms on the $-\text{SO}_3$ headgroups] radial distribution functions
185 (RDFs) were analyzed (Fig. 5a and c). An enlarged snapshot of PFOS interacted with K^+ and
186 Ca^{2+} are present in Fig. 5b and d, which provided an atomic picture for counterion-bridging
187 mechanism in PFOS adsorption. As shown in Fig. 5b, the K^+ associated with the sulfonate
188 headgroups of PFOS molecules at the surfaces, reducing the lateral repulsive force between the
189 anionic sulfonate groups and instead inducing an effective attraction. When CaCl_2 was added to
190 the system, the divalent Ca^{2+} substituted for monovalent K^+ as the bridging counterion (Fig. 5d).
191 The RDFs were dominated by two peaks at ~ 2.3 and ~ 4.3 Å for PFOS- K^+ interaction while at
192 ~ 2.0 and ~ 4.1 Å for PFOS- Ca^{2+} interaction (Fig. 5a and c), which corresponded to the distances
193 between the $\text{K}^+/\text{Ca}^{2+}$ and the neighbor and non-neighbor O atoms of two adjacent sulfonate

194 groups (Fig. 5b and d). These results of RDFs indicated that the salinity conditions imposed a
195 difference towards the surface aggregation of PFOS, and the highly charged Ca^{2+} led to stronger
196 interaction with the sulfonate groups.

197 **4. Conclusions**

198 The molecular dynamics (MD) results indicated that concentration and salinity conditions
199 significantly affected the assembling behavior of PFOS at the solid-water interface. At low and
200 intermediate concentrations, PFOS molecules directly interacted with the rutile surface mainly
201 by the sulfonate headgroups through electrostatic attraction, and aggregated in a regular
202 monolayer structure. As the concentration increased, PFOS molecules arrayed in a different
203 pattern, where an irregular assembling configuration was adsorbed on the monolayer structure by
204 the van der Waals interactions between the perfluoroalkyl chains, forming a multilayer
205 aggregate. When adding CaCl_2 to the system, the multi-layered structure changed to a monolayer
206 again, indicating that the addition of CaCl_2 enhanced the critical concentration value for the
207 occurrence of PFOS multilayer adsorption. An atomic picture for the counterion-bridging in
208 PFOS adsorption was provided, where the highly charged Ca^{2+} substituted for K^+ as the bridging
209 ion to link the sulfonate groups and thus caused the occurrence of surface aggregation. MD
210 simulation provides new perspective for the sorptive characteristics of PFOS, and may trigger
211 wide applications in study of perfluorinated compounds (PFCs) from atomic/molecular scale.

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215 the Materials Studio software.

216 **Appendix A. Supplementary data**

217 The validation of computational settings and models. Supplementary data associated with
218 this article can be found in the online version, at

219 **References**

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318

319

320 **Table 1.** Partial atomic charges used to model the assembly of PFOS at the water-rutile interface

