

1 **Nano architected cues as sustainable membranes for ultrafiltration in**  
2 **blood hemodialysis**  
3

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## 26 **Abstract**

27 Membranes with zeolites are encouraging for performing blood dialysis because zeolites can  
28 eliminate uremic toxins through molecular sieving. Although the addition of various pore-gen and  
29 adsorbent in the membrane can certainly impact the membrane production along with creatinine  
30 adsorption, however, it is not directed which pore-gen along with zeolite leads to better  
31 performance. The research was aimed at reducing the adsorption of protein-bound and uremic  
32 toxins by using mordenite zeolite as an adsorbent while polyethylene glycol and cellulose acetate  
33 as a pore generating agent. Membranes were cast by a phase-inversion technique which is cheap  
34 and easy to handle as compared to the electro-spinning technique. Through this strategy, the ability  
35 to adsorb creatinine and solute rejection percentage were measured and compared against the  
36 pristine PSU, when only PEG was used as a pore-modifier and when PEG along with CA was used  
37 as a pore-modifier along with a different concentration of zeolite. The experiments revealed that  
38 PEG membranes can give a better solute rejection percentage (93%) but with a low creatinine  
39 adsorption capacity that is 7654  $\mu\text{g/g}$  and low bio-compatibility (PRT 392s, HR 0.46%). However,  
40 PEG/CA membranes give maximum creatinine adsorption that is 9643  $\mu\text{g/gm}$  and also better bio-  
41 compatibility (PRT 490s, HR 0.37%) but with a low BSA rejection (72%) as compared to the  
42 pristine PSU and PEG membranes. The present study finds that the concentration of mordenite  
43 zeolite affects the membrane performance because its entrapment and large pore size of the  
44 membrane decreases solute rejection but increases creatinine uptake level along with the better  
45 bio-compatibility.

46

47 **Keywords:** Sustainable hemodialysis membranes; Mordenite Zeolite; Poly-sulfone;  
48 Hydrophilicity; Creatinine Adsorption; Uremic Toxins and Biocompatibility.

## 49 **1 Introduction**

50 Chronic kidney disease (CKD) is one of the major and serious health problems around the globe  
51 due to its high deadliness [1]. According to the U.S. Renal Data System, End-stage renal disease  
52 (ESRD) has been increased since 2003. According to literature in India and Pakistan, nearly  
53 220,000 to 275,000 new patients were reported for renal therapy [2]. The treatment of kidney  
54 failure is either a kidney transplant or hemodialysis. The option of kidney transplant is not  
55 affordable for every patient and it is a risky procedure too but hemodialysis is an alternative cure  
56 and it is a very cost-efficient treatment. In hemodialysis, there is a blood cleaning system along  
57 with the hemodialysis-membrane-based dialyzer, which circulates the blood by purifying it  
58 coming from the CKD patient [3]. Many clinical up-gradations are required, and some middle-size  
59 toxin molecules are still unresolved like indoxyl sulfate, p-cresol and creatinine. Chronic kidney  
60 disease is linked with these toxins' development and aggravation.

61  
62 In hemodialysis, the core component is the membrane. Many researchers had worked on the  
63 modification of the membrane by using different polymers along with additives to optimize the  
64 rejection capabilities. Different varieties of polymers that were used are cellulose acetate (CA),  
65 polysulfone (PSU) [4], polyethersulfone (PES) [5], polymethyl methacrylate (PMMA),  
66 polyacrylonitrile (PAN) [6], polyvinyl alcohol (PVA) [7], polylactic acid (PLA), polypropylene  
67 (PP), polyamide, chitosan, here every polymer had different abilities in terms of biocompatibility  
68 and performance efficiency [8]. PSU is considered the best base polymer because it contains the  
69 best mechanical, chemical property and processability. Meanwhile, PSU can remain steady in all  
70 disinfection conditions (steam, ethylene oxide, and gamma radiation, etc.) even it is one of the few  
71 biomaterials [9]. The PSU membrane becomes the main selection for the clinicians managing

72 dialysis as from literature shows the best and higher clearance rate of uremic toxins as well in  
73 comparison to PES and CA membranes [10]. As the PSU itself is highly hydrophobic in nature  
74 that leads to respectively low hemocompatibility. Hence anticoagulants are always used to reduce  
75 clot formation during the treatment [11].

76

77 Before using the PSU-based hemodialysis membranes, certain modifications are required. To  
78 increase the biocompatibility, the researchers have tried to increase the surface modification  
79 techniques such as sulfonated hydroxypropyl chitosan (SHPCS) was grafted from PSU membrane  
80 material by Schiff-Base reaction [12]. Similarly, the additives can also increase the pore formation  
81 and the distribution of the pores are affected [13]. The degradability of the dialysis membrane is  
82 affected as between the flowing fluid and the membrane for hemodialysis the shear force can head  
83 to loosen the additives from the surface of the membrane [14]. The elution of additives is also  
84 affected by the type of dialysis membrane, sterilization method, storage period, and pre-flush  
85 methods [15]. Hence optimized handling methods are required to prepare the PSU-based  
86 hemodialysis membrane.

87

88 The major toxins that are present in the human blood are classified into 3 basic categories  
89 depending upon their weight and protein-binding capacity: (1) Small molecular weight water-  
90 soluble compounds; (2) Protein-bound compounds and (3) Large sized molecules. The biological  
91 effects of protein-bound uremic toxins were also reviewed systematically [16]. To decrease the  
92 quantity of protein-bound solutes such as creatinine certain methods are designed which even  
93 includes the modification of dialysis procedure such as increasing the  $K_{oA}$  and  $Q_d$  to increase the

94 removal or by using sorbents or by just restricting their production [17]. But these methods are  
95 expensive.

96

97 The hemodialysis method can remove only 30% of the protein-bound toxins because they are bind  
98 with albumin. However, it can clear more than 60% of urea & creatinine [18]. The studies  
99 suggested that uremic cardiovascular disease and kidney damage are responsible for the most  
100 functional deterioration, and it appears that the toxicity of creatinine and supports their roles in  
101 vascular and renal disease progression. Because of the activation of the accompaniment of  
102 different pathways, it is a major life-threatening complication because it could lead to more  
103 adsorption of the proteins during the hemodialysis treatment as PSU is highly hydrophobic in  
104 nature [19]. The membrane rejection performance and permeability also deteriorate because of the  
105 deposition of the protein on the membrane.

106

107 Through the study of the previous literature, it was depicted that the researchers utilized large-  
108 sized or rod-shaped adsorbents which lead to decrease surface area and reducing the adsorption of  
109 protein-bound toxin [23]. So this study worked on the gap highlighted above by using mordenite  
110 zeolite with spherical shape and size which is less than 50 nm that provides more surface area  
111 considered as the better adsorbent into the dialysis membrane as it can adsorb more protein-bound  
112 toxins onto the porous particle hence also decreasing the risk of cardiovascular disease [21].  
113 Similarly, as a hydrophilic and non-toxic polymer with high portability and anti-interference  
114 property to plasma proteins or platelets, PEG [15] and CA [22] used a pore-modifier that can  
115 upgrade the hydrophobicity, hemocompatibility, and biocompatibility of PSU.

116

117 Hence, the main motivation of this research is to make a composite membrane by using the phase  
118 inversion method which is also a cheap method and is associated with mild handling as compared  
119 to the hard handling of the electrospinning technique. This strategy directs towards the easy  
120 fabrication of the hemodialysis membrane. Similarly, the PEG and CA were used as a pore-  
121 modifier to increase the number of pores to reduce the platelets on the membrane surface. In  
122 addition, the adsorbent particles named mordenite zeolite of spherical shape were used as an  
123 additive in the membrane composition and their size was also smaller than 50 nm hence providing  
124 more adsorption sites when incorporated inside the membrane solution. Since it is also a bio-  
125 compatible material hence less platelet adhesion would also be obtained. Maximum tests were  
126 conducted to find out the performance and biocompatibility of the membrane while changing the  
127 concentration of mordenite zeolite. The ability to adsorb creatinine and solute rejection percentage  
128 were measured and compared against the pristine PSU, when only PEG was used as a pore-  
129 modifier and when PEG along with CA was used as a pore-modifier in terms of urea clearance and  
130 BSA rejection and that can also eliminate the protein-bound toxins.

## 131 **2 Material and methods**

### 132 **2.1 Materials**

133 As a membrane forming basic polymer the PSU with an average molecular weight of 30,000 Da  
134 (Sigma Aldrich) was used. The solvent N, N-dimethylacetamide (DMAc) with analytical purity of  
135 99% purchased from Sigma Aldrich. Cellulose Acetate with an average molecular weight of  
136 30,000 Da was purchased from Sigma Aldrich, PEG 400 was taken from Aladdin. Distilled water,  
137 n-hexane, and methanol were purchased from Sigma Aldrich and were used as a non-solvent agent.  
138 Mordenite zeolite as an adsorbent was purchased from Sigma Aldrich. Experiments were  
139 performed using urea with a molecular weight of 60.02 and creatinine were purchased from Sigma

140 Aldrich. Bovine serum albumin (BSA, purity > 97%) was purchased from Sigma-Aldrich. The  
141 anticoagulant sheep whole blood was purchased from Slaughter House.

## 142 **2.2 Fabrication of pristine PSU and modified membranes**

143 To synthesize the membrane, the PSU flakes were firstly dried in the drying oven at 60 °C for 24  
144 hours. Then 18 wt.% of the solution was prepared by mixing 18gm of PSU dried flakes as solute  
145 into DMAc solvent from which 7 mL of solution was utilized as a base solution. The additive PEG  
146 was prepared as 16 % weight into the DMAc solvent and from which 4 mL of solution was mixed  
147 with the base solution for pore generation. The CA solution was prepared in 8 wt.% into the DMAc  
148 solvent and from which 2 mL was added into the base solution and at the end, the mordenite zeolite  
149 was added in different concentrations while the rest was DMAc as a solvent. The solution was  
150 stirred for 24 hours at 25 °C to make the solution homogenous. The complete membrane  
151 compositions are given in Table 1.

152

153 The solutions were then sonicated for 30 minutes to remove any kind of trapped air bubbles in the  
154 solution as these bubbles can deform the membrane surface after casting. The formulated solution  
155 was then cast by using doctors' blade on the glass slab by pouring that much solution to make the  
156 membrane to the size of 0.00146 m<sup>2</sup> with a thickness of 200 μm. After evaporating for 30 to 45  
157 seconds, it is then immersed in the distilled water coagulation bath for 24 hours then in an n-hexane  
158 and methanol coagulation bath for 2 hours respectively to complete the phase inversion process as  
159 shown in Fig. 1. The side facing the non-solvent is the reactive side of the membrane. The distilled  
160 water helps in solidifying the membrane while n-hexane and methanol act as a non-solvent agent.  
161 Then the membrane is placed in a neat place for 24 hours for drying.

## 162 **2.3 Membranes characterization**

163 In SEM model JSM 6490A, JEOL analysis, the dried samples react with the electron beams when  
164 the voltage was kept at the 10 kV and produced surface and cross-sectional morphology of the  
165 membranes and their relative composition at various magnifications [23]. Dried samples were  
166 sputtered with liquid nitrogen to obtain clear cross-sectional morphology. In AFM JSPM-5200,  
167 three-dimensional topographies were obtained for the image down to the sub-nanometer range by  
168 using the high-resolution technique [24]. The AFM non-contact mode was utilized by using a  
169 silicon nitrate tip. AFM software program determined the roughness of the sample membranes  
170 from AFM 3D micrographs. MID-IR instrument was used for the chemical composition  
171 measurements of the modified membranes. The transmission method was used to find the FTIR  
172 and the range was kept at the resolution of the 4 cm<sup>-1</sup> and spectrometer range 400-4000 cm<sup>-1</sup>.  
173 Infrared radiations were used to study various functional organic groups in the membrane samples  
174 [24]. All characterization techniques were performed on the side of the membrane facing the non-  
175 solvent during the membrane fabrication.

## 176 **2.4 Hydrophilicity tests**

### 177 **2.4.1 Porosity**

178 The 1×1 cm<sup>2</sup> area of the membrane was oven-dried and weighed and then immersed in distilled  
179 water for 24 hours and weighed again. The membrane porosity can be obtained by using Eq. (1).  
180 W is the weight of the wet and dry membranes (grams),  $\rho_w$  is the density of the pure water  
181 (gm/cm<sup>3</sup>) and  $\rho_p$  is the density of polymer (gm/cm<sup>3</sup>), respectively [25].

$$182 \quad \text{Porosity } (\epsilon) = \frac{\frac{W_{wet} - W_{dry}}{\rho_w}}{\frac{W_{wet} - W_{dry}}{\rho_w} + \frac{W_{dry}}{\rho_p}} \quad (1)$$

183 **2.4.2 Degree of swelling**

184 Membranes were pre-heated at 60 °C for 12 hours and weighted ( $W_{dry}$ ) in grams. The soaked  
185 membranes were removed from the water after 24 hours and weighed again ( $W_{wet}$ ) in grams [26].  
186 The measurement was made at 5 different positions and the average was taken [26]. Then from  
187 Eq. (2).

188 
$$\text{Water Uptake (\%)} = \frac{W_{wet} - W_{dry}}{W_{dry}} \times 100 \quad (2)$$

189 **2.4.3 Contact angle measurement**

190 Contact angle system OCA (Data physics, USA) was used for this experimentation. The sessile  
191 drop method was used to measure the stable contact angle. Synthesized membrane area  $1 \times 5 \text{ cm}^2$   
192 was taken and attached to the glass slide. Distilled water was poured on the sample by using a  
193 micro syringe with a constant dosing rate of  $0.2 \mu\text{L}$ . The water drop angle was recorded and  
194 measured on the surface three times for average angle [27].

195 **2.5 Performance tests of membranes**

196 **2.5.1 Tensile strength test**

197 The ultimate tensile strength experiment was carried out with the help of Shimadzu; AGS-X series  
198 of 50KN. ASTM-standard D 8802-02 was at a strain rate of  $0.5 \text{ mm/min}$ , the stress-strain behaviour  
199 was observed for all samples[28].

200 **2.5.2 Water flux and permeability**

201 Water flux, as well as permeability experiments were performed in dead-end filtration cell with 2  
202 bar pressure maintained by nitrogen gas [29].  $0.00146 \text{ m}^2$  area of the membrane sample was used  
203 and the permeate was calculated after every 10 minutes and after 1 hour 40 minutes the flux  
204 became constant. The pure water flux and permeability were then calculated by using Eq. (3) and

205 Eq. (4) [25]. Where J is the flux in L/m<sup>2</sup> h. V is the volume of the permeated water in Litres. T is  
206 the time in hours. A represents the total area of the membrane in cm<sup>2</sup>.

$$207 \quad J = \frac{V}{A \times T} \quad (3)$$

$$208 \quad \text{Permeability} = \frac{\text{Flux}}{\text{pressure}} \quad (4)$$

### 209 210 **2.5.3 BSA rejection percentage and urea clearance**

211 After feeding 1 mg/mL BSA solution and 1mg/mL urea solution the permeate obtained after  
212 applying 2.5 bar pressure was observed under the (Shidmazu UV 1240) spectrophotometer at a  
213 wavelength of 278 nm and 190 nm. The BSA rejection percentage and urea clearance percentage  
214 were then calculated by Eq. (5) and Eq. (6) [29]. Where C<sub>p</sub> (gm/L) and C<sub>r</sub> (gm/L) are  
215 concentrations of permeate and retentate, respectively [29]. Moreover, C<sub>i</sub> (gm/L) and C<sub>f</sub> (gm/L)  
216 are initial and final concentrations at time t, respectively [29].

$$217 \quad \text{BSA Rejection (\%)} = 1 - \frac{C_p}{C_r} \times 100 \quad (5)$$

$$218 \quad \text{Solute Clearance (\%)} = \frac{C_i - C_f}{C_i} \times 100 \quad (6)$$

### 219 **2.5.4 Creatinine adsorption capacity**

220 The round modified membranes with 10mm in diameter were placed in a syringe filter cartridge  
221 (EMD Millipore, CA) to measure the creatinine adsorption capacity [21]. Then, a 400 mmol/L  
222 creatinine solution was added into the inlet of the cartridge that flows through the membrane at 2.5  
223 bar pressure. Finally, the solution collected at the outlet were measured by (Shidmazu UV 1240)  
224 UV absorption spectra at a wavelength of 190 nm. Each type of membrane was tested thrice.

225 **2.6 Hemocompatibility study of membranes**

226 **2.6.1 Static Platelet adhesion test**

227 Platelet attachment is observed by the SEM technique. Firstly, the plasma-rich-plasma (PRP) was  
228 obtained when 10 mL of anticoagulant whole blood was centrifuged at 1000 rpm for ten minutes.  
229 1×1 cm<sup>2</sup> membrane samples were washed with the phosphate buffer solution (PBS). 100 μL PRP  
230 was inserted on the samples in 24-well cultural plate. After incubation for 2 hours at 37 °C for the  
231 removal of unstable platelets, the samples were washed thrice. The membranes were then  
232 immersed in 2.5 wt% glutaraldehyde solution which was used to fasten the absorbed protein on  
233 the membrane surface for 24 hours. Further drying was done with the graded ethanol of  
234 compositions 50, 75, 85, 95 and 100% [30].

235 **2.6.2 Hemolysis Ratio measurements**

236 1×1 cm<sup>2</sup> membranes samples were washed thrice with the 0.9 wt% of the NaCl solution for ten  
237 minutes in sequence [12]. At 37 °C in a water bath for one hour the samples were kept immersed  
238 in the NaCl solution and whole blood of 200 μL was added to the membrane samples. Then this  
239 solution was centrifuged for 10 minutes at 1500 rpm and top layer absorbance was measured using  
240 545 nm by (Shidmazu UV 1240) UV spectrophotometer. The ratio was calculated using Eq. (7) in  
241 which HP and HN represent absorption value of negative reference and absorption value of the  
242 positive reference, respectively. HS represents the absorption value of membrane samples [31].

243 
$$HR = \frac{HS - HN}{HP - HN} \quad (7)$$

244 **2.6.3 Thrombus formation measurements**

245 Membrane samples of 1×1 cm<sup>2</sup> were immersed in the 1.5 mL whole blood and incubated in 5% of  
246 CO<sub>2</sub> for 2 hours at 37 °C. PBS was used after the incubation to wash the samples. The in vitro  
247 thrombus formation on the membrane surface was measured by the graded ethanol and critical

248 point drying [32]. The Eq. (8) is used in which DT is the degree of thrombus;  $W_t$  (gm) and  $W_d$   
249 (gm) represent the weight of blood-coagulated membrane and weight of the dry membrane.

$$250 \quad DT = \frac{W_t - W_d}{W_d} \quad (8)$$

#### 251 **2.6.4 Plasma clotting time measurements**

252 The plasma poor plasma (PPP) was obtained when anticoagulant 10 mL blood was centrifuged at  
253 3000 rpm for 15 minutes. 200  $\mu$ L of PPP was poured on the  $1 \times 1$  cm<sup>2</sup> membrane samples and the  
254 cultural plate was incubated in a water bath for 10 minutes at 37 °C. Then 100  $\mu$ L of the 0.025  
255 mol/L CaCl<sub>2</sub> solution was added to the samples. The mixture was stirred until any thread formed,  
256 the time consumed was recorded [33].

### 257 **3 Results and discussion**

#### 258 **3.1 Morphology and chemistry of the membranes**

259 SEM is the analytical technique that explains the surface and cross-sectional morphology of the  
260 membrane. It explains the effect on the PSU membrane by adding CA and PEG as a pore-  
261 generating agent. All the membranes SEM comparison images are shown in Fig. 2. The surface  
262 morphology revealed that when the additives were added the surface contains more non-uniform  
263 pores and long finger-like structural pores along with sub pores are formed in the cross-section  
264 [9]. With the addition of mordenite, the surface and cross-sectional morphology changed as the  
265 zeolite nanoparticles can be seen in the SEM images properly dispersed inside the cross-section  
266 but also on the surface of the membrane. It can also be seen that as the concentration of mordenite  
267 zeolite increased from 0.18 to 0.98 gm the pores become smaller making the membrane denser  
268 again which also affects the selectivity of the membrane for toxin removal.

269

270 The surface image of the pristine PSU membrane showed that pores of the top layer were very  
271 small that they cannot be shown by SEM hence the membrane surface is highly dense, whilst the  
272 PEG/CA and PEG membranes showed pores and gap structure on the membrane surface. When  
273 PEG and CA were added to the membrane solution, the casted membrane surface showed more  
274 non-uniform pores with large size along with macro voids as shown in Fig. 2(b) and (c) in  
275 comparison to the pristine PSU but at the bottom, the pores were smaller or of the same size of  
276 pristine PSU when inspected at the cross-sectional image Fig. 2(b) and (c). With the addition of  
277 only PEG, the membrane surface pores were of smaller and equal size up till the bottom of the  
278 membrane as shown in Fig. 2(d), (e), and (f). With the addition of PEG, the fingers length increased  
279 in the cross-section of the membrane as shown in Fig 2(d, e, f). However, with the addition of PEG  
280 along with CA, the width of the fingers also increased as shown in Fig. 2. The integral membrane  
281 surface roughness became very rough, even though there were some small circular structural pores  
282 on the skin of the membrane after the addition of additives. Less surface roughness can also prevent  
283 the adsorption of significant quantities of protein [34].

284

285 The surface chemical composition of modified membranes was determined by FT-IR spectroscopy  
286 and is shown in Fig. 3. In cases of PEG-1, PEG-3, PEG-5 membranes, the bands at  $1244\text{ cm}^{-1}$   
287 were due to the C-O-C bond that shows the existence of PEG additive. At near  $3023\text{ cm}^{-1}$  the slight  
288 increase in the band was due to the C-H bond that also showed the presence of PEG. In all the  
289 spectra, the existence of C-O at  $1478\text{ cm}^{-1}$  and C-O-C ether group at  $1198\text{ cm}^{-1}$  shows the presence  
290 of PEG due to asymmetric stretching in all the PEG blend membranes. Meanwhile, when PEG  
291 along with CA act as a pore-modifier for PEG/CA-1, PEG/CA-3, PEG/CA-5 membranes at  $1500\text{ cm}^{-1}$   
292 the increase in the band was due to the C-O bond this is because of the presence of CA.

293 However, the sharp band at  $1244\text{ cm}^{-1}$  was due to the C-O-C bond of the PEG additive. These C-  
294 O-C and C-O bonds not shown much stretching due to their constant wt%. The R-SO<sub>2</sub> bonds near  
295  $1100\text{ cm}^{-1}$  showed a proper dispersion of mordenite zeolite in the membrane structure and the  
296 slight stretching in the band was due to the change in its concentration. The PSU characteristic  
297 bands were around  $1149$  and  $1168\text{ cm}^{-1}$  (SO<sub>2</sub> symmetrical stretching),  $1244\text{ cm}^{-1}$  (aryl-O-aryl C-  
298 O stretching),  $1582\text{ cm}^{-1}$  (SO<sub>2</sub> asymmetric stretching),  $1677\text{ cm}^{-1}$  (asymmetric-CH<sub>3</sub>), and  $2151$   
299  $\text{cm}^{-1}$  (C=C) [35].