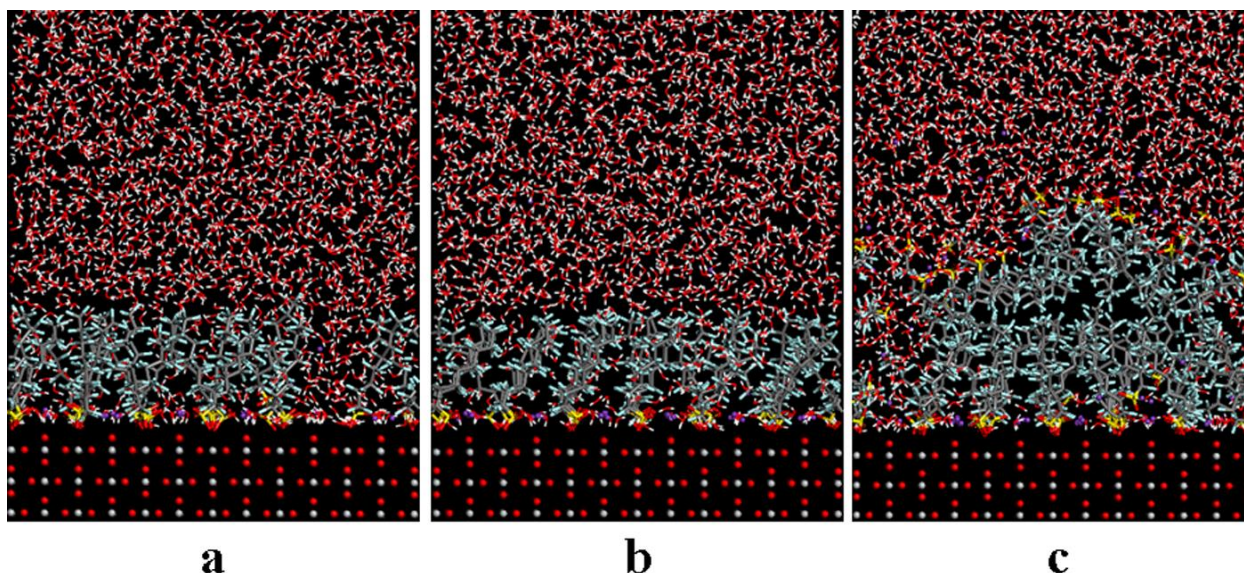
Atom	Charges/e
S	1.474
O (in SO ₃)	-0.706
C (in S-CF ₂)	0.166
C (in CF ₂ -CF ₂)	0.520
C (in CF ₃)	0.780
F	-0.260
K	1.000
Ti	1.600
O (in TiO ₂)	-0.800
O (in H ₂ O)	-0.820
H	0.410

321 **Table 2.** Initial structure of PFOS molecule*

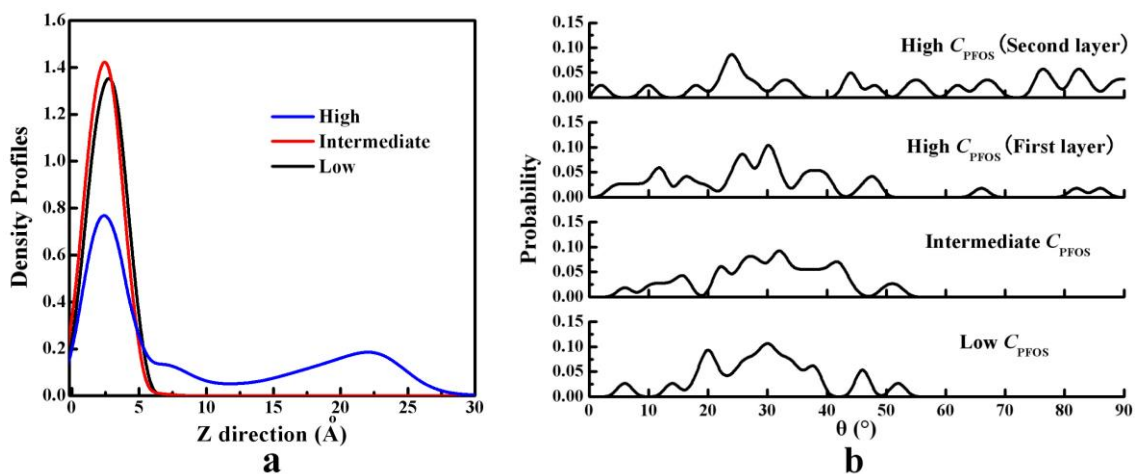
Distance	Value (Å)	Angle	Value (°)
S-O	1.48	S-C-C	116
S-C	1.91	C-C-C	114
C-C	1.57		
C-F	1.36		

322 *The structural parameters are from the DFT-calculated data in the literature (Torres et al., 2009).