300

301 The AFM explains the surface topography of the membrane surface as shown in Fig. 4. All the  
302 samples were examined under AFM in tapping mode. 3D AFM images of the top surface of all the  
303 membranes with a scanning area of ( $10\times 10\text{ }\mu\text{m}$ ) were taken as 3D images can identify the surface  
304 roughness and smoothness. The dark regions showed depths and the light regions defined the  
305 heights on the surface topography [36]. The optimum surface roughness is required to obtain better  
306 biocompatibility. Fig. 4 shows that the pristine PSU membrane surface was smooth enough but  
307 when the PEG, CA, and mordenite zeolite were added the membrane became highly rough because  
308 of the pore and the macro void formation, and as the amount of mordenite zeolite was increased  
309 the membrane roughness started decreasing because of the presences of mordenite zeolite because  
310 the void spaces between the pores started decreasing hence making the membrane less smooth.

311

312 Fig. 4(a) revealed the pristine PSU was highly smooth but when CA along with PEG were added  
313 as an additive, the membrane became highly rough Fig. 4(b, c). When only PEG acted as pore-  
314 modifier the membrane showed less roughness in comparison to PEG/CA membranes as shown in  
315 Fig. 4(d, e, f). The lesser the roughness, the better be the biocompatibility results because of the

316 low adsorption of protein on its surface. It would also give good fluxes and most importantly low  
317 fouling rates [37]. From Fig. 4 the AFM images justify the statement that PEG membranes show  
318 optimum smoothness than PEG/CA and PSU membranes.

## 319 **3.2 Membrane performance evaluation for dialysis**

### 320 **3.2.1. Evaluation of hydrophilicity level of the membranes**

321 The increment in the contact angle is due to the higher densities and compaction of the synthesized  
322 membranes. As shown in Fig. 5 (b) that pristine PSU is highly hydrophobic in nature because the  
323 presence of a dense surface giving an angle of  $87^{\circ} \pm 5^{\circ}$  that make it less promising for the  
324 hemodialysis process. However, when PEG along with CA used as a pore-modifier, the minimum  
325 angle reached  $48.1^{\circ} \pm 5^{\circ}$  for PEG/CA-1 membrane, this is just because the formation of the pore and  
326 macro voids on the surface of the membrane and sub pores were also increased in the finger-like  
327 structure of the membrane that can be seen by SEM images. However, when only PEG acted as a  
328 pore-generating agent, the minimum angle obtained was  $58.6^{\circ} \pm 5^{\circ}$  for the PEG-5 membrane. Fig. 5  
329 (b) justifies that now the modified membranes are hydrophilic in nature that is the main  
330 requirement for the optimum hemodialysis process. A contact angle lesser than  $60^{\circ}$  is considered  
331 as hydrophilic and more than or equal or closer to  $90^{\circ}$  is considered hydrophobic in nature [5].  
332 When the membrane is more hydrophilic best hemocompatibility results could be obtained [12].  
333 The PEG/CA membranes are more hydrophilic in nature than pristine PSU and PEG membranes.  
334 Because of the penetration of the water into the pores of the membrane due to the capillary tube  
335 affect the porosity also plays an important role in the contact angle measurement [5].

336

337 When the porosity and the pore size distribution are changed, it causes a major effect on the  
338 permeability of water, uremic toxin clearance, and protein adsorption and rejection clearance.

339 When PEG with CA and mordenite zeolite were added to the PSU solution, the porosity percentage  
340 increased abruptly as compare to the pristine PSU this was due to the increase in the number and  
341 size of pores on the surface but also the sub pores in the finger-like structure of the membrane that  
342 can be seen in SEM imaging Fig. 2. The porosity of PEG/CA-1 was  $93.5\% \pm 5\%$ . However, for  
343 PEG/CA-3 and PEG/CA-5, the porosity was  $79.6\% \pm 5\%$  and  $72.1\% \pm 5\%$  respectively. Fig. 5 (a)  
344 showed the trend that when the concentration of zeolite increased it blocked and captured the void  
345 spaces between the pores resulting in decreasing in the porosity. However, when only PEG as an  
346 additive was added the porosity trend increased as with the increase in the concentration of  
347 mordenite zeolite. and the maximum porosity obtained for the PEG-5 membrane was  $79.2\% \pm 5\%$   
348 as shown in Fig. 5 (a). This trend showed that as the concentration of mordenite zeolite increased,  
349 the surface area of the membrane morphology also increased hence resulting in increased  
350 membrane porosity.

351  
352 The hydrophilicity and hydrophobicity of the membrane can also be determined by the swelling  
353 percentage test. High water absorption means that the membrane is hydrophilic in nature [9]. Fig.  
354 5 (a) elaborates the trend in the results. The pristine PSU membrane showed very less water  
355 absorption as the surface of the membrane was very dense containing very little and no pores. But  
356 when the PEG along with CA and mordenite zeolite as an additive were added the water absorption  
357 increased to a very high extent because of the increase in porosity justified by Fig. 5(a). As shown  
358 in Fig. 5(a), the maximum percentage was obtained for the PEG/CA-1 membrane was  $1189\% \pm$   
359  $10\%$  this was just because of an increase in the number and size of the pores on the surface but  
360 also in the cross-section of the membrane that can be seen in the SEM image Fig. 2. Hence, the  
361 PEG/CA-1 membrane is highly hydrophilic that can absorb maximum water. On the other hand,

362 when only PEG acts as a pore-modifier the trend started increasing with the increase in the  
363 concentration of the mordenite zeolite as shown in Fig. 5 (a) this happened because zeolite provides  
364 more surface area when incorporated in the membrane. PEG-5 membrane was highly hydrophilic  
365 in nature giving the percentage of  $666\% \pm 10\%$ .

### 366 **3.2.2. Effect of the toughness of membranes on hydrophilicity**

367 The addition of hydrophilic elements like PEG and CA can also affect the mechanical properties  
368 of the membranes that can also be compared by their morphology. Fig. 6 represents that the pristine  
369 PSU membrane showed the highest tensile stress of 30.76 MPa. Because the pristine PSU  
370 membrane contains a dense structure as shown in SEM image Fig. 2, the high tensile stress and  
371 strain curve justify that it was very dense and contained fewer pores on the surface, as well as  
372 short-sized fingers in the cross-section of the membrane, were present. However, when the  
373 additives like PEG along with CA were added to the membrane the tensile stress decreased  
374 abruptly to 9.98 MPa of PEG/CA-3 membrane, due to the less density, polymer packing and  
375 favourable interfacial adhesion.

376

377 However, with the addition of PEG, the tensile stress was 25.8 MPa for the PEG-1 membrane  
378 which was still lesser than pristine PSU. The mechanical properties decrease when the structure  
379 changes to more porosity and hydrophilicity [38]. Further, when mordenite zeolite concentration  
380 increased in PEG/CA membranes it increased the stress as well as strain than the previous  
381 compositions showing that the void spaces between the pores now start blocking hence making  
382 the membrane denser again. Similarly, because of the addition of the pore-modifier, the elongation  
383 rate also increased as cross-sectional morphology also influenced mechanical properties justifying

384 that the membrane contained more pores, and sub pores were also produced in the fingers in the  
385 cross-section of the membrane as shown in Fig. 2.

### 386 **3.2.3. Water flux and permeability of the membranes**

387 The efficiency evaluation of the modified and pristine PSU membrane, pure water permeability  
388 test was performed. The distilled water as a solvent was used to determine the behaviour of the  
389 membranes. The flux was then calculated and then the graph was plotted as shown in Fig. 7. When  
390 the additives were added, the void spaces, pore sizes and sub pores the permeability and flux of  
391 the modified membrane were increased [39]. The increase in hydrophilicity and porosity enhances  
392 the selectivity of the membrane to pass water molecules through it. In the permeability test, the  
393 dead-end filtration cell was utilized. The flux and permeance were measured after every 10  
394 minutes, the final measurements were obtained after 80 minutes where all the membranes gave  
395 constant fluxes at a constant pressure of 2 bar. The pure water flux that was obtained for pristine  
396 PSU was very low  $17.91 \pm 2$  L/m<sup>2</sup>h which was extremely less than the modified membrane. This  
397 behaviour occurs because of the less bonding interaction and also due to the dense surface  
398 morphology (fewer and small surface pores) of the membrane).

399  
400 The PEG/CA-1 membrane showed maximum flux and permeability because of its less contact  
401 angle and high porosity which was  $45.5 \pm 2$  L/m<sup>2</sup>h and  $22.2 \pm 5$  L/m<sup>2</sup>h bar respectively. As shown  
402 in Fig. 7 that as the concentration of the mordenite zeolite increased it decreased the flux because  
403 the void spaces between the pores start blocking the membrane and making the membrane denser.  
404 Hemodialysis requires moderate water flux so that less water will be lost from the blood during  
405 the dialysis process so from the graph PEG/CA-3 membrane showed moderate flux and  
406 permeability that was  $41.224 \pm 2$  L/m<sup>2</sup>h and  $20.612 \pm 5$  L/m<sup>2</sup>h bar respectively as shown in Fig. 7

407 [9]. Similarly, when only PEG as a pore-modifier was added, the trend increased gradually as  
408 shown in Fig. 7 so PEG-3 can be considered as the better membrane with moderate  $24.901 \pm 2 \text{ L/m}^2$   
409 h of flux and  $12.445 \pm 5 \text{ L/m}^2 \text{ h bar}$  respectively of permeability.

#### 410 **3.2.4. Solute rejection and clearance percentage analysis**

411 The albumin loss can be controlled by the membrane morphology and composition to justify this  
412 BSA with a molecular weight of 67 kDa was used to determine the solute rejection percentage [9].  
413 Some membranes can have poor water flux but BSA retention should be higher than 75% for good  
414 dialysis treatment. For to characterize the loss of BSA during the 6 h dialysis simulation is used to  
415 determine the loss of beneficial proteins [40]. Fig. 8 showed that pristine PSU cannot reject BSA  
416 because of the dense surface of the membrane. Meanwhile, the maximum rejection of the BSA  
417 was obtained in the PEG/CA-1 membrane that was  $83.21\% \pm 5\%$  because of the irregular porous  
418 surface and also the sub pores in the fingers of the membrane. However, when only PEG as a pore-  
419 modifier was added, the BSA rejection increased to  $93.5\% \pm 5\%$  in the PEG-5 membrane as shown  
420 in Fig. 8. This difference in the membranes for BSA rejection justifies that optimization is required  
421 in the pore size of the PEG/CA membranes to increase the BSA rejections to a maximum  
422 percentage. All PSU family polymers need very optimized handling to make optimized pore size  
423 [41].

424  
425 As the uremic toxins also contain urea which is important to remove from the blood during the  
426 dialysis process. Fig. 8 shows that PEG/CA-1 membrane gave clearance of  $72\% \pm 5\%$ , which was  
427 more than the pristine PSU membrane with the increase in water flux. The urea clearance then  
428 decreased after that due to denser membranes because of the presence of high concentrations of  
429 mordenite zeolite and blockage of the pores with urea molecules coming from different directions.

430 For PLA membrane the maximum urea clearance was over 70% [42]. From Fig. 8 when only PEG  
431 acts as an additive, the urea clearance reached  $93\% \pm 5\%$ . In the literature, when using CA as a base  
432 polymer the maximum clearance for the urea was  $80.39\% \pm 5\%$ . [43]. So, with the addition of only  
433 PEG as a pore-modifier, the pores were uniformly scattered throughout the membrane, hence it  
434 gives maximum urea clearance this statement can also be justified by the SEM morphology Fig.  
435 2(d, e, f). But here with the increase in the concentration of mordenite zeolite the urea clearance  
436 also increases because the mordenite zeolite provides more surface area along with the macro-void  
437 formation with the pores.

### 438 **3.2.5. Creatinine adsorption capacity by composite membranes**

439 Creatinine is a uremic toxin formed in the muscles by the degradation of creatine phosphate. In  
440 relation to the degree of creatinine absorption, the size and shape of zeolite particles can  
441 theoretically affect the efficiency of the membranes. The effect of the concentration of mordenite  
442 zeolite inside the membrane on the creatinine uptake level was observed. The mordenite zeolite  
443 was selected as it has a spherical shape and the size of the particle is 48 nm in diameter that will  
444 provide more adsorption sites for the adsorption of creatinine. From the literature, the spherical-  
445 shaped particles work better inside the membranes than rod-shaped zeolite [44]. As the powdered  
446 zeolite can adsorb more creatinine rather than when it was incorporated inside the membrane this  
447 is just because 1/3 of the adsorption site of the nanoparticle was blocked when particles were  
448 incorporated on the surface but also inside the fingers of the membranes [21].

449

450 From Fig. 9, the trend explains that as the concentration of the mordenite zeolite increases in the  
451 composition of the membranes the adsorption capacity of the creatinine also increased hence  
452 making the membrane more suitable for the removal of protein-bound toxins. Hence the PEG/CA-

453 5 with 9643  $\mu\text{g}/\text{gm}$  can adsorb maximum creatinine compared to the PEG-5 membrane with 7654  
454  $\mu\text{g}/\text{g}$  as shown in Fig. 9. This is because the pore size is small and are less in number so when the  
455 nano-particles were incorporated the reactive sites are masked in the membrane. As the shape and  
456 size of the nanoparticle integrated inside the membrane had a great effect on the creatinine  
457 adsorption as in literature while using PAN as a polymer along with the rod-shaped zeolite particle  
458 the creatinine adsorption was 7000  $\mu\text{g}/\text{gm}$  [21].

### 459 **3.3 Biocompatibility evaluation of membranes**

460 The objective of this research was to reduce the number of platelets on the surface of the  
461 membrane. SEM photographs have been utilized to observe the platelet adhesion behaviour over  
462 the skin of the membrane. As shown in Fig. 10, the overall surface of the pristine PSU membrane  
463 had evident platelet adsorption. In addition, all reticulate pseudopodia structures displayed  
464 adhesion of platelets, suggesting activation of platelets. But the involvement of ether bond in PEG  
465 by hydrogen bonding can be closely coupled with water molecules that form a hydration layer  
466 (physical and energetic barrier) near the surface with the addition of PEG/CA and mordenite  
467 zeolite to keep the bio components from adsorbing the polymer surface as also shown in Fig. 3 of  
468 FT-IR spectrum [45]. From the literature, a large number of platelets aggregated on the  
469 hydrophobic membrane surfaces such as pristine PSU or PLA membranes [44]. Therefore, the  
470 PEG/CA-1 layer, expressed on the membrane surface by PEG chains, enables the best anti-protein  
471 surface that effectively inhibits platelet accumulation. It can be assumed that this strong  
472 performance of anti-surface assimilation can be sustained for a long time rather than steadily  
473 decreasing over time since PEG and PSU are covalently bonded with each other. Although the  
474 platelet cannot adequately aggregate on membrane surface due to the larger pore sizes, hence less  
475 platelet adhesion can be observed in Fig. 10(b) as compared to Fig. 10(c, d).

476

477 The thrombus formation was being examined by using the whole blood. In the formulation of the  
478 blood-contacting membrane, the main obstacle is self-induced thrombosis [45]. Therefore, as the  
479 pristine PSU membrane was highly hydrophobic in nature due to less hydrogen bonding justified  
480 by the contact angle measurement from Fig. 5(b), hence it contained the highest thrombus  
481 formation value which was  $9\% \pm 0.3\%$  as the platelets were highly aggregated on the surface of the  
482 membrane. However, with the addition of additives, the thrombus formation decreases. From Fig.  
483 11(a), when PEG along with CA act as a pore-modifier the minimum value was obtained for  
484 PEG/CA-1 membrane which was  $5\% \pm 0.3\%$ . When only PEG acts as a pore-modifier the minimum  
485 value of thrombus formation was  $5.06\% \pm 0.3\%$  for PEG-1 membrane. However, from Fig. 11(a)  
486 the trend explains that with the increase in the concentration of the mordenite zeolite the thrombus  
487 formation also increased slightly. So, the more accumulation of the platelets on the surface of the  
488 membrane occurs the more thrombus formation value is obtained.

489

490 Because of the association of erythrocytes with the membranes, erythrocytes can burst and let out  
491 hemoglobin (known as hemolysis). To assess the degree of damage to the erythrocytes by the  
492 dialysis membranes, HR is then used. The ASTM F-756-08 finds that HR below 5% is considered  
493 to be harmless. In comparison to all of the polymers, the pristine PSU membrane gave  $0.55\% \pm$   
494  $0.03\%$  however with the addition of PEG along with CA the value decreased to  $0.37\% \pm 0.03\%$   
495 for PEG/CA-1 membrane which was extremely lesser than 5% hence proving that the membranes  
496 have excellent hydrophilicity, electronegativity and anti hemolytic activity justified by the contact  
497 angle. Similarly, from Fig. 11(a), when only PEG acts as a pore-modifier the values were  $0.46\% \pm$   
498  $0.03\%$  for PEG-1 membrane which was lesser than 5% and still lesser than in the literature as it

499 reduces the damage to erythrocytes with addition to blood clotting and platelet adhesion prevention  
500 [45]. The slight increase in the trend can be seen due to the decreased porosity and contact angle  
501 measurement as shown in Fig. 5(b) [41]. Certain polymers give certain hemolysis ratios like when  
502 PSF/PSF-*g*-TPG is used the HR was 0.53% [11].

503  
504 The clotting time and the presence and absence of clotting factor can be determined by PRT [31].  
505 When the blood meets activated factor VIII and the presence of  $\text{Ca}^{2+}$ , fibrous proteins cross-linked  
506 with each other lead to the formation of thrombus. Thrombus formation time depends upon the  
507 hydrophilicity and presence of hydroxyl, carboxyl groups [32]. The PRT of the pristine PSU  
508 membrane was  $321\text{s} \pm 20\text{s}$  as shown in Fig. 11(b). As the thrombus formation is greatly reliant on  
509 the hydrophilicity of the membrane. PEG is considered as a hydrophilic and biocompatible  
510 material which is the main cause of increment in the PRT because the functional groups increase  
511 plasma slowly forms the adsorptive layer on the surface resulting in the enhancement of the  
512 biocompatibility. So with the addition of the additive (PEG), the PRT increased to  $392\text{s} \pm 20\text{s}$  and  
513 when (CA along with PEG) was added the PRT increased to  $490\text{s} \pm 20\text{s}$  which proved that the  
514 activation of fibrinogen on PEG/CA-1 membranes was repressed, due to the improvement of  
515 membrane hydrophilicity [46]. But due to the increase in the concentration of the mordenite  
516 zeolite, the PRT starts decreasing as the membrane surface starts becoming hydrophobic in nature  
517 Fig. 11 (b) justifies the statement.

#### 518 **4 Conclusion**

519 Fabrication and characterization of Poly-sulfone hemodialysis membranes with better  
520 biocompatibility and uremic toxin clearance were obtained by the addition of hydrophilic  
521 compounds like PEG and CA. The spherical structure and uniform-sized zeolite particles named

522 mordenite zeolite adsorb more medium toxins hence provided the maximum reactive site while it  
523 was incorporated in the membrane. When PEG along with the CA was added in membrane  
524 solution, membrane became more hydrophilic such that 9643  $\mu\text{g/gm}$  of creatinine were adsorbed  
525 along with plasma recalcification time of 490s along with the lowest hemolysis ratio that is 0.37%  
526 but the solute rejection was only 83% as compared to PEG membranes. However, when only PEG  
527 was added to the membrane solution, creatinine adsorption was 7654  $\mu\text{g/g}$  with less PRT 392s but  
528 the maximum solute rejection obtained was 93%. The modified membranes also showed excellent  
529 stability in water. Medium toxins like indoxyl sulfate and p-cresol adsorption and adsorption effect  
530 on pH and salts tests can be performed to justify that with the smaller sized zeolite particle more  
531 amount of medium toxins can be adsorbed in the membrane.

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536

537 **References**

- 538 [1] S. K. Verma, A. Modi, A. K. Singh, R. Teotia, and J. Bellare, “Improved hemodialysis with  
539 hemocompatible polyethersulfone hollow fiber membranes: *In vitro* performance,” *Journal*  
540 *of Biomedical Materials Research - Part B Applied Biomaterials*, vol. 106, no. 3, pp. 1286–  
541 1298, 2018.
- 542 [2] V. Jha, “Current status of end-stage renal disease care in India and Pakistan,” *Kidney*  
543 *International Supplements*, vol. 3, no. 2, pp. 157–160, 2013.
- 544 [3] R. Sinnakirouchenan and J. L. Holley, “Peritoneal dialysis versus hemodialysis: Risks,  
545 benefits, and access issues,” *Advances in Chronic Kidney Diseases*, vol. 18, no. 6, pp. 428–  
546 432, 2011.
- 547 [4] H. Wang *et al.*, “Enhanced hemocompatibility of flat and hollow fiber membranes via a  
548 heparin free surface crosslinking strategy,” *Reactive and Functional Polymer*, vol. 124, no.  
549 December 2017, pp. 104–114, 2018.
- 550 [5] M. Irfan, A. Idris, N. M. Yusof, N. F. M. Khairuddin, and H. Akhmal, “Surface modification  
551 and performance enhancement of nano-hybrid f-MWCNT/PVP90/PES hemodialysis  
552 membranes,” *Journal of Membrane Sciences*, vol. 467, pp. 73–84, 2014.
- 553 [6] L. Ma *et al.*, “Hemocompatible poly(lactic acid) membranes prepared by immobilizing  
554 carboxylated graphene oxide: Via mussel-inspired method for hemodialysis,” *RSC*  
555 *Advances*, vol. 8, no. 1, pp. 153–161, 2018.
- 556 [7] X. Yu *et al.*, “High performance thin-film nanofibrous composite hemodialysis membranes  
557 with efficient middle-molecule uremic toxin removal,” *Journal of Membrane Sciences*, vol.  
558 523, pp. 173–184, 2017.
- 559 [8] F. Galiano, K. Briceño, T. Marino, A. Molino, K. V. Christensen, and A. Figoli, “Advances  
560 in biopolymer-based membrane preparation and applications,” *Journal of Membrane*  
561 *Sciences*, vol. 564, no. May, pp. 562–586, 2018.
- 562 [9] D. Zhong, Z. Wang, J. Zhou, and Y. Wang, “Additive-free preparation of hemodialysis  
563 membranes from block copolymers of polysulfone and polyethylene glycol,” *Journal of*  
564 *Membrane Sciences*, vol. 618, no. August 2020, p. 118690, 2021.
- 565 [10] O. E. M. ter Beek, D. Pavlenko, and D. Stamatialis, “Hollow fiber membranes for long-  
566 term hemodialysis based on polyethersulfone-SlipSkin™ polymer blends,” *Journal of*  
567 *Membrane Sciences*, vol. 604, no. January, p. 118068, 2020.
- 568 [11] G. J. Dahe, R. S. Teotia, S. S. Kadam, and J. R. Bellare, “The biocompatibility and  
569 separation performance of antioxidative polysulfone/vitamin E TPGS composite hollow  
570 fiber membranes,” *Biomaterials*, vol. 32, no. 2, pp. 352–365, 2011.
- 571 [12] T. M. Liu, J. J. Xu, and Y. R. Qiu, “A novel kind of polysulfone material with excellent  
572 biocompatibility modified by the sulfonated hydroxypropyl chitosan,” *Materials Science*  
573 *and Engineering C*, vol. 79, pp. 570–580, 2017.
- 574 [13] E. Salehi and S. S. Madaeni, “Influence of poly(ethylene glycol) as pore-generator on  
575 morphology and performance of chitosan/poly(vinyl alcohol) membrane adsorbents,”  
576 *Applied Surface Sciences*, vol. 288, pp. 537–541, 2014.
- 577 [14] K. Namekawa, M. Matsuda, M. Fukuda, A. Kaneko, and K. Sakai, “Poly(N-vinyl-2-  
578 pyrrolidone) elution from polysulfone dialysis membranes by varying solvent and wall  
579 shear stress,” *Journal of Artificial Organs*, vol. 15, no. 2, pp. 185–192, 2012.
- 580 [15] M. Kohlová, C. G. Amorim, A. da Nova Araújo, A. Santos-Silva, P. Solich, and M. C. B.  
581 S. M. Montenegro, “In vitro assessment of polyethylene glycol and polyvinylpyrrolidone as