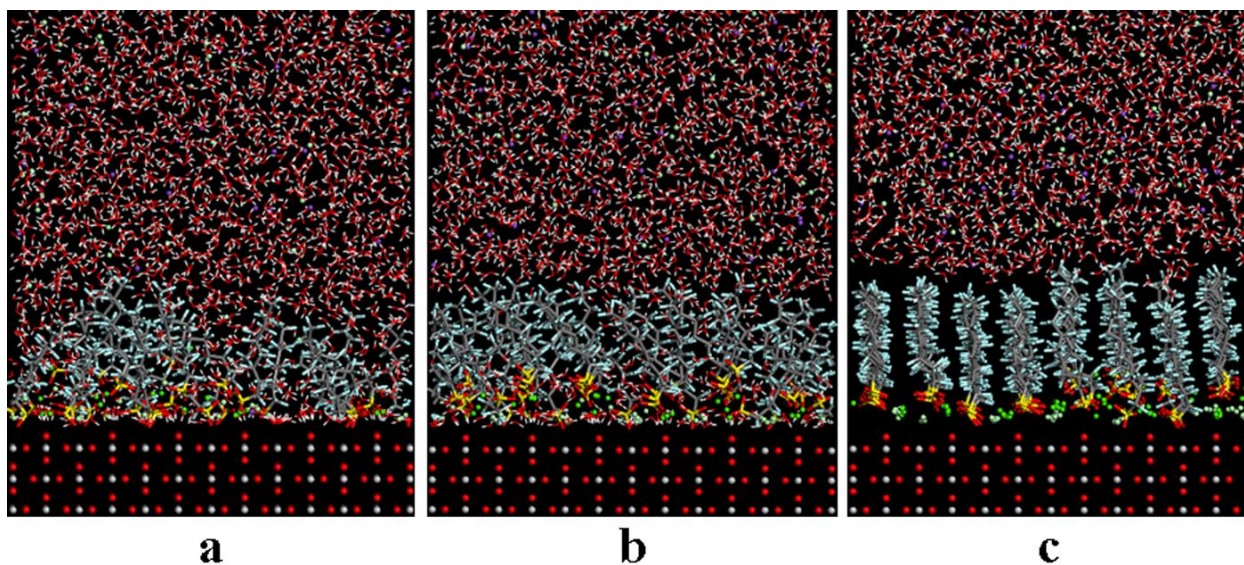
323



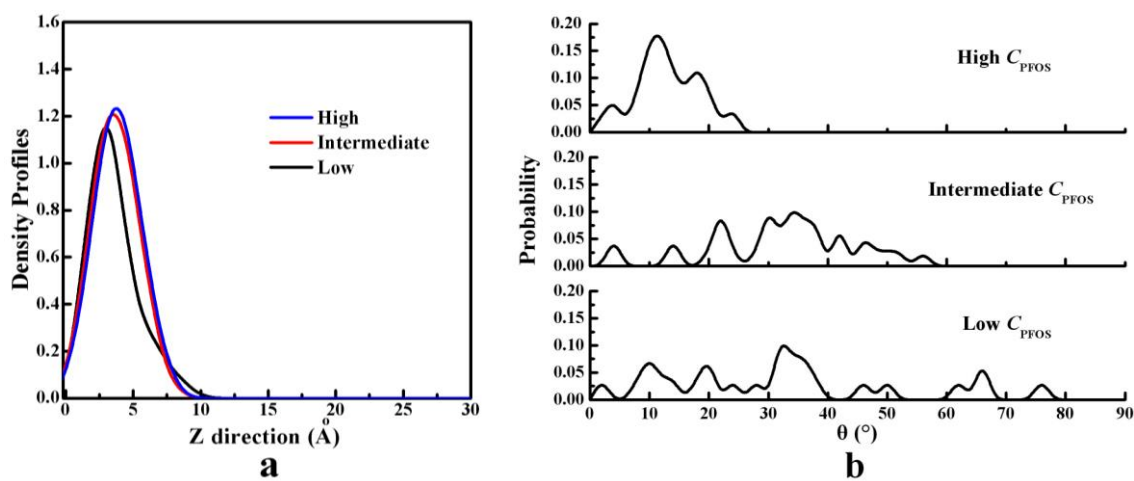
324 **Fig. 1.** Snapshot of the final structure of the PFOS aggregates on the rutile (110) surface under
 325 three different concentration conditions [(a) low; (b) intermediate; (c) high]. Yellow balls with
 326 three surrounding red balls denote the sulfonate headgroups, and the blue-gray sticks denote the
 327 C-F chains. The small purple circles represent K^+ ions, and the red-gray lines represent water
 328 molecules. For TiO_2 , gray and red balls represent Ti and O atoms, respectively.
 329