- 582 *hydrophilic additives on bioseparation by polysulfone membranes,*” Journal of Materials  
583 Science, vol. 55, no. 3, pp. 1292–1307, 2020.
- 584 [16] R. Vanholder, E. Schepers, A. Pletinck, E. V. Nagler, and G. Glorieux, “*The uremic toxicity*  
585 *of indoxyl sulfate and p-cresyl sulfate: A systematic review,*” Journal of the American  
586 Society of Nephrology, vol. 25, no. 9, pp. 1897–1907, 2014.
- 587 [17] O. Stress, “Hemodialysis Impairs Endothelial Function,” pp. 1002–1007, 2015.
- 588 [18] Y. Itoh, A. Ezawa, K. Kikuchi, Y. Tsuruta, and T. Niwa, “*Protein-bound uremic toxins in*  
589 *hemodialysis patients measured by liquid chromatography/tandem mass spectrometry and*  
590 *their effects on endothelial ROS production,*” Analytical and Bioanalytical Chemistry, vol.  
591 403, no. 7, pp. 1841–1850, 2012.
- 592 [19] M. Ji, X. Chen, J. Luo, and Y. Wan, “*Improved blood compatibility of polysulfone*  
593 *membrane by anticoagulant protein immobilization,*” Colloids Surfaces B Biointerfaces,  
594 vol. 175, no. December 2018, pp. 586–595, 2019.
- 595 [20] M. Tijink *et al.*, “*Development of novel membranes for blood purification therapies based*  
596 *on copolymers of N-vinylpyrrolidone and n-butylmethacrylate,*” Journal of Materials  
597 Chemistry B, vol. 1, no. 44, pp. 6066–6077, 2013.
- 598 [21] L. Lu, C. Chen, C. Samarasekera, and J. T. W. Yeow, “*Influence of zeolite shape and*  
599 *particle size on their capacity to adsorb uremic toxin as powders and as fillers in*  
600 *membranes,*” Journal of Biomedical Materials Research - Part B Applied Biomaterials, vol.  
601 105, no. 6, pp. 1594–1601, 2017.
- 602 [22] A. Idris and L. K. Yet, “*The effect of different molecular weight PEG additives on cellulose*  
603 *acetate asymmetric dialysis membrane performance,*” Journal of Membrane Sciences, vol.  
604 280, no. 1–2, pp. 920–927, 2006.
- 605 [23] Q. Chen, Y. He, Y. Zhao, and L. Chen, “*Tannic acid and Poly(N-acryloyl morpholine)*  
606 *layer-by-layer built hemodialysis membrane surface for intervening oxidative stress*  
607 *integrated with high biocompatibility and dialysis performance,*” Journal of Membrane  
608 Sciences, vol. 621, p. 118896, 2021.
- 609 [24] O. T. Mahlangu, R. Nackaerts, J. M. Thwala, B. B. Mamba, and A. R. D. Verliefde,  
610 “*Hydrophilic fouling-resistant GO-ZnO/PES membranes for wastewater reclamation,*”  
611 Journal of Membrane Sciences, vol. 524, no. November 2016, pp. 43–55, 2017.
- 612 [25] H. Waheed, F. T. Minhas, and A. Hussain, “*Cellulose acetate/sericin blend membranes for*  
613 *use in dialysis,*” Polymer Bulletin, vol. 75, no. 9, pp. 3935–3950, 2018.
- 614 [26] D. Zhong, Z. Wang, Q. Lan, and Y. Wang, “*Selective swelling of block copolymer*  
615 *ultrafiltration membranes for enhanced water permeability and fouling resistance,*” Journal  
616 of Membrane Sciences, vol. 558, no. April, pp. 106–112, 2018.
- 617 [27] H. Waheed and A. Hussain, “*Preparation and Solvents Effect Study of Asymmetric*  
618 *Cellulose Acetated/Polyethyleneimine Blended Membranes for Dialysis Application,*”  
619 International Journal of Health and Medicines, vol. 2, no. 4, p. 5, 2017.
- 620 [28] A. Bernal-Ballén, I. Kuritka, and P. Saha, “*Preparation and Characterization of a*  
621 *Bioartificial Polymeric Material: Bilayer of Cellulose Acetate-PVA,*” International Journal  
622 of Polymeric Sciences, vol. 2016, 2016.
- 623 [29] L. Zhu, F. Liu, X. Yu, A. Gao, and L. Xue, “*Surface zwitterionization of hemocompatible*  
624 *poly (lactic acid) membranes for hemodia fi ltration,*” Journal of Membrane Sciences, vol.  
625 475, pp. 469–479, 2015.
- 626 [30] S. Senthilkumar, S. Rajesh, A. Jayalakshmi, and D. Mohan, “*Biocompatibility studies of*  
627 *polyacrylonitrile membranes modi fi ed with carboxylated polyetherimide ☆,*” Materials

- 628 Science and Engineering C, vol. 33, no. 7, pp. 3615–3626, 2013.
- 629 [31] A. Gao, F. Liu, and L. Xue, “Preparation and evaluation of heparin-immobilized poly  
630 (lactic acid) (PLA) membrane for hemodialysis,” *Journal of Membrane Sciences*, vol. 452,  
631 pp. 390–399, 2014.
- 632 [32] X. Yu, Y. Zhu, C. Cheng, T. Zhang, X. Wang, and B. S. Hsiao, “Novel thin-film nanofibrous  
633 composite membranes containing directional toxin transport nanochannels for efficient and  
634 safe hemodialysis application,” *Journal of Membrane Sciences*, vol. 582, no. April, pp.  
635 151–163, 2019.
- 636 [33] A. Mollahosseini, A. Abdelrasoul, and A. Shoker, “A critical review of recent advances in  
637 hemodialysis membranes hemocompatibility and guidelines for future development,”  
638 *Materials Chemistry and Physics*, vol. 248, no. March, p. 122911, 2020.
- 639 [34] R. Teotia, S. K. Verma, D. Kalita, A. K. Singh, G. Dahe, and J. Bellare, “Porosity and  
640 compatibility of novel polysulfone-/vitamin E-TPGS-grafted composite membrane,”  
641 *Journal of Materials. Science*, vol. 52, no. 20, pp. 12513–12523, 2017.
- 642 [35] S. Saki and N. Uzal, “Preparation and characterization of PSF/PEI/CaCO<sub>3</sub> nanocomposite  
643 membranes for oil/water separation,” *Environmental Science and Pollution Research*, vol.  
644 25, no. 25, pp. 25315–25326, 2018.
- 645 [36] E. Saljoughi, M. Amirilargani, and T. Mohammadi, “Effect of PEG additive and  
646 coagulation bath temperature on the morphology, permeability and thermal/chemical  
647 stability of asymmetric CA membranes,” *Desalination*, vol. 262, no. 1–3, pp. 72–78, 2010,
- 648 [37] J. Yin, H. Fan, and J. Zhou, “Cellulose acetate/poly(vinyl alcohol) and cellulose  
649 acetate/crosslinked poly(vinyl alcohol) blend membranes: preparation, characterization,  
650 and antifouling properties,” *Desalination and Water Treatment*, vol. 57, no. 23, pp. 10572–  
651 10584, 2016.
- 652 [38] X. Yu, X. Mi, Z. He, M. Meng, H. Li, and Y. Yan, “Fouling resistant CA/PVA/TiO<sub>2</sub>  
653 imprinted membranes for selective recognition and separation salicylic acid from waste  
654 water,” *Frontiers in Chemistry*, vol. 5, no. JAN, pp. 1–13, 2017.
- 655 [39] A. Gao, F. Liu, H. Shi, and L. Xue, “Controllable transition from finger-like pores to inter-  
656 connected pores of PLLA membranes,” *Journal of Membrane Sciences*, vol. 478, pp. 96–  
657 104, 2015.
- 658 [40] X. Yu, Z. Xiong, J. Li, Z. Wu, Y. Wang, and F. Liu, “Surface PEGylation on PLA  
659 membranes via micro-swelling and crosslinking for improved  
660 biocompatibility/hemocompatibility,” *RSC Advances*, vol. 5, no. 130, pp. 107949–107956,  
661 2015.
- 662 [41] H. Song, F. Ran, H. Fan, X. Niu, L. Kang, and C. Zhao, “Hemocompatibility and  
663 ultrafiltration performance of surface-functionalized polyethersulfone membrane by  
664 blending comb-like amphiphilic block copolymer,” *Journal of Membrane Sciences*, vol.  
665 471, pp. 319–327, 2014.
- 666 [42] Z. Xiong, F. Liu, H. Lin, J. Li, and Y. Wang, “Covalent Bonding of Heparin on the  
667 Crystallized Poly(lactic acid) (PLA) Membrane to Improve Hemocompatibility via Surface  
668 Cross-Linking and Glycidyl Ether Reaction,” *ACS Biomaterial Sciences and Engineering*,  
669 vol. 2, no. 12, pp. 2207–2216, 2016.
- 670 [43] Y. Raharjo, S. Wafiroh, M. Nayla, V. Yuliana, and M. Z. Fahmi, “Primary study of  
671 cellulose acetate hollow fiber as a green membrane applied to hemodialysis,” *Journal of*  
672 *Chemical Technology and Metallurgy*, vol. 52, no. 6, pp. 1021–1026, 2017.
- 673 [44] R. Takai, R. Kurimoto, Y. Nakagawa, Y. Kotsuchibashi, K. Namekawa, and M. Ebara,

- 674            *“Towards a Rational Design of Zeolite-Polymer Composite Nanofibers for Efficient*  
675            *Adsorption of Creatinine,”* Journal of Nanomater., vol. 2016, 2016.
- 676 [45] S. Mulyati, S. Aprilia, Safiah, Syawaliah, M. A. Armando, and H. Mawardi, *“The effect of*  
677            *poly ethylene glycol additive on the characteristics and performance of cellulose acetate*  
678            *ultrafiltration membrane for removal of Cr(III) from aqueous solution,”* IOP Conference.  
679            Series: Material Science Engineering, vol. 352, no. **1**, 2018.
- 680 [46] C. Liu *et al.*, *“PMWCNT/PVDF ultrafiltration membranes with enhanced antifouling*  
681            *properties intensified by electric field for efficient blood purification,”* Journal of  
682            Membrane Sciences., vol. 576, no. December 2018, pp. 48–58, 2019.
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684

685

## List of Tables

686

687 **Table 1.** Recipe of pristine PSU and modified membranes with phase inversion technique.

688

Membrane	PSU (wt%)	Sol (mL)	PEG (wt%)	Sol (mL)	CA wt%	Sol (mL)	Mordenite Zeolite
PSU	18	-	-	-	-	-	-
PEG/CA-1	18	7	16	4	8	2	0.18
PEG/CA-3	18	7	16	4	8	2	0.48
PEG/CA-5	18	7	16	4	8	2	0.98
PEG-1	18	7	16	4	-	-	0.18
PEG-3	18	7	16	4	-	-	0.48
PEG-5	18	7	16	4	-	-	0.98

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**Table 2.** Comparison of results of Pristine PSU and PSU (with additives).

<b>Membrane materials</b>	<b>BSA rejection (%)</b>	<b>Urea toxins clearance (%)</b>	<b>Creatinine adsorption (<math>\mu\text{g/gm}</math>)</b>	<b>Thrombus formation (%)</b>	<b>Hemolysis ratio (%)</b>	<b>Recalcification time (s)</b>
PSU	N/A	N/A	N/A	N/A	0.55	311
PEG/CA	83	74	9643	4.90	0.37	490
PEG	93	89	7654	5.04	0.46	392

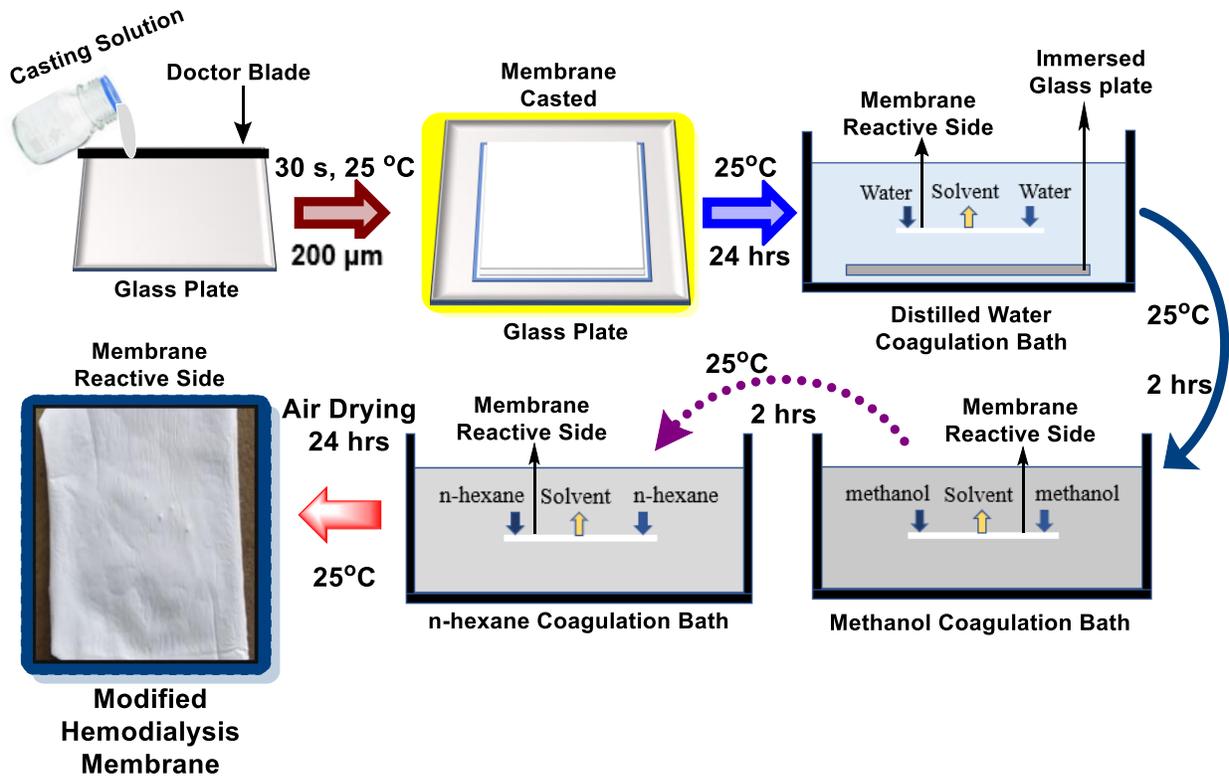
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## List of Figures

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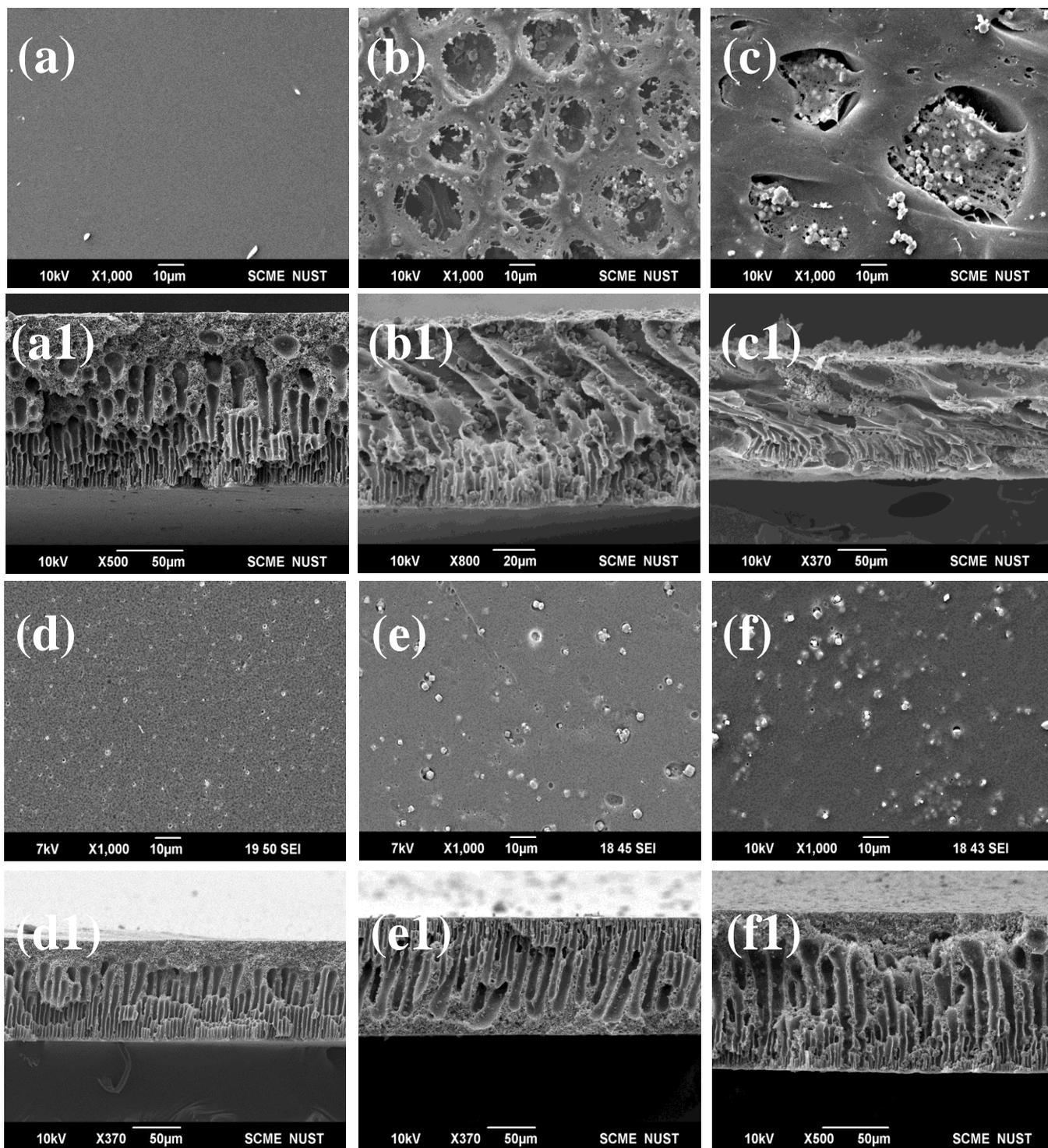
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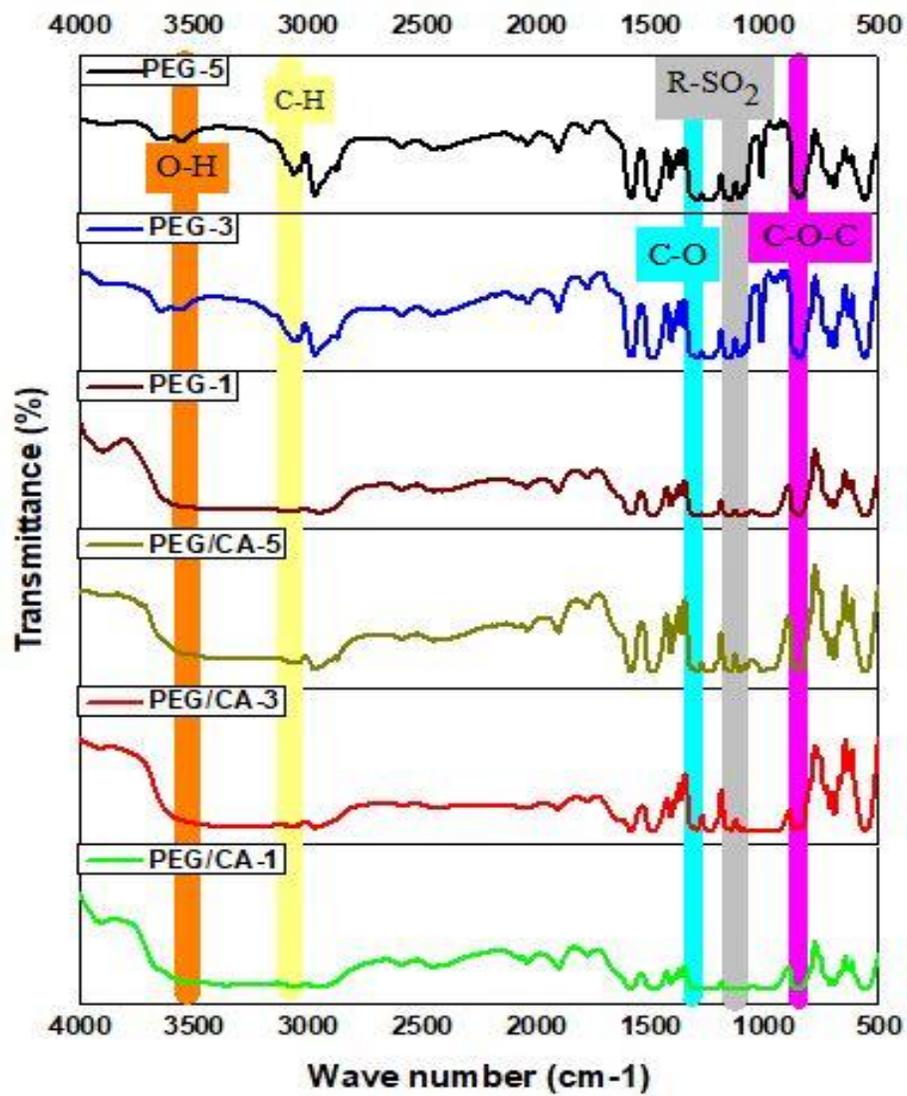
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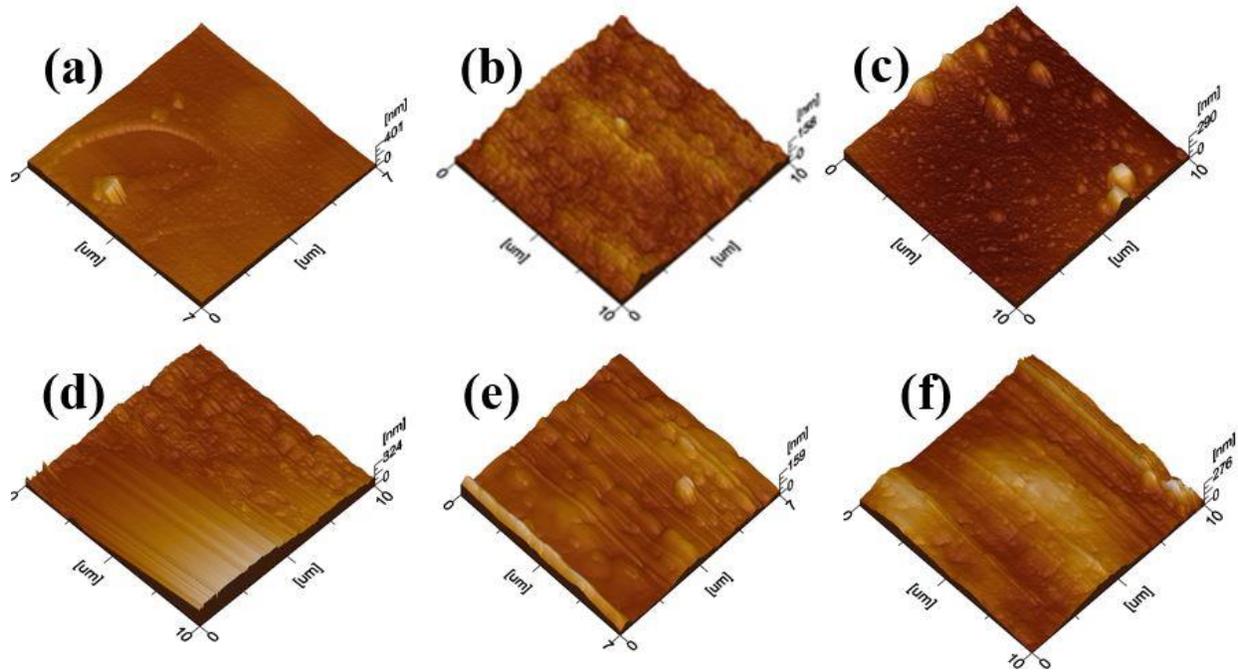
**Fig. 1.** Fabrication of the membranes by phase inversion technique.