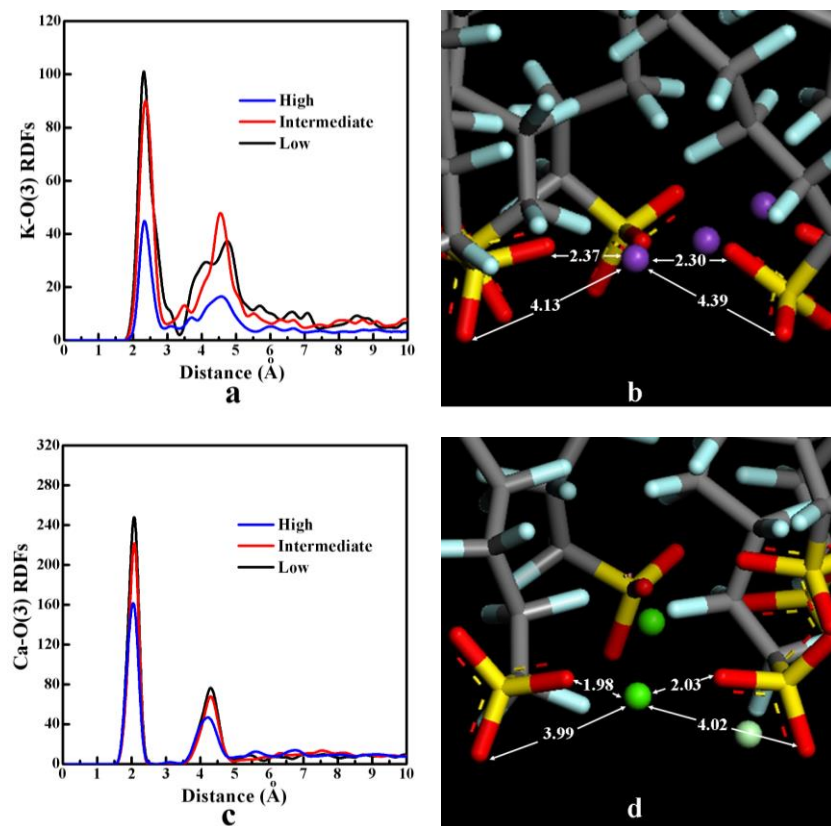
330 **Fig. 2.** Conformational feature of PFOS molecules on the rutile (110) surface at three different
 331 concentrations: (a) density profiles of the O atoms of the sulfonate headgroups along the Z
 332 direction; (b) angle distribution of PFOS with respect to the surface normal.
 333



334
 335 **Fig. 3.** Snapshot of the final structure of the PFOS aggregates on the rutile (110) surface under
 336 three different concentration conditions [(a) low; (b) intermediate; (c) high] with CaCl_2 added to
 337 the systems. The green circles represent Ca^{2+} ions, and the cyan circles represent Cl^- ions. The
 338 color legend of the other atoms is the same as that of Fig. 1.



339
 340 **Fig. 4.** Conformational feature of PFOS molecules on the rutile (110) surface at three different
 341 concentrations with CaCl_2 added to the systems: (a) density profiles of the O atoms of the
 342 sulfonate headgroups along the Z direction; (b) angle distribution of PFOS with respect to the
 343 surface normal.



344

345 **Fig. 5.** Counterion bridging in PFOS surface aggregation: (a) K-O(3) RDFs; (b) a snapshot of
 346 PFOS-K⁺ interaction; (c) Ca-O(3) RDFs; (d) a snapshot of PFOS-Ca²⁺ interaction. The color
 347 legend is the same as that of Fig. 1 and 3. The distances are shown in angstroms.