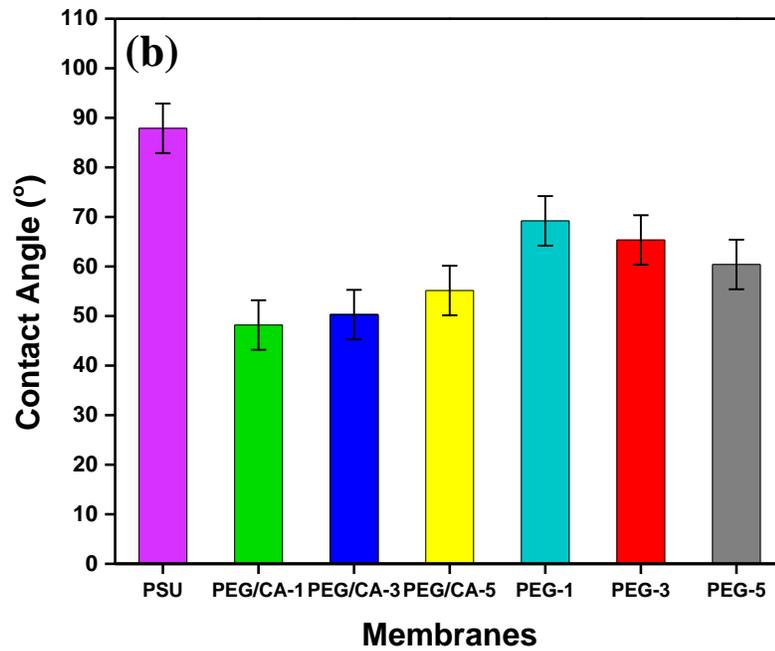
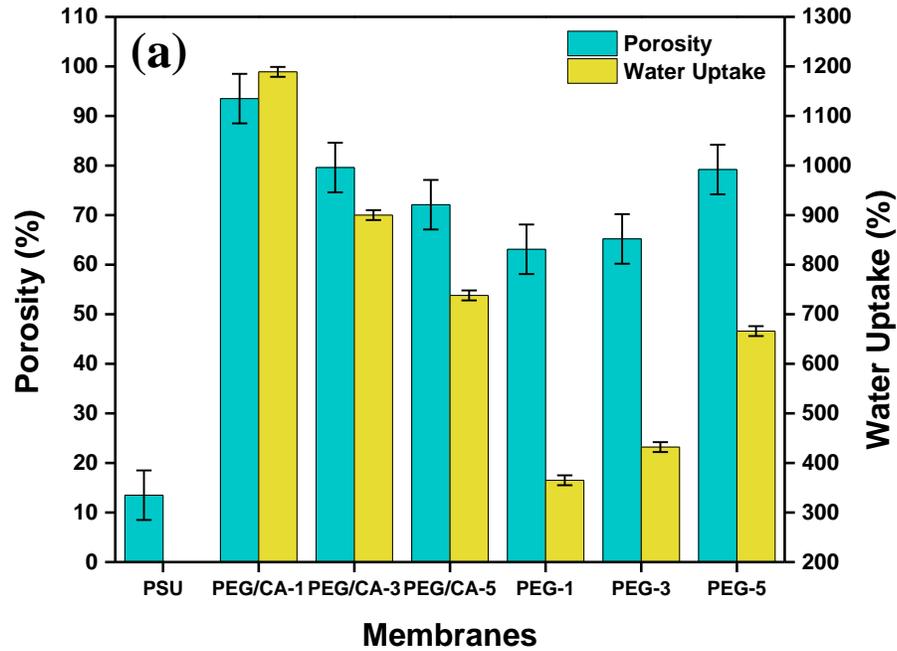
**Fig. 2.** SEM imaging of the surface (top) and cross-sectional (bottom) morphology of the membranes; (a) PSU (b) PEG/CA-1, (c) PEG/CA-3, (d) PEG-1, (e) PEG-3, (f) PEG-5.



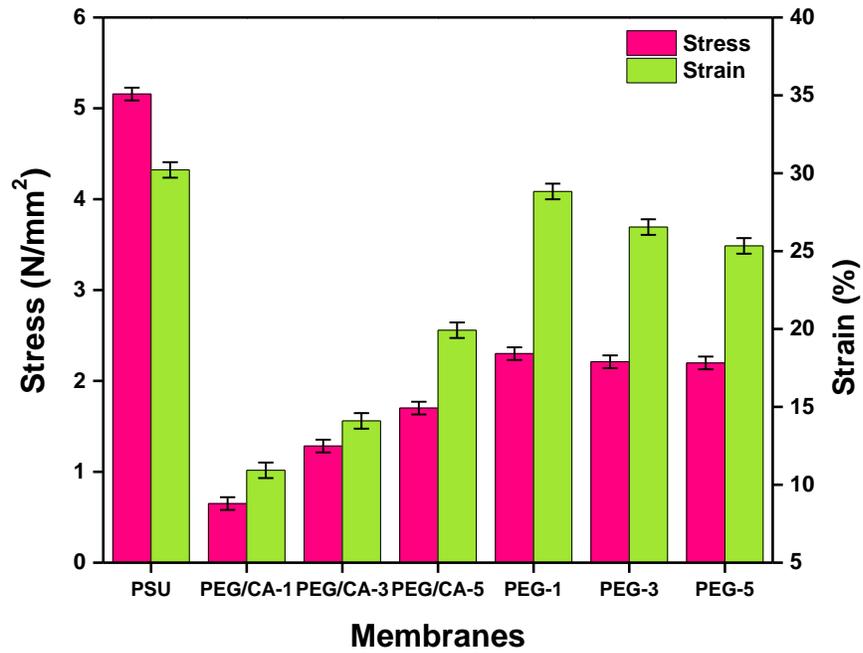
**Fig. 3.** FT-IR spectrum of modified membranes when PEG and CA, when PEG alone act as an additive.



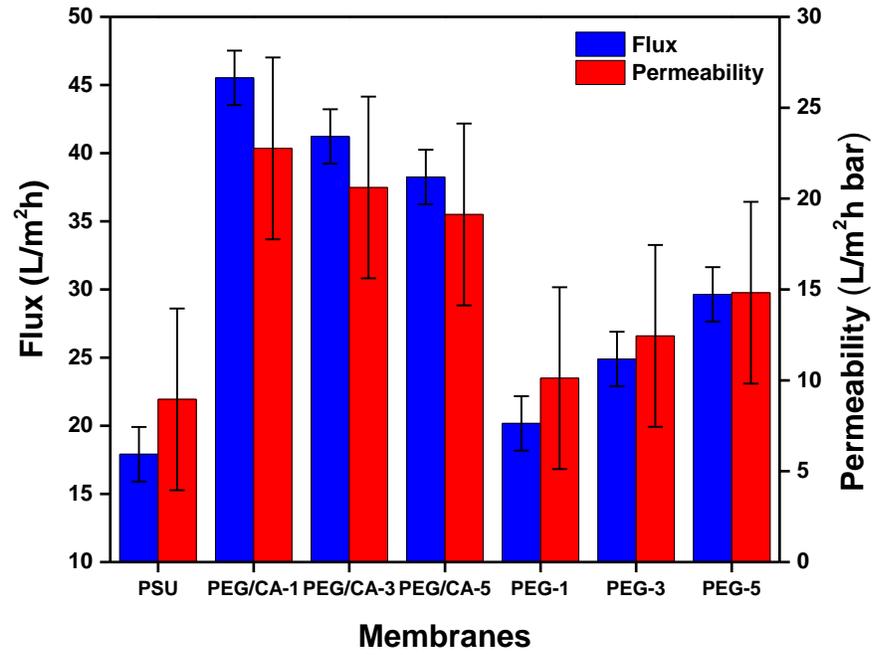
**Fig. 4.** The AFM imaging to determine the membrane's active layer roughness of pristine PSU and modified membranes; (a) PSU, (b) PEG/CA-1, (c) PEG/CA-3, (d) PEG-1, (e) PEG-3 and (f) PEG-5.



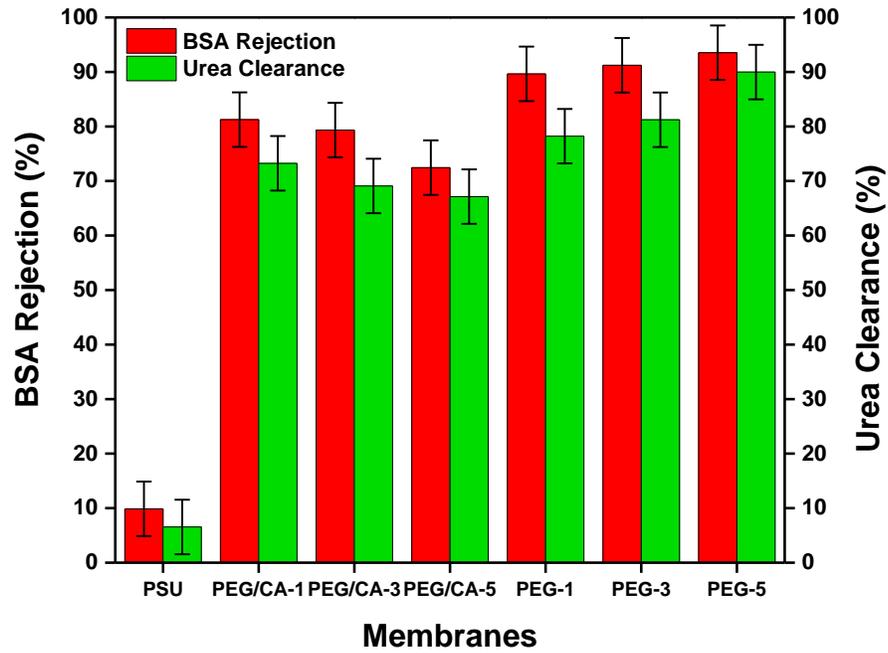
**Fig. 5.** Measurement of hydrophilicity and hydrophobicity under; (a) The porosity and swelling of the pristine PSU and modified porous membranes after the addition of additives and mordenite zeolite (b) Contact angle measurement of pristine PSU and composite membranes.



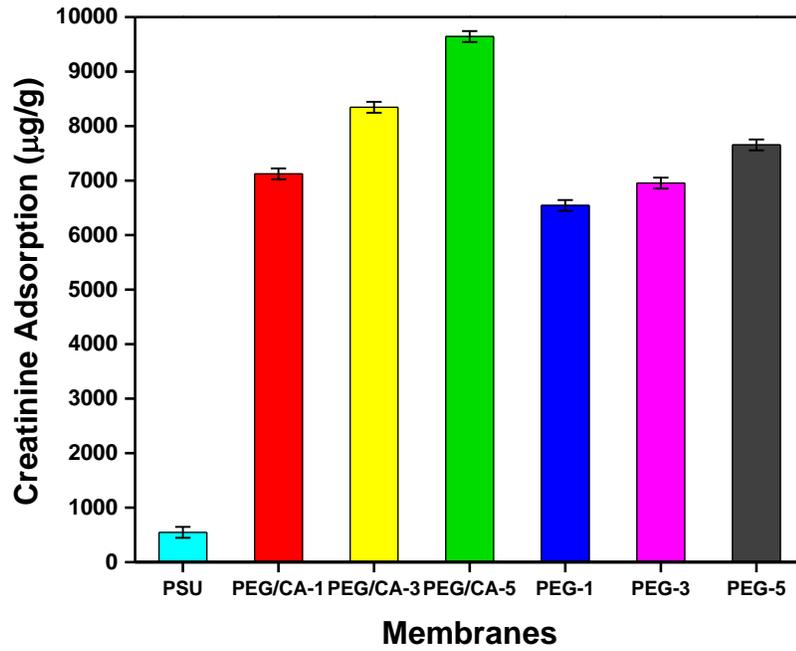
**Fig. 6.** Stress-strain curve of pristine PSU and modified membranes after the addition of pore-generators along with different concentrations of mordenite zeolite



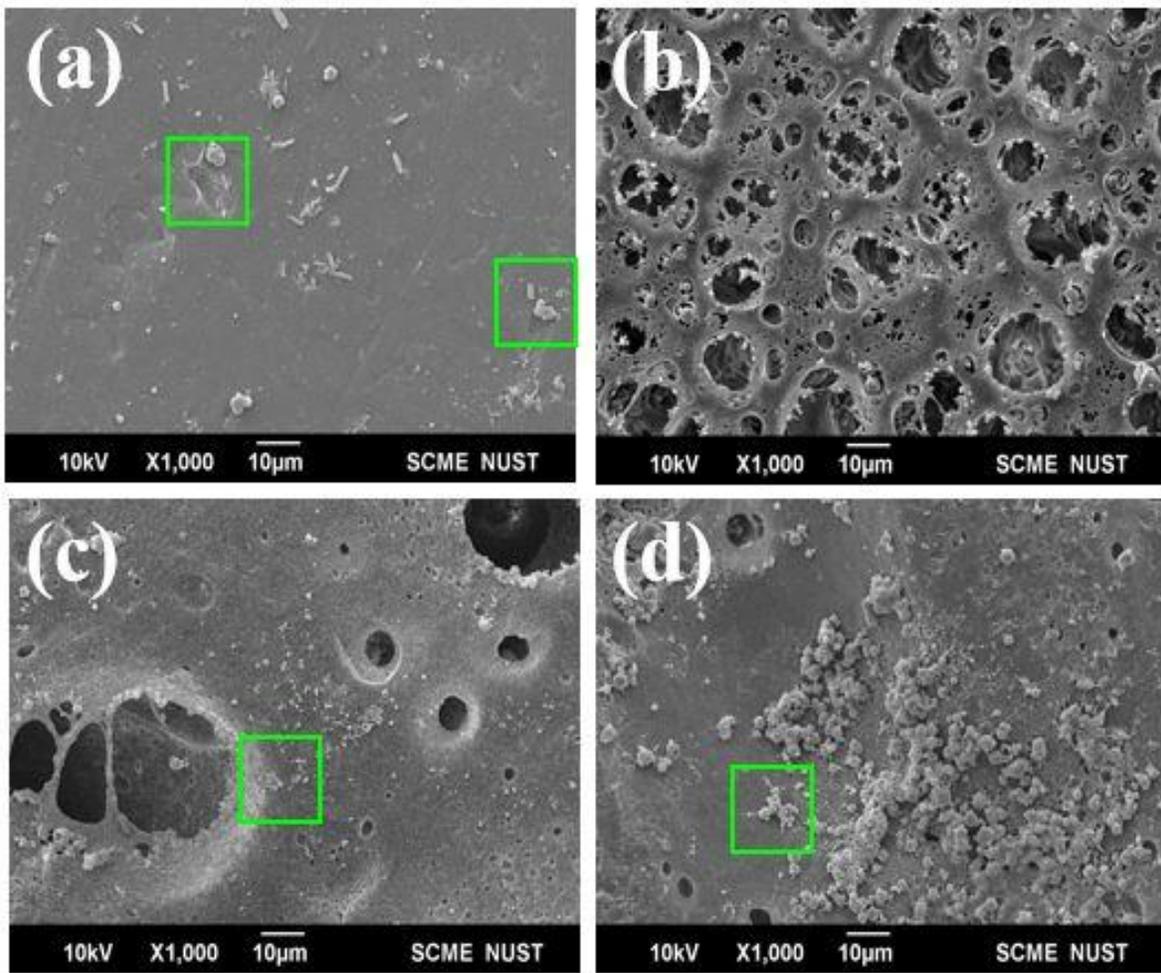
**Fig. 7.** Water flux and permeability of the membranes when static water was obtained in the permeate in the dead-end filtration cell at (80 minutes).



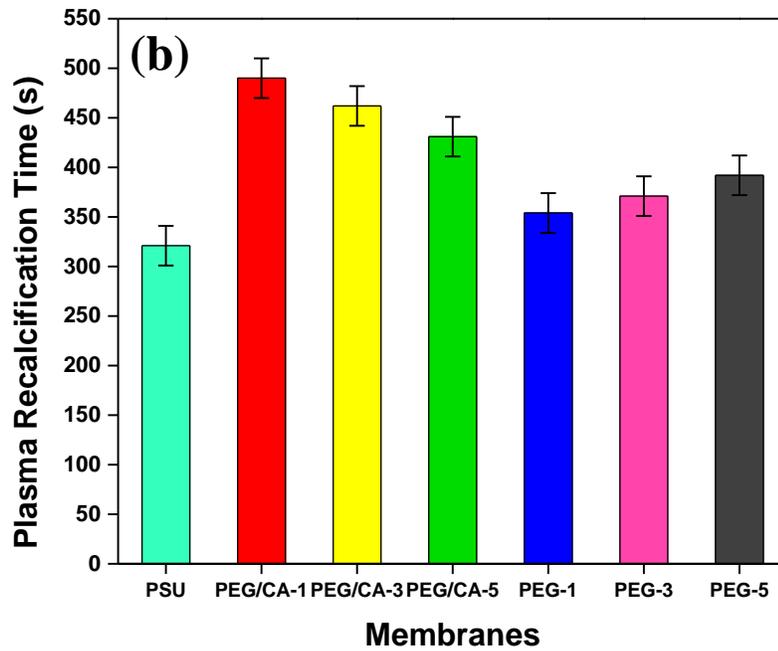
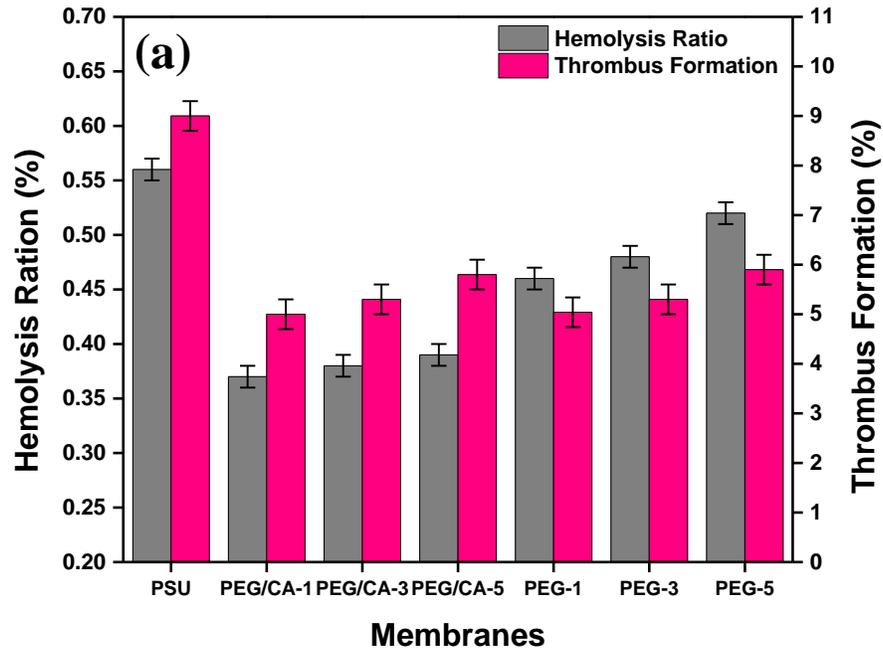
**Fig. 8.** Urea clearance, BSA rejection of the pristine PSU and modified membranes with different concentrations of mordenite zeolite after 4 hours simulating in dead-end filtration cell.



**Fig. 9.** Comparison of creatinine adsorption capacity of mordenite zeolite in membranes (by membrane mass and by zeolite mass).



**Fig. 10.** SEM images of the platelet gathered on pristine PSU and modified membranes; (a) PSU, (b) PEG/CA-1 (No platelet adhesion), (c) PEG-1 and (d) PEG-3.



**Fig. 11.** Hemocompatibility evaluation under; (a) Thrombus formation and Hemolysis ratio of pristine PSU and modified membranes, (b) Improved clotting time of the fabricated membranes with different concentration of mordenite zeolite.